

SHORT COMMUNICATIONS

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Copper-tin sulphides in the system Cu-Sn-S

It is well known that copper and tin form complex sulphides. In two preliminary studies, Moh (1960, 1963) identified four phases in the system Cu-Sn-S. Both Cu_2SnS_3 and $\text{Cu}_2\text{Sn}_2\text{S}_5$ are on the join Cu₂S-SnS₂. Cu_2SnS_3 is characterized by a tetragonal-cubic transition at 780°C, whereas $\text{Cu}_2\text{Sn}_2\text{S}_5$ exhibits an extensive range of solid solution between Cu_2SnS_3 and SnS₂. The other two phases, Cu_5SnS_4 and Cu_5SnS_6 , are stable below 600°C. Subsequent studies suggested that $\text{Cu}_2\text{Sn}_2\text{S}_5$ and Cu_5SnS_4 be discarded (Wang, 1974), and seven new copper-tin sulphides were proposed: $\text{Cu}_2\text{Sn}_3\text{S}_7$ (Wang, 1974; Sobott and Teh, 1977), Cu_4SnS_4 (Wang, 1974; Jaulmes *et al.*, 1977), $\text{Cu}_{9.67}\text{Sn}_{2.33}\text{S}_{13}$ (Wang, 1976), $\text{Cu}_5\text{Sn}_2\text{S}_7$, $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ (Wang, 1977), Cu_4SnS_6 (Wang, 1981) and Cu_3SnS_4 (Moh, 1982). This study examines their relations in the system Cu-Sn-S.

Experimental procedure. Experiments were performed by means of the conventional evacuated glass capsule technique using elements of 99.99% purity as specified by suppliers' analyses. Heat treatment was conducted in muffle furnaces controlled to $\pm 3^\circ\text{C}$. Quenched products were examined by X-ray powder diffraction, reflected light microscopy, and electron microprobe analysis. Cell dimensions $\pm 0.005 \text{ \AA}$ were computed by using a least-squares refinement program.

Results and discussion. Phase relations in the system Cu-Sn-S were studied in the temperature range between 400 and 650°C. Five copper-tin sulphides are stable in the system, $\text{Cu}_2\text{Sn}_3\text{S}_7$, Cu_2SnS_3 , Cu_4SnS_4 , $\text{Cu}_5\text{Sn}_2\text{S}_7$, and $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$. The latter is structurally an equivalent to $\text{Cu}_{9.67}\text{Sn}_{2.33}\text{S}_{13}$, but with a slightly different composition. X-ray powder diffraction data of $\text{Cu}_2\text{Sn}_3\text{S}_7$, Cu_2SnS_3 , Cu_4SnS_4 , and $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$ are listed in Table I.

Phase relations in a 500°C isothermal section are illustrated in fig. 1. Metallic liquid develops only in the Sn-rich portion of the system, and is isolated from sulphide assemblages, except SnS, by the join SnS-Cu₃Sn. In the S-rich portion, sulphur in the form of liquid exists in equilibrium with sulphides.

$\text{Cu}_2\text{Sn}_3\text{S}_7$ is monoclinic with cell dimensions of $a = 12.68$, $b = 7.351$, $c = 12.76 \text{ \AA}$, and $\beta = 109.60^\circ$. The X-ray powder diffraction pattern is very similar to that of rhodostannite, as Wang (1974) pointed

out, except for several weak reflection lines. Electron probe analysis give a formula $\text{Cu}_{2.1}\text{Sn}_{3.4}\text{S}_{7.5}$ comparable with Wang's $\text{Cu}_2\text{Sn}_{3.34}\text{S}_{7.68}$. Cu_2SnS_3 (Wang, 1974; Moh, 1976) has a triclinic superstructure of sphalerite, pseudomonoclinic in symmetry with cell dimensions of $a = 6.640$, $b = 11.53$, $c = 19.91 \text{ \AA}$, $\alpha = 90^\circ$, $\beta = 109.45^\circ$, and $\gamma = 90^\circ$ which are in good agreement with published data. Both phases are stable in the temperature range of this study and have no detectable range of solid solution. The composition of $\text{Cu}_2\text{Sn}_2\text{S}_5$ produced an assemblage of $\text{Cu}_2\text{Sn}_3\text{S}_7$ and Cu_2SnS_3 .

