

## Crystal structure of cymrite

V. A. Drits, A. A. Kashaev, and G. V. Sokolova

Geological Institute, Academy of Sciences of the USSR  
(Submitted September 26, 1973; resubmitted April 4, 1974)  
Kristallografiya, 20, 280-286 (March-April 1975)

A space group of  $P2_1$  and  $a = 5.33$ ,  $b = 36.6$ ,  $c = 7.67 \text{ \AA}$  and  $\beta = 90^\circ$  have been established for cymrite  $\text{BaAl}_2\text{Si}_2\text{O}_8 \cdot \text{H}_2\text{O}$ . The cymrite layer of the structure consists of blocks alternating through  $b/2$  which are displaced in the positive and negative directions along the  $z$  axis. In addition, successive blocks are distinguished by a "stepwise" displacement of the atoms along the  $x$  axis.

An x-ray structural study of cymrite from the Baikal region showed that it may be considered as a hydrated form of  $\alpha$ -celsian having the chemical formula<sup>1</sup>  $\text{BaSi}_2\text{Al}_2\text{O}_8 \cdot \text{H}_2\text{O}$ . In ref. 1 the structure of cymrite was established by a hexagonal approach and the presence of secondary reflections (satellites) on the x-ray patterns was not explained. After a new find of cymrite in Alaska<sup>2</sup> the symmetry of cymrite was established as orthorhombic. Analysis of the intensities of both the primary and secondary reflections enabled us to identify the distortions in the structure giving rise to the satellites on the x-ray diffraction pattern. Their presence is caused by displacements of the Ba atoms along the  $x$  axis, the magnitude and direction of which depend on the  $y$  coordinates. From the projection of  $hk0$  the convergence between the experimental and calculated structural factors was reduced to  $R = 18\%$ , while from the three-dimensional data  $R = 23.5\%$ . Attempts to improve the agreement between the structural factors were unsuccessful.

A structural study of single crystals of cymrite found recently in California was more successful. A detailed analysis of the intensities for  $hk1$ ,  $hk2$ , etc. scans showed only one plane of symmetry, perpendicular to the  $b$  axis. A quite clear distinction from Alaskan cymrite was established for  $h0l$ ,  $h1l$  scans because no planes of symmetry were detected for these sections of the reciprocal lattice. From these data it follows that cymrite has a monoclinic symmetry and that the crystals studied previously were triplets (the specimen from the Baikal region) and twins (the specimen from Alaska).

The unit-cell parameters of Californian cymrite are:  $a = 5.33$ ,  $b = 36.6$ ,  $c = 7.67 \text{ \AA}$ ,  $\beta = 90^\circ$ . The geometry of the x-ray diffraction patterns is such that each  $hk l$  reflection having  $k = 4n$  is accompanied by satellites having indices  $hk \pm 1l$  and sometimes  $hk \pm 2l$ . The satellites are absent for the  $0k0$  reflection, which indicates extinctions of the  $0k0$  reflections having  $k = 2n \pm 1$ . Under these conditions groups of  $P2_1/a$  and  $P2_1$  symmetry are possible. Since we were unable to arrange the atoms from the previously established structural model of cymrite in the first group, the structure of the mineral was refined within the scope of space group  $P2_1$ , taking into account that the extinctions of the  $h0l$  reflections having  $h = 2n \pm 1$  are caused by the pseudo-hexagonal nature of the structure.

The experimental data were corrected for the LP factor of the reflection intensities from  $0kl$ ,  $hk0$ - $hk4$  scans, obtained with a KFOR goniometer with Mo radiation from a single crystal of size 0.1-0.2 mm. Intensities were estimated by means of the "blackening spots."

The atomic coordinates determined for Alaskan cymrite were taken as the initial coordinates. An appreciable

