

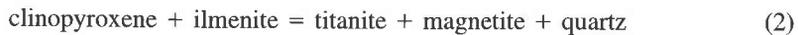
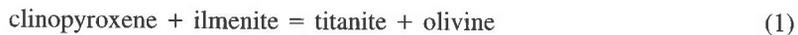
Equilibria among titanite, hedenbergite, fayalite, quartz, ilmenite, and magnetite: Experiments and internally consistent thermodynamic data for titanite

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ABSTRACT

For iron-rich compositions, reactions among titanite, clinopyroxene, olivine, quartz, magnetite, and ilmenite can constrain the conditions under which common rocks form. Phase equilibrium experiments in the system CaO-FeO-Fe₂O₃-TiO₂-SiO₂ constrain two of these reactions:



in the range 600–1100 °C and <1 bar –3.8 kbar, and under controlled f_{O_2} conditions. The experimental results indicate that the assemblage clinopyroxene-ilmenite represents reducing conditions relative to titanite-magnetite-quartz. However, the latter is not necessarily indicative of oxygen fugacity conditions higher than that of the FMQ buffer. The data also strongly suggest that the equilibrium positions of reaction 1 and 2 reflect changes in pressure, temperature, f_{O_2} , a_{SiO_2} , $\mu_{\text{Fe}^{2+}\text{Ti}^{4+}\text{Fe}^{3+}_2}$, and $\mu_{\text{CaFe}_{-1}}$ in the coexisting Fe-Ti oxides and silicates. Standard state thermodynamic data and solution models for Ca-Fe olivine and clinopyroxene solid solutions, Fe-Ti ilmenite and magnetite solid solutions, and quartz (QUILF database) allow refinement of standard state thermochemical data for end-member titanite (A2/a space group):

$$\Delta H^0 = -2607.41 \text{ kJ/mol}$$

$$S^0 = 106.00 \text{ J/(mol}\cdot\text{K)}$$

$$\Delta G^0 = -2458.95 \text{ kJ/mol}$$

and require modification of published heat capacity equations to:

$$C_p = \frac{591\,460\,608}{T^3} - \frac{5\,118\,324}{T^2} - \frac{1038.4}{\sqrt{T}} + 247.47.$$

The derived data for titanite are internally consistent with an enthalpy of formation of -2610.13 ± 2.90 (2 std. error) kJ/mol, established by drop solution calorimetry, and are specific to the QUILF database. However, analysis of the experimental results using different internally consistent databases suggest that the revision proposed in this study of the currently accepted entropy value of 129.20 J/(mol·K) is largely independent of the databases used. The new entropy is required by the enthalpy of formation for titanite adopted here, which may also require revision of the heat capacity.

INTRODUCTION

Titanite (Sphene), CaTiSiO₅, is a common accessory mineral in igneous and metamorphic rocks, and it is probably the most abundant titanium mineral other than ilmenite and rutile. Its principal igneous occurrence is in plutonic rocks (e.g., syenites, diorites, and granites; Deer

et al. 1982), and it is less common in volcanic rocks (Nakada 1991). Titanite is also common in gneisses and schists rich in ferromagnesian minerals (Deer et al. 1982), in calcareous schists (e.g., Gieré 1992), and in high-pressure metamorphic rocks (e.g., Franz and Spear 1985).

Titanite can participate in mineral-mineral, mineral-melt, and mineral-fluid equilibria that can be extremely sensitive to changes in intensive parameters such as pressure, temperature, oxygen fugacity, and fluid composition (e.g., Hunt and Kerrick 1977; Spear 1981; Manning and

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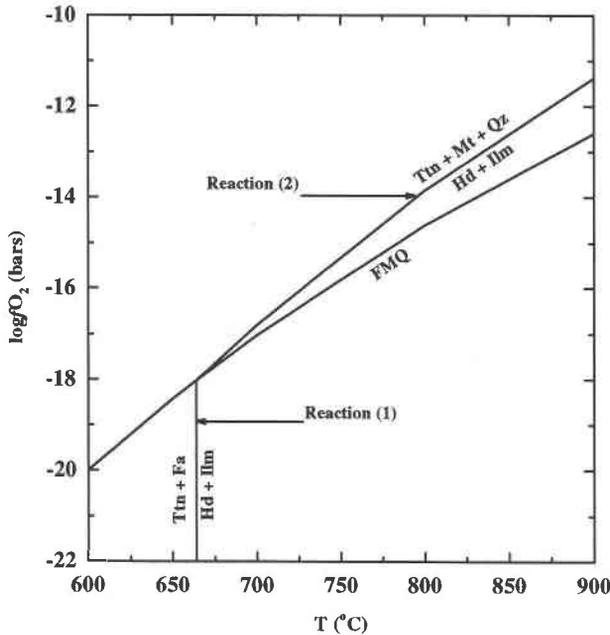
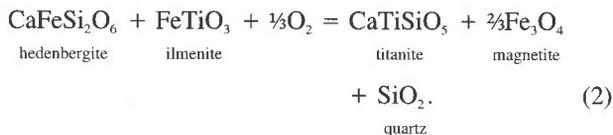
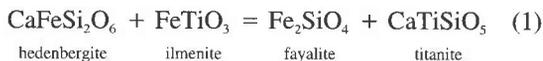