Cu_4SnS_4 was successfully synthesized, but with much difficulty from the required initial charge. A prolonged period of heat treatment of 120 days, compared to 60 days for the other phases, with intermediate grinding was necessary. It is orthorhombic with $a = 13.50$, $b = 7.66$, and $c = 6.395 \text{ \AA}$, and the formula derived from probe data is $\text{Cu}_{3.9}\text{Sn}_{1.1}\text{S}_4$. This phase forms binary assemblages with Cu_2S , SnS, Cu_2SnS_3 , and $\text{Cu}_5\text{Sn}_2\text{S}_7$.

The other two copper-tin sulphides in the system at 500°C have compositions of $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$ and $\text{Cu}_5\text{Sn}_2\text{S}_7$ assigned on the basis of probe compositions of $\text{Cu}_{9.7}\text{Sn}_{2.2}\text{S}_{13}$ and $\text{Cu}_{4.9}\text{Sn}_2\text{S}_7$. The former is comparable to $\text{Cu}_{9.67}\text{Sn}_{2.33}\text{S}_{13}$ (Wang, 1976), and the latter is identical with Wang's $\text{Cu}_5\text{Sn}_2\text{S}_7$ (1977). $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$ is tetragonal with $a = 9.540$ and $c = 10.93 \text{ \AA}$ which match well with $a = 9.53$ and $c = 10.93 \text{ \AA}$ of $\text{Cu}_{9.67}\text{Sn}_{2.33}\text{S}_{13}$. According to Wang (1976), $\text{Cu}_{9.67}\text{Sn}_{2.33}\text{S}_{13}$ coexists with CuS in samples of compositions with Cu/Sn ratio higher than 4.15, and coexists with Cu_2SnS_3 when the ratio is lower than 4.15 at 500°C. As shown in fig. 1, $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$ does not coexist with Cu_2SnS_3 in any compositional region. $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$ is stable up to $641 \pm 3^\circ\text{C}$. With the melting of CuS at 507°C (Kullerud, 1965), the equilibrium assemblages of $\text{Cu}_{2-x}\text{S} + \text{Cu}_{10}\text{Sn}_2\text{S}_{13} + \text{CuS}$, $\text{CuS} + \text{Cu}_{10}\text{Sn}_2\text{S}_{13} + \text{S(L)}$, and $\text{Cu}_{10}\text{Sn}_2\text{S}_{13} + \text{Cu}_5\text{Sn}_2\text{S}_7 + \text{S(L)}$ at 500°C are replaced by the assemblage $\text{Cu}_{2-x}\text{S} + \text{Cu}_5\text{Sn}_2\text{S}_7 + \text{S(L)}$ at 650°C, the upper temperature limit reached in this study.

Attempts to index the X-ray powder diffraction pattern of $\text{Cu}_5\text{Sn}_2\text{S}_7$ were not successful. Its d -values and relative intensities are ($d \text{ \AA}$, (I/I_0)) 5.251(5), 4.911(5), 4.887(5), 4.724(5), 4.202(5), 3.292(5), 3.109(100), 3.034(5), 2.949(5), 2.882(5),

Table I. X-ray Powder Diffraction Data of Cu-Sn Sulfides.