improvement in the convergence between the calculated and experimental structural factors is obtained when Ba is displaced from the origin along the  $x$  axis. The least-squares method was used initially to refine the  $x$  and  $y$  coordinates of the atoms from the intensities of the  $hk0$  zones ( $R_{hk0} = 9.3\%$ ). The same method, but for the whole  $hk l$  block, was used to refine the coordinates of Ba and Si to  $R = 17.7\%$ . However, refinement of the coordinates of the O atoms did not lower the values of  $R_{hk l}$ . For this reason their coordinates were taken from a three-dimensional synthesis and subsequently adjusted by the method of least squares. This leads to  $R_{hk0} = 8.3\%$ ,  $R_{0kl} = 15.0\%$ , and  $R_{hk l} = 16.7\%$ . Since the value of  $R_{0kl}$  exceeded  $R_{hk0}$  substantially the  $z$  coordinates of O and the  $\text{H}_2\text{O}$  molecules were first refined (initially by the least-squares method for the  $0kl$  zone), and then the variants were counted in which the  $z$  coordinates of each O were varied by intervals of  $0.01 \text{ \AA}$  in the higher and lower directions until there was an improvement in  $R_{0kl}$ . The final coordinates of the atoms are shown in Table 1 ( $R_{0kl} = 11\%$ ,  $R_{hk0} = 8.3\%$ ,  $R_{hk l} = 13.5\%$ ) and the interatomic distances calculated from them are presented in Table 2.

The  $xy$  and  $yz$  projections of the structure of cymrite are presented in Figs. 1 and 2. The average cation-anion distance  $r = 1.675 \text{ \AA}$  over all the tetrahedra corresponds to that expected for tetrahedra having the composition  $\text{Si}_{0.5}\text{Al}_{0.5}$ , if we assume<sup>3</sup> that  $r(\text{Si}-\text{O}) = 1.61 \text{ \AA}$  and  $r(\text{Al}-\text{O}) = 1.75 \text{ \AA}$ . The average cation-anion distances for the majority of the tetrahedra are  $1.67 \pm 0.01 \text{ \AA}$ , which indicates a virtually disordered distribution of Si and Al through these tetrahedra with a slight predominance of Si atoms in them. Exceptions are the tetrahedra  $T_1$  and  $T_3$  (Figs. 1 and 2) stretched along the  $a$  chain, which are more enriched in Al and for which  $r_1 = r_3 = 1.70 \text{ \AA}$ ; consequently, their composition is approximately  $\text{Si}_{0.4}\text{Al}_{0.6}$  and the remainder  $\text{Si}_{0.6}\text{Al}_{0.4}$  ( $r = 1.67 \text{ \AA}$ ) with variations from  $\text{Si}_{0.65}\text{Al}_{0.35}$  ( $r = 1.66 \text{ \AA}$ ) to  $\text{Si}_{0.5}\text{Al}_{0.5}$  ( $r = 1.68 \text{ \AA}$ ). The average distances from the two tetrahedral cations to the common basal O are presented<sup>1)</sup> in Fig. 1.

A characteristic peculiarity of the cymrite layer is that the tetrahedra are developed about the normals to their bases, which form a clearly defined pseudoditrigonal pattern. Here the direction of rotation of the tetrahedra in the upper lattices (odd numbers) is opposite to that of the tetrahedra in the lower lattices (even numbers) (Fig. 1), while the angles of rotation of the tetrahedra in the upper lattices from zero to  $b/2$  along  $y$  ( $\alpha = 12^\circ 30'$ ) are markedly larger than those in the tetrahedra of the lower lattices ( $\alpha = 5^\circ 48'$ ). Since the structure in the second half of the unit cell is reversed by the axis  $2_1$ , each individual layer consists of blocks alternating along the  $y$  axis which differ in the shape of the six loops of the upper