FIGURE 1. Plot of f_{O_2} vs. T showing the metastable reactions: (1) $Fe_2SiO_6 + CaTiSiO_5 = FeTiO_3 + CaFeSi_2O_6$ and (2) $3CaFeSi_2O_6 + 3FeTiO_3 + O_2 = CaTiSiO_5 + 2Fe_3O_4 + SiO_2$ according to Wones (1989) who calculated for pure phases that reactions 1 and 2 should intersect at 664 °C and f_{O_2} conditions of the FMQ buffer at 1 bar (adapted from Fig. 1 of Wones 1989).

Bohlen 1991). Additionally, because of the ability of titanite to retain certain radionuclides (i.e., uranium, thorium, radium), titanite-based ceramics and glass-ceramics have been proposed as hosts of nuclear waste (e.g., Hayward and Cechetto 1982; Gascoyne 1986).

For iron-rich compositions, reactions among titanite, Fe-Ti oxides, clinopyroxene, olivine, and quartz can constrain the P - T - f_{O_2} conditions under which common titanite-bearing rocks form, such as ilmenite-bearing vs. magnetite-and titanite-bearing granites. Moreover, the presence of titanite places tight constraints on the stability of the more abundant rock-forming minerals.

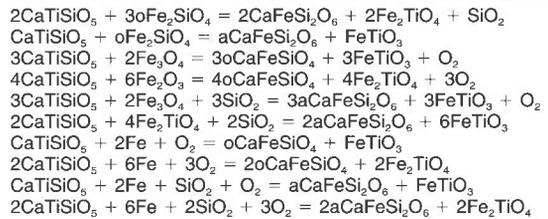
Wones (1989) discussed the importance of titanite-bearing equilibria, particularly the importance of the assemblage titanite-magnetite-quartz. He also calculated the equilibrium T and f_{O_2} conditions of the following titanite-bearing reactions for pure phases in the system CaO - FeO - Fe_2O_3 - TiO_2 - SiO_2 at 1 bar:



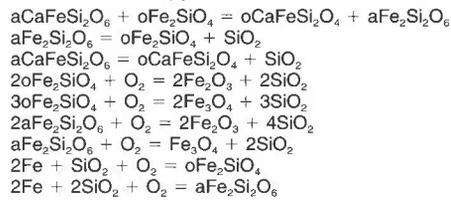
He proposed that reaction 1 and 2 should generate an invariant point at their intersection with the FMQ buffer curve (Fig. 1). Wones' location of reaction 2 above the

TABLE 1. Equilibria among clinopyroxene, olivine, ilmenite, magnetite, titanite, quartz, and iron in the system CaO - FeO - Fe_2O_3 - TiO_2 - SiO_2 pertaining to Figure 2

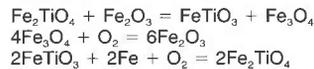
1. Titanite + olivine + clinopyroxene + Fe-Ti oxides + quartz + iron



2. Olivine + clinopyroxene + Fe-Ti oxides + quartz + iron

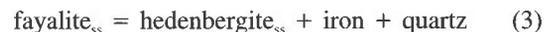


3. Fe-Ti oxides + iron



Note: Titanite and quartz are considered pure phases whereas olivine (o) is treated as solid solution of Fe_2SiO_4 - $CaFeSiO_4$, clinopyroxene (a) of $CaFeSi_2O_6$ - $Fe_2Si_2O_6$, ilmenite of Fe_2O_3 - $FeTiO_3$, and magnetite of Fe_3O_4 - Fe_2TiO_4 components.

