#### THE AMERICAN MINERALOGIST, VOL. 56, NOVEMBER-DECEMBER, 1971

# THE CRYSTAL STRUCTURE OF THE LOW-TEMPERATURE FORM OF SILVER SELENIDE

# G. A. WIEGERS, Laboratorium voor Anorganische Chemie, Materials Science Center, Rijksuniversiteit, Bloemsingel 10, Groningen, The Netherlands.

### ABSTRACT

The crystal structure of the low-temperature form of Ag<sub>2</sub>Se (Ag<sub>2</sub>Se II, naumannite, stable below 133°C) has been determined by X-ray powder methods. The structure differs from the one found by Pinsker *et al.* by electron diffraction studies of thin sublimed layers.

Ag<sub>2</sub>Se is orthorhombic with unit-cell dimensions a 4.333, b 7.062, c 7.764 Å. The space group is P 2<sub>1</sub>2<sub>1</sub>2<sub>1</sub> and there are four units Ag<sub>2</sub>Se in the unit cell. All atoms lie on the fourfold general positions. The structure is closely related to the structure of the monoclinic modification of silver sulfide (Ag<sub>2</sub>S III, acanthite). There is an almost planar distribution of selenium atoms perpendicular to the b axis. There are two types of structurally different silver atoms. Silver atoms of type I lie close to the planes of selenium atoms and are coordinated almost tetrahedrally by selenium at distances of 2.62, 2.76, 2.79, and 2.86 Å. Silver atoms of type II lie halfway between the planes of selenium; they are coordinated triangularly by selenium at distances of 2.72, 2.76, and 2.81 Å.

The anion packing is essentially body-centered cubic.

### INTRODUCTION

Silver selenide, Ag<sub>2</sub>Se, occurs at atmospheric pressure in two polymorphic forms. The high-temperature form, form I, is stable above 133°C. The structure is body-centered cubic with spacing a=4.983 Å; the unit cell contains two units Ag<sub>2</sub>Se. The selenium atoms form a bodycentered packing, while the silver atoms are statistically distributed over several types of interstitial sites (Rahlfs, 1936).

The low-temperature form, Ag<sub>2</sub>Se II, stable below  $133^{\circ}$ C, is known as the mineral naumannite. Ag<sub>2</sub>Se is orthorhombic with cell constants a 4.344, b 7.111, and c 7.790 Å (Conn and Taylor, 1960). There are four units Ag<sub>2</sub>Se in the unit cell. The structure of thin sublimed films has been determined by Pinsker *et al.* (1965) by electron diffraction investigation. The cell constants, a 4.325, b 7.05, and c 7.82 Å agree with those reported by Conn and Taylor. The space group is given as  $P 2_122$ . The silver atoms form a framework of a diamond-type lattice; the Ag-Ag distances are short, ranging from 2.61 to 2.82 Å (in metallic silver 2.89 Å). This structure is penetrated by Ag-Se-Ag-Se chains with Ag-Se distances of 2.54 Å.

The intensities of the X-ray powder lines of Ag<sub>2</sub>Se II observed by Early (1950), Novoseleva *et al.* (1967) and by us are, however, in disagreement with the intensities calculated from the atomic parameters given by Pinsker *et al.* 

STRUCTURE OF LOW SILVER SELENIDE

By comparing the cell constants (Table 1) and intensities of the powder lines of Ag<sub>2</sub>S III (U.S. Nat. Bur. Stan., 1960) and Ag<sub>2</sub>Se II a resemblance is noted, which suggests a close relationship of both structures. The structure of Ag<sub>2</sub>S III has been determined by Frueh (1958) by singlecrystal methods. The space group is  $P 2_1/n$  and there are four units Ag<sub>2</sub>S in the unit cell. All atoms are located on fourfold general positions (Table 1). There is an almost planar distribution of sulfur atoms perpendicular to the *b* axis. There are two types of silver atoms; silver atoms of type I lie slightly above or below the planes of sulfur and are triangularly coordinated to sulfur atoms at distances of 2.50, 2.61, and 2.69 Å.

Silver atoms of type II lie halfway between the planes of sulfur and link them together by having one close sulfur atom in the plane above and one in the plane below; the Ag-S distances are 2.49 and 2.52 Å respectively. The sulfur atoms are arranged in a slightly distorted bodycentered cubic array with one of the twofold axes of the cube parallel to [010], the  $2_1$  axis of the monoclinic space group. The faces of the cube lie in the (103), (121), and (121) planes.