TABLE 1. Atomic Coordinates in the Structure of Cymrite

Atom	x	y	z	Atom	x	y	z
Ba <sub>1</sub>	-0.0249	-0.0015	0.0083	O <sub>11</sub>	0.294	0.02	0.218
Ba <sub>2</sub>	0.5267	0.1235	0.0083	O <sub>12</sub>	0.218	0.02	-0.221
Ba <sub>3</sub>	0.0292	0.2485	0.0083	O <sub>13</sub>	0.766	0.062	0.185
Ba <sub>4</sub>	0.5267	0.3735	-0.0083	O <sub>14</sub>	0.750	0.064	-0.171
T <sub>1</sub>	0.506	0.0416	0.280	O <sub>15</sub>	-0.0416	0.125	0.225
T <sub>2</sub>	0.510	0.0416	-0.270	O <sub>16</sub>	0.036	0.125	-0.180
T <sub>3</sub>	0.0092	0.083	0.304	O <sub>17</sub>	0.311	0.178	0.230
T <sub>4</sub>	0.0204	0.083	-0.265	O <sub>18</sub>	0.770	0.186	-0.180
T <sub>5</sub>	0.0292	0.167	0.302	O <sub>19</sub>	0.811	0.196	0.230
T <sub>6</sub>	0.0283	0.167	-0.268	O <sub>20</sub>	0.270	0.189	-0.180
T <sub>7</sub>	0.529	0.208	0.302	O <sub>21</sub>	0.453	0.250	0.230
T <sub>8</sub>	0.527	0.208	-0.260	O <sub>22</sub>	0.527	0.250	-0.185
T <sub>9</sub>	0.535	0.292	0.292	O <sub>23</sub>	0.811	0.304	0.233
T <sub>10</sub>	0.528	0.292	-0.265	O <sub>24</sub>	0.269	0.311	-0.174
T <sub>11</sub>	0.0308	0.333	0.302	O <sub>25</sub>	0.324	0.324	0.240
T <sub>12</sub>	0.0308	0.333	-0.260	O <sub>26</sub>	0.770	0.314	-0.180
T <sub>13</sub>	0.0304	0.417	0.300	O <sub>27</sub>	-0.047	0.375	0.230
T <sub>14</sub>	0.0204	0.417	-0.265	O <sub>28</sub>	0.025	0.375	-0.184
T <sub>15</sub>	0.521	0.458	0.302	O <sub>29</sub>	0.307	0.431	0.226
T <sub>16</sub>	0.521	0.458	-0.267	O <sub>30</sub>	0.770	0.436	-0.180
O <sub>1</sub>	0.516	0.0416	0.506	O <sub>31</sub>	0.805	0.444	0.225
O <sub>2</sub>	0.0158	0.0835	0.522	O <sub>32</sub>	0.267	0.439	-0.179
O <sub>3</sub>	0.0375	0.167	0.514	H <sub>2</sub> O (1)	-0.025	-0.002	0.387
O <sub>4</sub>	0.525	0.210	0.516	H <sub>2</sub> O (2)	0.527	0.124	0.387
O <sub>5</sub>	0.536	0.290	0.516	H <sub>2</sub> O (3)	0.029	0.248	0.387
O <sub>6</sub>	0.0258	0.333	0.516	H <sub>2</sub> O (4)	0.527	0.374	-0.387
O <sub>7</sub>	0.0204	0.416	0.520	H <sub>2</sub> O (5)	-0.025	0.006	0.624
O <sub>8</sub>	0.510	0.458	0.516	H <sub>2</sub> O (6)	0.527	0.131	0.624
O <sub>9</sub>	0.474	0	0.194	H <sub>2</sub> O (7)	0.029	0.256	0.624
O <sub>10</sub>	0.544	-0.0006	-0.222	H <sub>2</sub> O (8)	0.527	0.382	-0.624

and lower tetrahedral lattices. The boundary between the blocks occurs at the chain of T<sub>1</sub> and T<sub>3</sub> tetrahedra (Fig. 1); the z coordinates of the basal O atoms differ greatly in each block (Fig. 2). Whereas the z coordinates of the basal O atoms in the first block (up to b/2 along the y axis) vary from 0.22 to 0.24, in the following block the corresponding coordinates are 0.17-0.19. The "break" in the blocks along the z axis occurs at the basal O atoms common to the tetrahedra T<sub>1</sub>, T<sub>3</sub> and T<sub>2</sub>, T<sub>4</sub> (Fig. 2).

Let us consider in more detail the nature of the distortions in the lattices in the first block of the layer. It is known that on passing from the hexagonal pattern of a tetrahedral lattice having a parameter b<sub>t</sub> to a ditrigonal pattern the b parameter of the lattice (where b = b<sub>t</sub> cos α) decreases with an increase in the angle α. Since in the cymrite layer being considered the b parameters for the upper and lower lattices are the same, while the α angles are different, the linear dimensions of the upper lattice in the ab plane should be greater in the absence of rotation of the tetrahedra (the angle α is increased) than the dimensions of the lower lattice. In fact, it follows from Table 2 that the average edge length of the bases for all the tetrahedrons is 2.73 Å for the upper tetrahedral lattice and 2.68 Å for the lower. Here, the edge lengths of the bases of the tetrahedra (the average for each) in the upper lattice increase regularly along the y axis from the T<sub>3</sub> tetrahedra (2.69 Å) to the T<sub>9</sub>-T<sub>11</sub> tetrahedra (2.76 Å) and decrease to the T<sub>15</sub>-T<sub>2</sub> tetrahedra (2.70 Å) (Fig. 2).