FMQ buffer curve is correct for metastable equilibrium among the pure phases. Consequently, the invariant point generated by their intersection is also metastable. If we assume for the moment that titanite (Ttn) and quartz (Qz) are pure phases, and we treat hedenbergite (Hd_{ss}), fayalite (Fa_{ss}), ilmenite (Il_{ss}), and magnetite (Mt_{ss}), respectively, as $CaFeSi_2O_6$ - $Fe_2Si_2O_6$ (Hd-Cfs), Fe_2SiO_4 - $CaFeSiO_4$ (Fa-Ks), $FeTiO_3$ - Fe_3O_3 (Il-Hem), and Fe_3O_4 - Fe_2TiO_4 (Mt-Usp) solid solutions, the variables are: P , T , f_{O_2} , a_{SiO_2} , $\mu_{Fe^{2+}Ti^{4+}Fe^{3+}}$, and $\mu_{CaFe_{-1}}$ (Lindsley and Frost 1992). Hence, chemical equilibrium among physically real phases could be modeled by writing the stoichiometric reactions among phase components (Table 1). Moreover, the reactions among Ttn, Hd_{ss}, Fa_{ss}, Il_{ss}, Mt_{ss}, and Qz should generate a stable invariant point I that lies below the FMQ buffer (Fig. 2). Note that at constant pressure and low f_{O_2} , the divariant assemblage Hd_{ss}+Il_{ss}+Fa_{ss}+Ttn (reaction 1) is terminated at invariant point II by its intersection with the reaction



where fayalite_{ss} is Ca-saturated with respect to hedenbergite_{ss} (Fig. 2). At low oxygen fugacity conditions, the intersection of reaction 3 with



generates a third invariant point (III), which is the termination of the Ttn-absent (Ca-QUILF) reaction

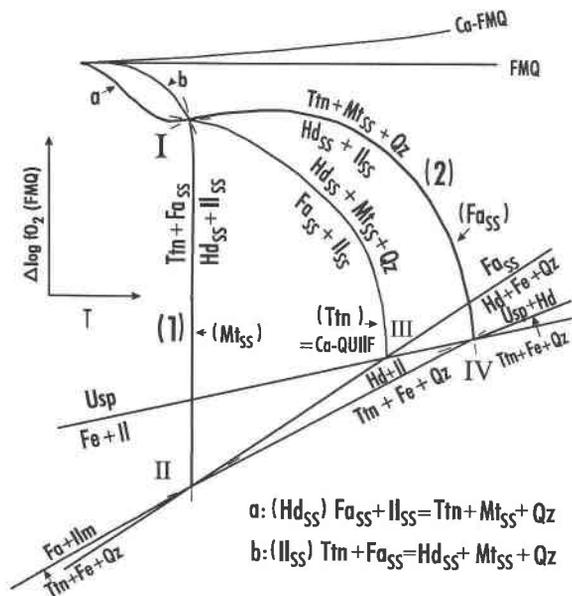
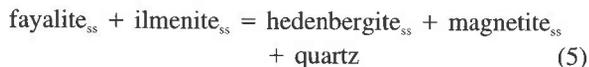
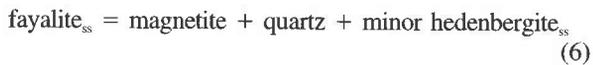


FIGURE 2. Schreinemaker analysis of low pressure equilibria among titanite, hedenbergite_{ss}, fayalite_{ss}, quartz, ilmenite_{ss}, magnetite_{ss}, and Fe⁰.



(Fig. 2). In addition, at constant pressure, relatively low f_{O_2} , and high-temperature conditions the intersection of reaction 4 with reaction 2 generates a fourth invariant point (IV). Thus, for real phases, the path of reaction 2 is very different from that deduced by Wones for pure phases. At point IV, magnetite_{ss}, ilmenite_{ss}, and hedenbergite_{ss} would have to approach the Fe₂TiO₄, FeTiO₃, and CaFeSi₂O₆ end-member compositions, respectively. At higher oxygen fugacity and lower temperature conditions, the Ca-FMQ reaction



and the Hd_{ss}- and Il_{ss}-absent reactions of Figure 2 must all merge asymptotically with the FMQ buffer curve as the fayalite loses Ca and the magnetite loses Ti with decreasing temperature.