		1122204							
			Ag2S	III			Ag <sub>2</sub> Se II		
			a =	4.229 Å			a = 4.33	3 Å	
	cell constants		b = 6.931 Å				b = 7.06	2 Å	
			C =	7.862 Å			c = 7.76	4 Å	
			β =	99.61					
	space group		P 2	/n			P 212121		
coordinates of			x,y,z; <del>x,y,z</del>				x,y,z; 1/2+x,y,z		
	equivalent positions		1/2+x,1/2-y,1/2+z; 1/2-x,1/2+y,1/2-z				x,1/2-y,1/2+z; 1/2-x,1/2+y,1/2-z.		
	coordinates	Ag I Ag II S or Se	× •285 •758 359	y .320 .015 239	z •435 •305	x .107(4) .728(4) .358(4)	y •369(2) •029(2) •235(3)	z .456(2) .361(2) .149(4)	
		5 51 56			8-24				

TABLE 1. CRYSTAL DATA OF Ag<sub>2</sub>S III AND Ag<sub>2</sub>Se II

The cell constants of Ag<sub>2</sub>S III are the more accurate ones determined by the N.B.S. (1960) The origin in the cell of Ag<sub>2</sub>Se II is shifted over  $\frac{1}{44}0$  compared with the origin in the conventional setting (International Tables for Crystallography, I; see text). Standard deviations, in units of the last decimal place, are given in parentheses.

#### G. A. WIEGERS

The structure of Ag<sub>2</sub>Se II could easily be solved by X-ray powder methods taking into account the relationship of the diffraction patterns of Ag<sub>2</sub>Se II and Ag<sub>2</sub>S III. No attempts were made to investigate single crystals. In the case of Ag<sub>2</sub>S a single crystal of the body-centered cubic form (form II, stable between 176 and 600°C) on cooling rapidly converts to a polycrystalline species of the monoclinic form while retaining the cubic morphology (Frueh, 1958; Taylor, 1969). The same conversion probably occurs in Ag<sub>2</sub>Se.

#### EXPERIMENTAL

Ag<sub>2</sub>Se was prepared from the elements in the stoichiometric proportion. The mixture of the elements was heated at 500°C for four days in an evacuated quartz tube, followed by slow cooling (10°C per hour) to room temperature.

X-ray powder patterns were registered by a Philips diffractometer and a Guinier-de Wolff camera (Nonius); CuK $\alpha$  radiation was used in both cases. The powder pattern essentially agreed with that of the mineral naumannite (Early, 1950; good agreement was found with the powder pattern of synthetic Ag<sub>2</sub>Se published by Novoseleva *et al.* (1967).

Accurate *d*-values were obtained from a Guinier photograph; silicon powder (a = 5.4305 Å) was used as an internal standard. The cell constants obtained by least-squares are a = 4.333, b = 7.062, and c = 7.764 Å; the standard deviation is 0.05 percent.

The intensities of the powder lines were measured by planimetering of the diffractograms.

Atomic positions were refined by means of the least-squares program T53C written by Dr. H. M. Rietveld (R.C.N., Petten, The Netherlands). This full-matrix program permits to include overlapping powder lines in the calculations.

### DETERMINATION OF THE STRUCTURE

The systematic extinctions: h00, 0k0, and 00l reflections only present for h, k, l=2n, are in agreement with space group  $P 2_12_12_1$ . In the monoclinic cell of Ag<sub>2</sub>S III, space group  $P 2_1/n$ , the atoms occupy the fourfold general positions:  $x,y,z; \bar{x},\bar{y},\bar{z}; \frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}+z; \frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}-z$ . The coordinates of the three independent atoms are given in Table 1. The fourfold general positions in space group  $P 2_12_12_1$  are  $x,y,z; \frac{1}{2}+x,\bar{y},\bar{z};$  $\bar{x},\frac{1}{2}-y,\frac{1}{2}+z; \frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}-z$  if the origin is shifted over  $\frac{1}{44}0$  compared with the origin in the conventional setting halfway between the three non-intersecting twofold screw axes (International Tables for Crystallography, I), in order to demonstrate the relationship with the fourfold positions in space group  $P 2_1/n$ . The (100) projections in both space groups have the same symmetry.

The intensities of the 0kl reflections of Ag<sub>2</sub>Se II were calculated using the coordinates of the corresponding atoms in Ag<sub>2</sub>S III. The agreement between calculated and observed intensities was reasonable and a refinement was carried out with 15 0kl reflections. The index  $R_{\rm I}$ =  $100 \sum [I_0 - I_c]/I_0$  was 14 percent. Starting values for the *x* coordinates

## STRUCTURE OF LOW SILVER SELENIDE

were found by trial and error. A least-squares refinement was carried out with 64 *hkl* reflections including 13 non-observed reflections, covering all possible reflections down to d=1.41 Å. The atomic scattering factors were those given by Moore (1963). All atoms were assumed to have the same (isotropic) temperature factor. Nine positional parameters, one scaling factor, and one temperature factor were also refined. The final agreement is good, the index  $R_{\rm I}$  being 8.3 percent. The coefficient B in the temperature factor was 0.75 Å<sup>2</sup>. The coordinates and the standard deviations are given in Table 1. Observed and calculated intensities are compared in Table 2.