On the other hand, for the lower lattices of the block the edge length of the base is shortened in the same direction from the T<sub>4</sub> tetrahedron (2.75 Å) to the T<sub>8</sub>-T<sub>12</sub> tetrahedra (2.66 Å) and then these values increase for the T<sub>16</sub>-T<sub>1</sub> tetrahedra (2.69; 2.70 Å). Since for the majority of the tetrahedra the r values are close to each other and, consequently, also close to the average edge length over the tetrahedron,<sup>4</sup> the data presented indicate that the tetrahedra of the upper lattice are flattened along the c axis (relative to their true geometrical shape), while the tetra-

hedra of the lower lattice are greatly elongated. Furthermore, the z coordinates of the basal O atoms of the upper lattice along the y axis from O<sub>11</sub> to O<sub>10</sub> (Fig. 2) first increase from 0.225 (O<sub>11</sub>) to 0.240 (O<sub>25</sub>) (the flattening of the tetrahedra is increased) and then decrease to 0.222 (O<sub>10</sub>) (the flattening of the tetrahedra is decreased). The z coordinates of the apical and basal O atoms (Table 1) also confirm the relationships noted for the degree of stretching and flattening of the tetrahedra along the z axis for the lattices of the tetrahedra in the first block of the layer.

Thus, in each block of the cymrite layer lattices of different linear dimensions in the tetrahedra are joined and their commensurability is achieved by the different angle of rotation of the two lattices and also by a certain twisting of the upper lattice (having a large parameter b<sub>t</sub>) relative to the lower one. The stresses which may arise in such a connection are "stripped," due to the successive rotation of the blocks by the 2<sub>1</sub> axis which, moreover, ensures that the layer is plane-parallel on the whole.

As may be seen from Fig. 2, the Ba atoms in each block are displaced markedly from the average positions, the displacement in the adjacent blocks taking place in the opposite directions along the x axis. The absolute displacement of all the Ba atoms is approximately the same (Table 1). In accordance with the results in ref. 2, the appearance of satellites among the hk0 reflections on the x-ray diffraction patterns of cymrite is caused by the "stepwise" displacement of the atoms, chiefly Ba, along the x axis with a change in the direction of the displacement through each b/2 along the y axis.

Likewise the appearance of satellites along the y\* axis for the 0k/ scans is due to the stepwise displacement of the atoms in each block along the z axis. Since it is the O and Si atoms, but not Ba, which are most strongly displaced along the z axis, the satellites on the 0k/ section have a markedly lower intensity than the satellites on the hk0 section.