In this paper, we concentrate on the geologically more important reactions 1 and 2. To date there have been no experimental constraints for these two reactions. The location of reactions 1 and 2, and therefore of the stable invariant point I, are poorly known. The calculations of Wones place reaction 1 on the FMQ buffer curve at 664 °C at 1 bar, and thus the location of the metastable invariant point on the FMQ buffer curve at that temperature (Fig. 1), values that are consistent with results of our experiments. In contrast, calculations using thermodynamic data for pure phases from Berman (1991) and Holland and Powell (1990) place reaction 1 at 1000 °C and at -62 °C, respectively. A partial explanation for this dis-

crepancy may lie in the fact that the available thermochemical data for titanite (King et al. 1954; Todd and Kelley 1956) are suspect (Xirouchakis et al. 1997a, 1997b). Thus experimental location of reactions 1 and 2 can provide important new constraints on those thermochemical data. We conducted reversed experiments under controlled oxygen fugacity conditions to locate reactions 1 and 2 at pressures ranging from <1 atm to 3.8 kbar. Subsequently, we used the phase equilibrium experiments to derive thermodynamic data for titanite, which is internally consistent with the thermodynamic database and solution models in Andersen et al. (1993) and with the new calorimetric measurements of the enthalpy of formation in Xirouchakis et al. (1997b).

EXPERIMENTAL DETAILS

Starting materials for the phase-equilibrium experiments were synthesized by reacting appropriate mixtures of high-purity chemicals [FeO: weighed as Fe "sponge" (JMC S50389B) and Fe₂O₃ (JM S5040B), SiO₂ (JM S50389B), TiO₂ (JMC 810420), and CaCO₃ (ALPHA Lot 050980)]. Details of the syntheses and microprobe data for starting materials are given in Xirouchakis (1997) and specifically for titanite in Xirouchakis et al. (1997a, 1997b). The phases used in the experiments were extremely pure, as indicated by powder X-ray diffraction (XRD), electron microprobe analysis, and optical examination. Reactions 1 and 2 were constrained by reacting mechanical mixtures of crystalline phases (Hd-II or Hd_{ss}-Il_{ss} and Ttn-Fa, and Hd-II or Hd_{ss}-Il_{ss} and Ttn-Mt_{ss}-Qtz or Ttn- $\frac{2}{3}$ Mt-Qtz) using the double capsule technique under anhydrous (in vacuo) and hydrothermal conditions.

The <1 atm experiments were performed in fast-quench (vertical) platinum-wound furnaces. The temperature was monitored with Pt-Pt₉₀Rh₁₀ thermocouples calibrated against the melting point of Au at room pressure. The starting mechanical mixtures were loaded into either iron capsules with tightly fitted lids or (for experiments at oxygen fugacity conditions of the WM, Co-SiO₂-Co₂SiO₄, and Co-CoO buffers) into Ag₈₀Pd₂₀ capsules ($T \leq 1050$ °C) or Ag₇₀Pd₃₀ capsules ($T \geq 1050$ °C) crimped at the top and sealed in silica-glass tubes. (While not strictly buffered, the actual oxygen fugacity conditions for the Fe-saturated experiments in silica-glass tubes are effectively limited between the reactions: iron + silica = fayalite and iron + ilmenite = spinel; lower and upper limit, respectively.) The buffer mixtures were also loaded in crimped Ag₈₀Pd₂₀ or Ag₇₀Pd₃₀ capsules and placed between the capsules that contained the starting mechanical mixtures. The experiments were terminated by dropping the silica-glass tubes into a glass beaker filled with water or by cooling them in a stream of compressed air. No changes were observed in the Fe capsules that could indicate oxidation during the experiments. Partial alloying of Co from the Co-CoO buffer with the inner walls of Ag₈₀Pd₂₀ capsules was observed. However, because the largest part of the Co-CoO mixture was recovered, it is believed that the integrity of the buffer was not affected.

The 1 to 3.8 kbar experiments were performed in standard cold-seal pressure vessels (e.g., Kerrick 1987) with H₂O as the pressure medium placed in horizontal resistance heaters. The temperature was measured by type K thermocouples. The hot spot in each pressure vessel was calibrated using a sheathed thermocouple calibrated against a NIST standard thermocouple. Pressure was measured by a Bourdon-tube gauge. Approximately 60 to 80 mg of the starting mixtures were loaded with 10 μ l H₂O in Ag₈₀Pd₂₀ capsules, welded shut, and enclosed together with the appropriate solid oxygen buffer (i.e., WM, Co-CoO, FMQ) plus 50 μ l H₂O in thick-walled, 38 mm long gold capsules. The experiments were terminated by taking the vessels out of the heaters and isobarically cooling them by flowing a stream of compressed air over them. Subsequently, the pressure vessels were left to cool for approximately 20 to 45 min under pressure and then the pressure was released. The gold and Ag₈₀Pd₂₀ capsules were retrieved, cleaned, and weighed to ensure that no significant weight change or vapor loss had occurred. The capsules were punctured, weighed, placed in a glass vial and heated at 110 °C, and weighed again. Vapor condensation on the walls of the vial and/or significant weight change after heating strongly suggest the presence of a vapor phase throughout the duration of the experiment. No alloying was detected optically or by occasional EDS analyses between the Co-CoO mixture and the Ag₈₀Pd₂₀ capsules in the hydrothermal experiments. Temperature uncertainties include thermocouple and hot-spot calibration uncertainties plus the standard deviation as calculated from the daily readings. Pressure uncertainties reflect the range of values as recorded daily.