Cu ₄ SnS ₄				Cu ₂ SnS ₃				Cu ₂ Sn ₃ S ₇				Cu ₁₀ Sn ₂ S ₁₃			
d(meas), Å	1/I ₀	d(calc), Å	(hkl)	d(meas), Å	1/I ₀	d(calc), Å	(hkl)	d(meas), Å	1/I ₀	d(calc), Å	(hkl)	d(meas), Å	1/I ₀	d(calc), Å	(hkl)
4.670	5	4.642	201	5.468	10	5.489	110	5.985	40	5.972	200	5.495	15	5.465	002
3.850	10	3.831	020	4.230	10	4.234	023	4.762	5	4.775	112̄	3.663	60	3.643	003
3.381	90	3.375	400	3.641	5	3.633	024	3.673	10	3.676	020	3.224	10	3.205	113
3.326	85	3.331	220	3.138	100	3.137	204̄			3.666	202	3.016	100	3.017	130
		3.317	311			3.121	200	3.134	100	3.141	204̄			2.908	131
3.196	40	3.193	121	2.840	10	2.847	223̄			3.130	220	2.909	15	2.895	023
3.110	70	3.111	102			2.831	124	3.004	15	3.005	004			2.749	032
2.965	10	2.954	221	2.714	50	2.718	206̄	2.600	90	2.599	404̄	2.748	60	2.732	004
2.884	60	2.883	112	2.611	5	2.617	043	2.385	20	2.387	224	2.312	25	2.314	140
2.702	5	2.704	212			2.611	043			2.383	204			2.301	124
2.657	50	2.654	321	2.439	5	2.454	044	2.326	5	2.319	313	1.999	40	1.995	043
2.605	15	2.606	302			2.419	213							1.908	340
2.495	10	2.487	501	2.289	10	2.299	143	2.124	5	2.113	602̄	1.898	40	1.901	234
		2.372	031			2.273	223							1.879	341
2.372	40	2.366	511	2.166	10	2.163	314	2.003	50	2.003	006	1.872	60	1.871	150
		2.366	511			2.161	150			1.991	600			1.871	150
2.311	15	2.307	222	2.121	10	2.122	244̄			1.841	226̄	1.833	15	1.840	243
2.159	15	2.159	610			2.117	240	1.840	100	1.838	040	1.741	25	1.736	334
2.103	10	2.098	331	2.059	10	2.066	323̄			1.832	404				
2.069	15	2.063	502			2.055	324	1.760	20	1.756	240	1.666	40	1.675	126
2.048	10	2.045	411	1.967	10	1.969	242			1.751	620			1.664	153
						2.966	326̄	1.590	15	1.578	802̄	1.592	20	1.603	226
				1.920	100	1.922	060	1.570	10	1.570	406̄			1.590	060
										1.565	440			1.567	335
				1.890	10	1.898	302					1.574	20	1.573	061
						1.895	321							1.567	335
				1.835	5	1.837	336̄					1.549	10	1.544	154
2.038	15	2.036	430	1.696	10	1.698	304								
		2.030	113			1.693	346̄								
1.996	25	1.992	512												
1.967	5	1.974	132			1.643	414								
		1.965	213	1.638	80	1.636	260								
1.948	90	1.940	620			1.636	2012̄								
1.918	20	1.915	040	1.570	20	1.575	426̄								
		1.913	232			1.560	400								
1.873	20	1.868	313												
		1.863	023												
1.821	5	1.824	332												
1.802	100	1.802	403												
		1.795	223												
1.726	5	1.721	323												
1.695	25	1.699	341												
		1.687	800												
1.679	5	1.673	503												
1.664	30	1.666	440												
		1.663	721												
1.656	15	1.659	622												
1.636	10	1.632	632												
		1.631	142												
1.621	35	1.614	712												
1.610	10	1.605	532												
		1.599	004												
1.549	10	1.559	204												
		1.447	603												
		1.543	342												

2.694(5), 2.484(5), 2.452(5), 2.401(5), 2.328(5), 2.274(5), 2.241(5), 2.179(5), 2.043(5), 1.997(5), 1.906(90), 1.858(5), 1.685(5), 1.627(7), and 1.559(15). Judged from major reflection lines (3.109, 2.694, 1.906, 1.627, and 1.559), this phase may also have a superstructure derived from sphalerite.