TABLE 2. Interatomic Distances (Å) in the Structure of Cymrite

$T_1 - O_1$	1.73	$T_5 - O_3$	1.63	$T_9 - O_5$	1.72	$T_{13} - O_7$	1.69
$T_1 - O_9$	1.68	$T_5 - O_{19}$	1.67	$T_9 - O_{23}$	1.60	$T_{13} - O_{29}$	1.66
$T_1 - O_{11}$	1.66	$T_5 - O_{17}$	1.65	$T_9 - O_{25}$	1.68	$T_{13} - O_{31}$	1.66
$T_1 - O_{13}$	1.74	$T_5 - O_{15}$	1.70	$T_9 - O_{21}$	1.68	$T_{13} - O_{27}$	1.69
Cp.	1.70	Cp.	1.66	Cp.	1.67	Cp.	1.67
Av.(calc.)	1.73	Av.(calc.)	1.67	Av.(calc.)	1.67	Av.(calc.)	1.67
$T_2 - O_1$	1.72	$T_6 - O_3$	1.67	$T_{10} - O_5$	1.68	$T_{14} - O_7$	1.65
$T_2 - O_{10}$	1.61	$T_6 - O_{18}$	1.69	$T_{10} - O_{28}$	1.66	$T_{14} - O_{32}$	1.68
$T_2 - O_{12}$	1.63	$T_6 - O_{20}$	1.66	$T_{10} - O_{24}$	1.70	$T_{14} - O_{30}$	1.64
$T_2 - O_{14}$	1.70	$T_6 - O_{16}$	1.69	$T_{10} - O_{22}$	1.67	$T_{14} - O_{28}$	1.67
Cp.	1.66	Cp.	1.68	Cp.	1.68	Cp.	1.66
Av.(calc.)	1.67	Av.(calc.)	1.68	Av.(calc.)	1.67	Av.(calc.)	1.67
$T_3 - O_2$	1.67	$T_7 - O_4$	1.64	$T_{11} - O_6$	1.64	$T_{15} - O_8$	1.64
$T_3 - O_{11}$	1.71	$T_7 - O_{19}$	1.66	$T_{11} - O_{23}$	1.67	$T_{15} - O_{29}$	1.62
$T_3 - O_{13}$	1.77	$T_7 - O_{17}$	1.70	$T_{11} - O_{25}$	1.67	$T_{15} - O_{31}$	1.70
$T_3 - O_{15}$	1.67	$T_7 - O_{21}$	1.69	$T_{11} - O_{27}$	1.69	$T_{15} - O_{10}$	1.68
Cp.	1.70	Cp.	1.67	Cp.	1.67	Cp.	1.66
Av.(calc.)	1.71	Av.(calc.)	1.67	Av.(calc.)	1.67	Av.(calc.)	1.68
$T_4 - O_2$	1.63	$T_8 - O_4$	1.72	$T_{12} - O_6$	1.72	$T_{16} - O_8$	1.66
$T_4 - O_{12}$	1.65	$T_8 - O_{18}$	1.65	$T_{12} - O_{28}$	1.67	$T_{16} - O_{32}$	1.66
$T_4 - O_{14}$	1.76	$T_8 - O_{20}$	1.66	$T_{12} - O_{24}$	1.64	$T_{16} - O_{30}$	1.70
$T_4 - O_{16}$	1.67	$T_8 - O_{22}$	1.65	$T_{12} - O_{22}$	1.65	$T_{16} - O_9$	1.65
Cp.	1.68	Cp.	1.67	Cp.	1.67	Cp.	1.67
Av.(calc.)	1.64	Av.(calc.)	1.67	Av.(calc.)	1.67	Av.(calc.)	1.68
$T_1$ tetrahedron		$T_4$ tetrahedron		$T_7$ tetrahedron		$T_{10}$ tetrahedron	
$O_1 - O_9$	2.85	$O_2 - O_{12}$	2.60	$O_4 - O_{17}$	2.74	$O_5 - O_{22}$	2.73
$O_1 - O_{13}$	2.90	$O_2 - O_{14}$	2.88	$O_4 - O_{19}$	2.72	$O_5 - O_{24}$	2.88
$O_1 - O_{11}$	2.74	$O_2 - O_{16}$	2.80	$O_4 - O_{21}$	2.67	$O_5 - O_{26}$	2.79
$O_9 - O_{13}$	2.77	$O_{12} - O_{16}$	2.95	$O_{19} - O_{17}$	2.75	$O_{22} - O_{24}$	2.64
$O_9 - O_{11}$	2.83	$O_{12} - O_{14}$	2.58	$O_{19} - O_{21}$	2.76	$O_{22} - O_{26}$	2.69
$O_{11} - O_{13}$	2.55	$O_{14} - O_{16}$	2.72	$O_{17} - O_{21}$	2.76	$O_{24} - O_{26}$	2.67
Cp.	2.78	Cp.	2.76	Cp.	2.73	Cp.	2.73
$T_2$ tetrahedron		$T_5$ tetrahedron		$T_8$ tetrahedron		$T_{11}$ tetrahedron	
$O_1 - O_{10}$	2.61	$O_3 - O_{15}$	2.74	$O_4 - O_{20}$	2.81	$O_6 - O_{23}$	2.68
$O_1 - O_{12}$	2.65	$O_3 - O_{17}$	2.65	$O_4 - O_{18}$	2.81	$O_6 - O_{25}$	2.67