The experimental products were examined in oil mounts under a petrographic microscope and by powder XRD using CuK α radiation. The buffer mixtures after each experiment were examined to ensure that all phases remained present i.e., (solids + vapor). In addition, some of the experimental products were examined optically by backscattered electron imaging and EDS analysis. Samples were also selected for quantitative analysis (WDS) using a CAMEBAX, four-spectrometer, electron microprobe. Analytical conditions were 15 kV, a 20 nA beam current, 1 μ m beam, and counting times of 50 or 60 s with required precision of 0.3%. The standards were anorthite (USMN 137041) and titanite (C.M. Taylor Corporation) for Ca (CaK α), Si(SiK α), and O(OK α), ilmenite (USMN 96189) for Ti(TiK α), and fayalite (USMN 85276) for Fe(FeK α). Oxygen analyses were performed with a multilayered WSi crystal. Raw data were reduced mainly with the PAP and with the ZAF procedure in some cases. The analyses were accepted if the sum of the elements was 100 \pm 2 wt% (Xirouchakis 1997; see Tables 2,¹ 3,¹ and 4¹).

¹ For a copy of Tables 2, 3, and 4, Document item AM-98-011, contact the Business Office of the Mineralogical Society of America (see inside front cover of recent issue) for price information. Deposit items may also be available on the American Mineralogist web site at <http://www.minsocam.org>.

It was not always feasible to collect mineral composition data because of the fine grain size and the textural relations, particularly for titanite that grew at the expense of hedenbergite and ilmenite. Fe-Ti oxide compositions were also determined from calibration curves with a precision of 1–2% (Lindsley, unpublished data) based on the XRD pattern of synthetic single phase Il_{ss} [i.e., (214), (116), and (204) peaks] and Mt_{ss} [i.e., (622), (533), (620), and (440) peaks]. Most of the reported compositions for hedenbergite_{ss} were determined with the microprobe. If the XRD pattern permitted, the (220), (31 $\bar{1}$), and (150) reflections were also utilized to derive the Ca content from the calibration curves of Turnock et al. (1973).

Grain overlap and fluorescence are likely responsible for the apparent Ti detected in fayalite and hedenbergite, Ca and Si in the iron-titanium oxides, and Fe in titanite. This conclusion is based on: (1) the observed textural relations in the experimental products; (2) the highly variable concentration of these elements within the same grain and from grain to grain; (3) that the concentration of these elements is similar in product phases and in synthetic phases from mechanical mixtures that showed no change after reaction and in starting mechanical mixtures; and (4) the fact that the apparent percentages of Ca and Si in the oxides, Ti in hedenbergite, and Fe in titanite correlate better with the weight content of each of these elements in the starting mechanical mixtures than with temperature and pressure or oxygen fugacity conditions (Xirouchakis 1997).

As a result, clinopyroxene and iron-titanium oxides are treated as CaFeSi₂O₆-Fe₂Si₂O₆, FeTiO₃-Fe₂O₃, and Fe₃O₄-Fe₂TiO₄ solid solutions, respectively. Minor adjustments were made to correct for the apparent Ca and Si in the oxides, and Ti in clinopyroxene based on the degree and sign of correlation among the elements from the analyses of the products and minerals of known composition in unreacted mechanical mixtures (Xirouchakis 1997). Based on the available data, titanite and fayalite are considered to retain end-member composition (i.e., CaTiSiO₅ and Fe₂SiO₄, respectively). The assumption that fayalite retains end-member composition is not critical because the Ca content of fayalite in the temperature range 600–800 °C would have been generally small, \leq 1 mol% (Davidson and Lindsley 1994). All but one of the titanite analyses come from reacted Ttn+ $\frac{1}{2}$ Mt+Qz mixtures that always produced the Ilmenite-absent assemblage Ttn+Mt_{ss}+Qz+Hd_{ss}. Thus, for titanite there are not as many chemical data to support firmly this assumption of end-member composition. However, indirect evidence comes from the observation that mixtures of end-member titanite, magnetite, and quartz equilibrated at *T*, *P*, and *f*_{O₂} conditions where titanite is stable, produced a final mixture of iron-free Ttn, Mt_{ss}, Qz, and Hd_{ss}.