# DISCUSSION OF THE STRUCTURE

The distances between the atoms were calculated using the coordinates given in Table 1. The standard deviations in the distances are about 0.03 Å. The (100) projections of  $Ag_2S$  III and  $Ag_2Se$  II are shown in Figure 1.

In both structures there is an almost planar arrangement of chalcogen atoms perpendicular to the *b* axis. There are two types of structurally different silver atoms. Silver atoms of type I lie a little above or below (in Ag<sub>2</sub>S 0.49 Å, in Ag<sub>2</sub>Se 0.84 Å) the planes of the chalcogen atoms. In Ag<sub>2</sub>S silver atoms of type I have a distorted tetrahedral coordination by sulfur. Three sulfur atoms are lying in the nearest plane, the Ag-S distances being 2.50, 2.61, 2.69 Å; the fourth sulfur atom lies in the next plane, the Ag-S distance is 3.00 Å. In Ag<sub>2</sub>Se the silver atoms of type I are tetrahedrally coordinated, the Ag-Se distances are 2.62, 2.79, 2.86 Å (to selenium atoms lying in one plane), and 2.71 Å (to selenium in the next plane).

The silver atoms of type II lie about halfway between the planes of chalcogen atoms. In Ag<sub>2</sub>S these silver atoms have five sulfur neighbours at distances of 2.49, 2.52, 3.07, 3.33, and 3.44 Å. The coordination by sulfur at the short distances of 2.49 and 2.52 Å is almost linear. In Ag<sub>2</sub>Se the Ag-Se distances of silver of type II are 2.72, 2.74, 2.81, 3.28, and 3.50 Å. The coordination of silver by selenium at the distances of 2.72, 2.74, and 2.81 Å is almost triangular.

It may be noted that  $Ag_2Te$  III also has two structurally different types of silver. Silver atoms of type I are surrounded by four tellurium atoms at distances of 2.87, 2.91, 3.04, and 2.99 Å forming a somewhat distorted tetrahedron. The other silver atoms have each five close tellurium atoms at distances of 2.85, 2.90, 2.95, 3.01, and 3.04 Å (Frueh, 1959).

The complicated stereochemistry of silver in chalcogenides has been ascribed to *d-s* mixing (Orgel, 1958). It may be seen from Table 3 that

1885

чон	
но	Läntna to 20 20 20 50 50 50 50 50 50 50 50 50 50 50 50 50
dcalc	
Lभूत	- 0 - 0 0 - 0 0 - 0 0 - 0 0 0 - 0 0 0 - 0 0 0 0 - 0 0 0 0 - 0 0 0 0 - 0
о I	<sup>5</sup> <sup>2</sup> <sup>3</sup> 4 <sup>2</sup> <sup>k</sup> <sup>2</sup> <sup>2</sup> <sup>k</sup> <sup>2</sup> <sup>k</sup> <sup>2</sup> <sup>k</sup> <sup>2</sup>
I c	40000000000000000000000000000000000000
dcalc	2.013 2.001 1.9999 1.9999 1.881 1.8872 1.8872 1.771 1.771 1.771 1.771 1.771 1.771 1.722 1.766 1.771 1.771 1.722 1.766 1.771 1.766 1.771 1.766
ЧKI	032 211 221 221 222 221 222 221 222 222 2
цо	210 210 2509 25
L	v@84448 v C 600 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0
dcalc	5.224 5.224 5.224 5.235
hkl	011 101 101 110 020 021 112 022 023 100 110 013 023 122 023 122 023 2200 130 023 2200 130 023

Table 2. Observed and calculated intensities of  ${\rm Ag}_{2}{\rm Se}$  II.

1886

G. A. WIEGERS

## STRUCTURE OF LOW SILVER SELENIDE



FIG. 1. (100) projections of (I) Ag<sub>2</sub>S III and (II) Ag<sub>2</sub>Se II. Sulfur and selenium atoms are shown as large open circles; silver atoms of type I as black circles and silver atoms of type II as hatched circles. The distances are in Å; x coordinates of the atoms are also given.

the number of coordinating atoms in compounds  $Ag_2X$  increases in the series X = O, S, Se, Te as may be expected from the size of the anions.

In both compounds Ag<sub>2</sub>S III and Ag<sub>2</sub>Se II each chalcogen atom is surrounded by nine silver atoms, six silver atoms forming a trigonal prism, the remaining three silver atoms are approximately in one plane with chalcogen.