$O_1 - O_{14}$	2.89	$O_3 - O_{19}$	2.71	$O_4 - O_{22}$	2.73	$O_6 - O_{27}$	2.71
$O_{10} - O_{12}$	2.55	$O_{15} - O_{17}$	2.71	$O_{18} - O_{20}$	2.67	$O_{25} - O_{23}$	2.83
$O_{10} - O_{14}$	2.63	$O_{15} - O_{19}$	2.73	$O_{20} - O_{22}$	2.63	$O_{25} - O_{27}$	2.73
$O_{12} - O_{14}$	2.91	$O_{19} - O_{17}$	2.75	$O_{22} - O_{18}$	2.69	$O_{25} - O_{27}$	2.73
Cp.	2.71	Cp.	2.71	Cp.	2.72	Cp.	2.72
$T_3$ tetrahedron		$T_6$ tetrahedron		$T_9$ tetrahedron		$T_{12}$ tetrahedron	
$O_2 - O_{13}$	3.01	$O_3 - O_{16}$	2.81	$O_5 - O_{21}$	2.68	$O_6 - O_{24}$	2.83
$O_2 - O_{11}$	2.79	$O_3 - O_{20}$	2.77	$O_5 - O_{23}$	2.66	$O_6 - O_{26}$	2.79
$O_2 - O_{15}$	2.76	$O_3 - O_{18}$	2.83	$O_5 - O_{25}$	2.71	$O_6 - O_{28}$	2.77
$O_{11} - O_{13}$	2.85	$O_{16} - O_{20}$	2.67	$O_{23} - O_{21}$	2.76	$O_{26} - O_{24}$	2.66
$O_{13} - O_{15}$	2.56	$O_{18} - O_{18}$	2.66	$O_{23} - O_{25}$	2.70	$O_{26} - O_{28}$	2.63
$O_{11} - O_{15}$	2.65	$O_{18} - O_{20}$	2.67	$O_{21} - O_{25}$	2.81	$O_{24} - O_{28}$	2.70
Cp.	2.77	Cp.	2.74	Cp.	2.72	Cp.	2.73
$T_{13}$ tetrahedron		$T_{14}$ tetrahedron		$T_{15}$ tetrahedron		$T_{16}$ tetrahedron	
$O_7 - O_{27}$	2.71	$O_7 - O_{28}$	2.73	$O_8 - O_{29}$	2.67	$O_8 - O_{30}$	2.83
$O_7 - O_{29}$	2.78	$O_7 - O_{30}$	2.76	$O_8 - O_{31}$	2.78	$O_8 - O_{32}$	2.76
$O_7 - O_{31}$	2.74	$O_7 - O_{32}$	2.79	$O_8 - O_{10}$	2.74	$O_8 - O_9$	2.71
$O_{29} - O_{27}$	2.80	$O_{28} - O_{30}$	2.69	$O_{31} - O_{29}$	2.70	$O_{32} - O_{30}$	2.68
$O_{29} - O_{31}$	2.72	$O_{30} - O_{32}$	2.65	$O_{31} - O_{10}$	2.78	$O_{32} - O_9$	2.64
$O_{31} - O_{27}$	2.66	$O_{28} - O_{32}$	2.63	$O_{29} - O_{10}$	2.64	$O_9 - O_{30}$	2.70
Cp.	2.73	Cp.	2.70	Cp.	2.72	Cp.	2.73
$Ba_1 - O_9$	3.03	$Ba_2 - O_{11}$	2.78	$Ba_3 - O_{19}$	2.83	$Ba_4 - O_{32}$	3.08
$Ba_1 - O_9'$	3.02	$Ba_2 - O_{12}$	3.63	$Ba_3 - O_{26}$	3.13	$Ba_4 - O_{29}$	3.02
$Ba_1 - O_{10}$	2.90	$Ba_2 - O_{13}$	2.88	$Ba_3 - O_{18}$	3.05	$Ba_4 - O_{30}$	2.95
$Ba_1 - O_{10}'$	3.51	$Ba_2 - O_{14}$	2.85	$Ba_3 - O_{23}$	2.92	$Ba_4 - O_{31}$	3.49
$Ba_1 - O_{11}$	3.58	$Ba_2 - O_{19}$	3.51	$Ba_3 - O_{17}$	3.45	$Ba_4 - O_{35}$	2.87
$Ba_1 - O_{12}$	2.90	$Ba_2 - O_{18}$	3.01	$Ba_3 - O_{24}$	2.98	$Ba_4 - O_{33}$	3.60
$Ba_1 - O_{13}$	2.93	$Ba_2 - O_{17}$	2.87	$Ba_3 - O_{20}$	2.92	$Ba_4 - O_{24}$	2.97
$Ba_1 - O_{14}$	3.03	$Ba_2 - O_{20}$	3.13	$Ba_3 - O_{25}$	3.66	$Ba_4 - O_{25}$	2.85
$Ba_1 - O_{32}$	2.86	$Ba_2 - O_{15}$	2.84	$Ba_3 - O_{21}$	2.83	$Ba_4 - O_{23}$	3.00
$Ba_1 - O_{29}$	3.42	$Ba_2 - O_{15}'$	3.46	$Ba_3 - O_{21}'$	3.51	$Ba_4 - O_{28}'$	2.98
$Ba_1 - O_{30}$	2.98	$Ba_2 - O_{16}$	2.99	$Ba_3 - O_{22}$	3.04	$Ba_4 - O_{27}$	2.92
$Ba_1 - O_{31}$	2.94	$Ba_2 - O_{16}'$	3.08	$Ba_3 - O_{22}'$	3.06	$Ba_4 - O_{27}'$	3.56