EXPERIMENTAL RESULTS

Reaction 1

The results for reaction 1 are listed in Table 5 and all but one data point at 3.4 kbar and 750 °C (CCO buffer)

TABLE 5. Reaction 1: Fe-saturated experiments (i.e., Fe capsule container)

<i>P</i> (bar)	<i>T</i> (°C)	Duration (h)	Reactants	Products	Experiment no.
<1	1000(3)	192	Ttn-Fa	Pxd, Il, Fa, Ttn	1
<1	1000(3)	192	Wo ₅₀ FS ₅₀ -Il ₁₀₀	Pxd, Il, Fa?	2
<1	900(10)	816	Wo ₅₀ FS ₅₀ -Il ₁₀₀ *	no change	87
<1	900(10)	816	Ttn-Fa†	Wo_{48±11} Il_{99±11} Fa, Ttn	88, C
<1	900	1084	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change	95
<1	900	1084	3Ttn-Fa	Hd, Il, Ttn	96, C
<1	800(10)	1764	Wo ₅₀ FS ₅₀ -Il ₁₀₀ *	no change	83
<1	800(10)	1764	Ttn-Fa†	Fa, Ttn, growth of Hd and appearance of Il	84
<1	725(10)	1056	Wo ₅₀ FS ₅₀ -Il ₁₀₀ *	Hd, Il	85
<1	725(10)	1056	Ttn-Fa†	Fa, Ttn, growth of Hd and appearance of Il	86
Experiments with the wüstite-magnetite (WM) buffer					
2000(10)	800(10)	21.5	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change; Wo _{50±11} Il _{99±11}	10
2000(10)	800(10)	21.5	Ttn-Fa	Wo_{48±11} Il_{98±21} Fa, Ttn	9, C
2000(10)	699(10)	48	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change; Wo _{49.5±11} Il _{99±11}	12
2000(10)	699(10)	48	Ttn-Fa	Fa, Ttn, Wo_{47±21} Il_{97±11}	11, C
Experiments with the Co-cobalt oxide (CCO) buffer					
3400(20)	750(10)	3.5	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change	91
3400(20)	750(10)	3.5	3Ttn-Fa	Wo_{42±21} Il_{96±2,196±11} Ttn	92, C
2000(10)	800(10)	29.5	Wo ₄₅ FS ₅₅ -Il ₉₅	no change, Il _{94±11}	89
2000(10)	800(10)	29.5	Ttn-Fa	Wo_{43±21} Il_{96±21} Ttn	90, C
2000(10)	700(10)	168	Wo ₄₅ FS ₅₅ -Il ₁₀₀	Wo _{45.5±11} Il, Fa??, Qz??	13a
2000(10)	700(10)	168	Ttn-Fa	Wo_{46±21} Il_{94±21} Ttn, Fa	13b, C
2000(10)	657(10)	524.5	Wo ₄₅ FS ₅₅ -Il ₁₀₀	no change	14a
2000(10)	657(10)	524.5	Ttn-Fa	Hd, Il, Fa, Ttn	14b
2000(10)	656(10)	550	Wo ₄₅ FS ₅₅ -Il ₁₀₀	no change	20
2000(10)	656(10)	550	Ttn-Fa	Hd, Il_{96±11} Ttn	21
2000(10)	645(10)	456	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change	22
2000(10)	645(10)	456	Ttn-Fa	Hd, Il_{96±11} Fa, Ttn	23
2000(10)	606(10)	456	Wo ₅₀ FS ₅₀ -Il ₁₀₀	no change	19
2000(10)	606(10)	456	Ttn-Fa	Il, Fa, Ttn	18
2000(10)	595(10)	550	Wo ₄₅ FS ₅₅ -Il ₁₀₀	no change; Wo _{45.5±11} Il ₁₀₀	15a
2000(10)	595(10)	550	Ttn-Fa	Wo_{46±21} Il_{96±1,197±11} Fa, Ttn	15b, C
1145(10)	595(10)	1360.5	Wo ₄₅ FS ₅₅ -Il ₁₀₀	Hd, Il, Qz??	28
1145(10)	595(10)	1360.5	Ttn-Fa	Hd, Il, Fa, Ttn	29