There are no short Ag-Ag distances in the two compounds; the dis-

TABLE 3. COORDINATION OF SILVER BY CHALCOGEN IN COMPOUNDS Ag<sub>2</sub>X

	the same state of the same state of the same state of the same	and the second
	Ag (I)-X	Ag(II)-X
Ag20	2.044, 2.044 Å linear	-
Ag <sub>2</sub> S III	° 2.50, 2.61, 2.69 A (3.00 A) pyramidal (tetrahedral)	2.49, 2.52 Å (3.07, 3.33, 3.44 Å) linear (square pyramidal)
Ag <sub>2</sub> Se II	2.62, 2.71, 2.79, 2.86 Å tetrahedral	2.72, 2.74, 2.81 Å (3.28, 3.50 Å) triangular (square pyramidal)
Ag <sub>2</sub> Te III	2.85, 2.90, 2.95, 3.01, 3.04 Å trigonal bipyramidal	2.87, 2.91, 2.99, 3.04 Å tetrahedral

References: Ag<sub>2</sub>O: Wyckoff, (1963); Ag<sub>2</sub>S III: Frueh, 1958; Ag<sub>2</sub>Se II: this work; Ag<sub>2</sub>Te: Frueh, 1959.

The coordination is only approximate (except for Ag<sub>2</sub>O). In parenthesis are given chalcogen atoms at somewhat larger distances and the coordination if these chalcogen atoms are also taken into account.

tances range from 3.04 to 3.71 Å in  $Ag_2S$  III and from 2.93 to 3.68 Å in  $Ag_2Se$  II.

The anion packing in both compounds is essentially body-centered cubic, one of the twofold axes of the cube being parallel to [010]. In Ag<sub>2</sub>S III the cube is only slightly distorted, in Ag<sub>2</sub>Se II the distortion is larger. The (fourteen) Se-Se distances range from 3.98 to 5.14 Å. In the high-temperature body-centered cubic form I the Se-Se distances are 4.30 Å (8x) and 4.98 Å (6x).

The structure of  $Ag_2Se$  found by Pinsker *et al.* (1965) by electron diffraction is completely different from the structure described here; it might correspond to a form which is only stable in thin layers.

#### Acknowledgement

I wish to thank Prof. Jellinek for his suggestions for improving the manuscript.

### References

- CONN, J. B., AND R. C. TAYLOR (1960) Thermoelectric and crystallographic properties of Ag<sub>2</sub>Se. J. Electrochem. Soc. 107, 977–982.
- EARLY, J. W. (1950) Tables for the identification of ore mineral by X-ray powder patterns. Amer. Mineral. 35, 337-364.
- FRUEH, A. J. (1958) The crystallography of silver sulfide, Ag<sub>2</sub>S. Z. Kristallogr. 110, 136– 144.

----- (1959) The structure of hessite, Ag<sub>2</sub>Te III. Z. Kristallogr. 112, 44-52.

- MOORE, F. H. (1963) Analytic constants for atomic scattering factors. Acta Crystallogr. 16, 1169–1176.
- NATIONAL BUREAU OF STANDARDS (1960) Standard X-ray diffraction powder patterns. U.S. Nat. Bur. Stand. Circ. 539, 10, 51-52.
- NOVOSELEVA, A. V., ZH. G. SHLEIFMAN, V. P. ZLOMANOV, AND R. K. SLOMA (1967) Study of the silver selenide-lead selenide system. *Izv. Akad. Nauk SSSR Neorg. Mater.* 3, 1143–1146 [Transl. *Inorg. Mater.* 3, 1010–1013].
- ORGEL, L. E. (1958a) Stereochemistry of metals of the B sub-groups. Part I. Ions with filled d-electron shells. J. Chem. Soc., 4186-4190.

----- (1958b) Lattice distortions due to d-s mixing. J. Phys. Chem. Solids 7, 286-287.

- PINSKER, Z. G., CHOU CHING LIANG, R. M. IMANOW, AND E. L. LAPIDUS (1965) Determination of the crystal structure of the low temperature phase αAg<sub>2</sub>Se. Kristallografiya 10, 275–283 [Transl. Sov. Phys. Crystallogr. 10, 225–231 (1965)].
- RAHLFS, P. (1936) Über die kubischen Hochtemperaturmodifikationen der Sulfide, Selenide und Telluride des Silbers und des einwertigen Kupfers. Z. Phys. Chem. B 31, 157-194.

TAYLOR, L. A (1969) The significance of twinning in Ag<sub>2</sub>S. Amer. Mineral. 54, 961–963. WYCKOFF, R. G. W. (1963) Crystal Structures, vol. 1. John Wiley and Sons, New York.

Manuscript received, March 9, 1971; accepted for publication, April 14, 1971.

1888