Let us consider the probable reasons for the distortions observed in the structure of cymrite. Basically, it would appear, the displacements of Ba are associated with a certain ordering in the distribution of Si and Al. As has already been mentioned, the  $T_1$  and  $T_3$  tetrahedra are distinguished by a maximum Al content and, consequently, their common basal O atoms are the most undersaturated by positive valences.<sup>2)</sup> This applies particularly to the  $O_{13}$  atom having the greatest increased distance from the nuclei of  $T_1$  and  $T_3$  (1.75 Å). The  $O_{14}$  atom would also appear to be highly undersaturated because the average distance from  $O_{14}$  to the nuclei of  $T_2$  and  $T_4$  is equal to 1.73 Å. For these reasons the  $Ba_2$  atom is displaced along the x axis, converging from  $O_{13}$  and  $O_{14}$  (Fig. 1). Furthermore, such a displacement is favored by the relatively high valency unsaturation of the  $O_{15}$  anion, from which the dis-

tance to the nuclei of  $T_3$  and  $T_5$  is equal to 1.69 Å. The  $Ba_1$  atom also tends to converge from  $O_{13}$  and  $O_{14}$  and is displaced along the x axis in the direction opposite to the displacement of  $Ba_2$  (Fig. 1). The displacement of  $Ba_1$  and  $Ba_2$  along the y axis scarcely occurs at all here. It would seem that if the  $Ba_1$  and  $Ba_2$  atoms move in opposition to each other along a straight line joining their centers this would also give rise to a stronger electrostatic repulsion between them. The relatively high unsaturation of  $O_{21}$  and  $O_{27}$  (Fig. 1) favors the displacement of  $Ba_3$  and  $Ba_4$  in the same direction as  $Ba_2$  along the x axis. Furthermore, with a unidirectional displacement of the Ba atoms they become uniformly distributed in the *ab* plane in each block through the hexagonal arrangement.

The displacement of the  $Ba_2$  atom to  $O_{15}$  leads to its



These data are sufficient to calculate the coordinates of the Ba and O atoms and to obtain the stepwise displacement of these atoms along the x axis automatically, as well as the ordering in the distribution of Si and Al. Table 2 shows the calculated average cation-anion distances for all the tetrahedra, which agree well with the experimental data.

We are sincerely grateful to Academician N. V. Belov for discussion of the investigation and for comments, and to Dr. E. J. Essene for kindly supplying specimens of California cymrite.

---

<sup>1</sup>The basal O atoms form the bases of the tetrahedra in the plane of the

cymrite layer.

<sup>2</sup>The average distance from a given basal O to the two nearest nuclei of tetrahedral cations can be taken conditionally as a measure of the relative valency unsaturation of this O atom by tetrahedral cations: To a first approximation, the greater this distance, the greater is the unsaturation of the anions.

---

<sup>1</sup>A. A. Kashaev, Dokl. Akad. Nauk SSSR, 169, 201 (1966).

<sup>2</sup>V. A. Drita and A. A. Kashaev, Kristallografiya, 13, 809 (1968) [Sov. Phys. - Crystallogr., 13, 700 (1969)].

<sup>3</sup>J. V. Smith and S. W. Bailey, Acta Crystallogr., 16, 801 (1963).

<sup>4</sup>V. A. Drita, Kristallografiya, 15, 913 (1970) [Sov. Phys. - Crystallogr., 13, 795 (1971)].