Note: Bold fonts indicate new phases. Italics denote phases in small mounts. Question marks denote the probable (?) vs. possible (??) presence of a phase. Brackets [] indicate mineral compositions determined using X-ray calibration curves; otherwise the reported compositions were established using the electron microprobe. Clinopyroxene compositions are represented by Wo_x (e.g., Hd₅₀FS₁₀ or Wo₄₅FS₅₅ by Wo₄₅), ilmenite_{ss} is represented by Il_x (e.g., Il₉₅Hm₅ by Il₉₅), and magnetite_{ss} is given by either usp_x or mt_x (e.g., Mt₅₀Usp₅₀ by Usp₅₀ or Mt₅₀). The letter C indicates experiments employed as constraints in the thermodynamic modeling.

* Seeded with 6 wt% CaTiSiO₃.

† Seeded with 5 wt% Wo₄₅FS₅₅.

are shown in Figures 3, 4, 5, and 6. In all experiments the mechanical mixtures of Ttn+Fa reacted to Hd_{ss} (or pyroxenoid)+Il_{ss}. The grain size of the experimental products did not exceed 10 μm. At higher temperatures (i.e., dry: ≥900 °C; hydrothermal: ≥700 °C) Hd_{ss}+Il_{ss} almost completely replaced Ttn+Fa. Less complete reaction was observed at lower temperatures, but nevertheless, the formation of Hd_{ss}+Il_{ss} was clear. At 1000 °C and <1 atm, Hd_{ss} partially transformed to pyroxenoid (ferrobustamite). Product Hd_{ss}+Il_{ss} tended to form aggregates overgrowing either Ttn or Fa depending on the starting molar ratios of these phases. Mixtures of Ttn and Fa seeded with Hd_{ss} also showed evidence of reaction: growth of the Hd peaks and appearance of Il peaks. In the mixtures seeded with Ttn grains, characteristic Ttn peaks [e.g., 111, 200, 002, 202] either disappeared or decreased.

Reaction 2

The results for the equilibrium hedenbergite_{ss}-ilmenite_{ss}-quartz-magnetite_{ss}-titanite (HIQMT) are presented in Table 6 and all but two data points at 3.4 kbar 700 and 800

°C (FMQ buffer) are depicted in Figures 3, 4, 5, and 6. The experimental results at 3.4 kbar are identical to the ones at 3 and 3.8 kbar (Fig. 6). At conditions where Hd_{ss}+Il_{ss} and Hd+Il mixtures reacted to Ttn+Mt_{ss}+Qz or the companion mixtures consisting of Ttn+Mt_{ss}+Qz or Ttn+½Mt+Qz produced either minor Hd_{ss} or Il_{ss} but never both. The presence of either Hd_{ss} or Il_{ss} depended on the Ti content of the Mt in the starting mechanical mixture. Where the original Mt contained no Ti, Hd was observed to contain Qz inclusions and to be in contact with grains of Mt_{ss} and Ttn. If the original Mt was titaniferous, then Il was dispersed in the matrix but only close to grains of Mt_{ss}. In contrast, where Hd_{ss} and Il_{ss} grew at the expense of Ttn+Mt_{ss}+Qz, or Ttn+½Mt+Qz, the companion mixtures of Hd+Il and Hd_{ss}+Il_{ss} did not change.

At *P* < 1 atm, the phase relations are complicated by the transformation of quartz to tridymite at *T* > 900 °C, by partial conversion of end-member hedenbergite and hedenbergite_{ss} to pyroxenoid at *T* ≥ 940 °C, and by partial melting at *T* ≥ 1050 ± 10 °C. Nevertheless, at *T* ≥ 800 °C (FMQ buffer) and at *T* ≥ 940 °C (Co-CoO buffer), the