Among the copper-tin sulphides, Cu₂SnS₃ and Cu₃SnS₄ have been identified in natural assemblages. Mohite (Cu₂SnS₃) (Kovalenker *et al.*, 1982) has cell dimensions of $a = 6.64$, $b = 11.51$, $c =$

19.93 Å, $\alpha = 90^\circ$, $\beta = 109.75^\circ$, and $\gamma = 90^\circ$, which are in good agreement with results of synthetic Cu₂SnS₃ obtained in this study. Kuramite (Kovalenker *et al.*, 1979) has a reported composition ranging from Cu_{2.6}(Fe,Zn)_{0.4}SnS₄ to Cu_{2.8}(Fe,Zn)_{0.2}SnS₄. Its (Fe,Zn)-free equivalent in the system Cu-Sn-S was produced by sulphidization of Cu₄SnS₄ at 90 °C and has a Cu/Sn ratio of 3.278 (Moh, 1982). According to Moh, this phase breaks down at temperatures above 330 °C.

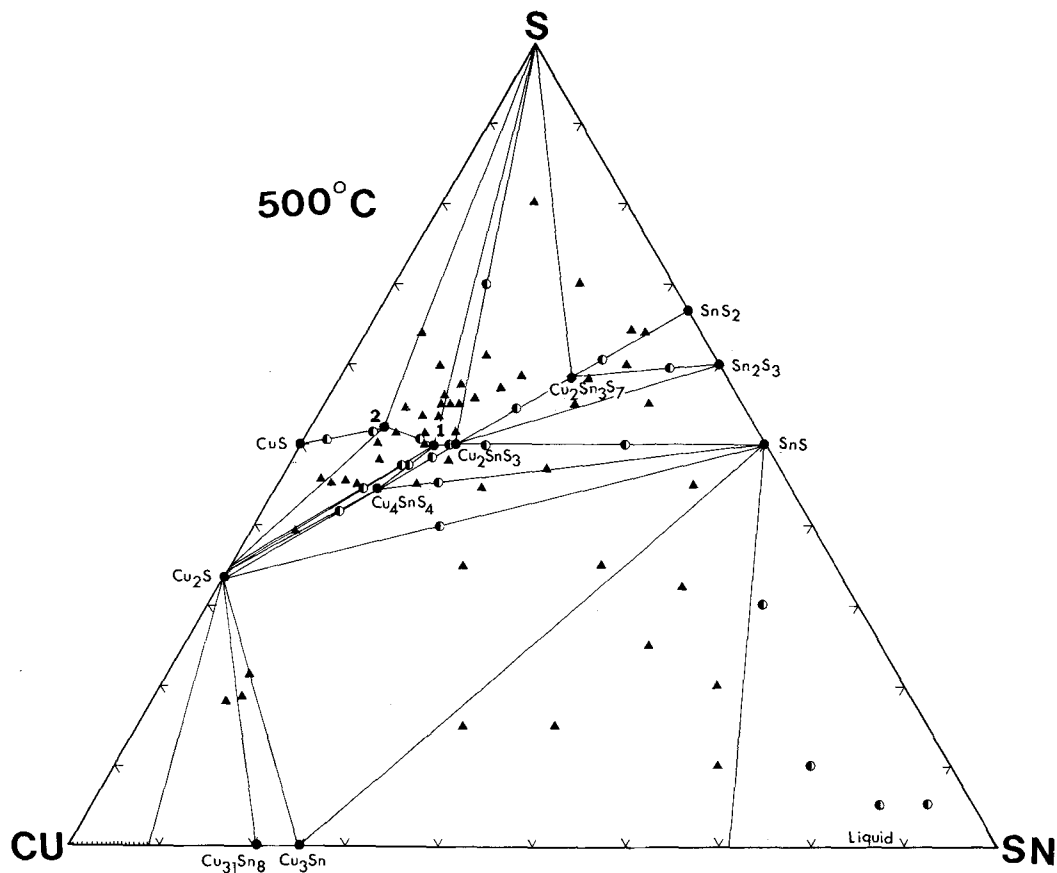


FIG. 1. Phase relations in the system Cu-Sn-S at 500°C. In the figure, solid circles, half-filled circles, and solid triangles represent one-, two-, and three-phase assemblages, respectively. '1' and '2' denote $\text{Cu}_5\text{Sn}_2\text{S}_7$ and $\text{Cu}_{10}\text{Sn}_2\text{S}_{13}$.

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