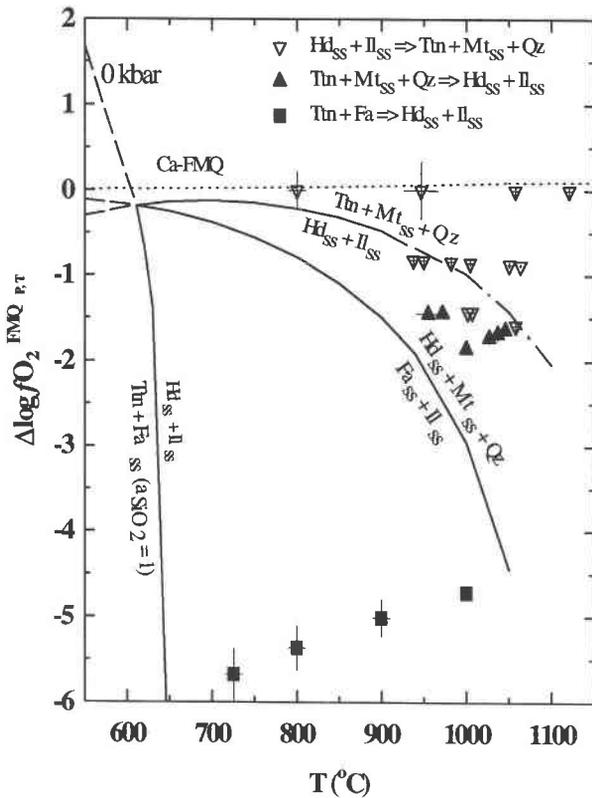


FIGURE 3. $\Delta \log f_{O_2}^{FMQ, 1 \text{ bar}, T}$ - T plot of experimental results at 0 kbar and calculated phase relations among Ttn, Hd_{ss} , Fa_{ss} , Il_{ss} , Mt_{ss} , and Qz with the derived thermodynamic data for titanite. Dashed lines represent the metastable extensions of the calculated reactions. At $T \geq 940$ °C the dashed-dotted line represents the calculated metastable extension of the isobarically univariant assemblage Ttn+ Mt_{ss} +Cristobalite+ Il_{ss} + Hd_{ss} . See text for discussion.

stable assemblage is titanite, magnetite_{ss}, and the stable silica polymorph. At lower f_{O_2} conditions, those of the Co-SiO₂-Co₂SiO₄ buffer at $T < 1000$ °C and WM at $T < 1050$ °C, pyroxenoid and ilmenite_{ss} are stable. Starting mechanical mixtures containing either pyroxenoid or tridymite were not used. However, powder XRD suggests that the transformation of hedenbergite or hedenbergite_{ss} to pyroxenoid, and quartz to tridymite, was mostly complete after equilibration. At higher pressures where the rest of the experiments were performed these complications were avoided.

The experimental results suggest that at 1 to 2 kbar and 600 to 800 °C reaction 2 is located between the FMQ and Co-CoO buffer curves whereas at ≥ 3 kbar and 675 to 750 °C it lies above the FMQ buffer curve.

DISCUSSION

The technique of mathematical programming (Berman et al. 1986; Berman 1988) was employed to derive thermodynamic data for titanite consistent with the phase equilibrium and calorimetry experiments (Xirouchakis et

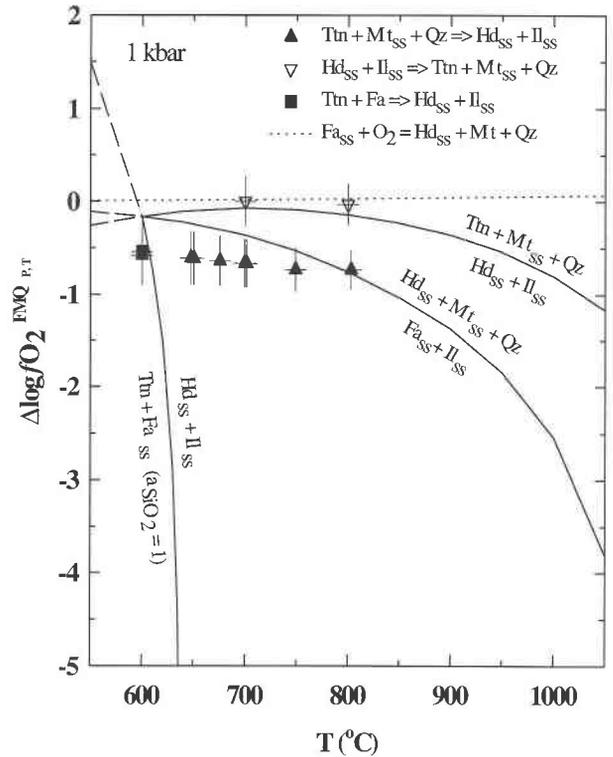


FIGURE 4. $\Delta \log f_{O_2}^{FMQ, 1 \text{ bar}, T}$ - T plot of experimental results at 1 kbar and calculated phase relations among Ttn, Hd_{ss} , Fa_{ss} , Il_{ss} , Mt_{ss} , and Qz with the preferred thermodynamic data for titanite. Dashed lines represent the metastable extensions of the calculated reactions.

al. 1996b, 1997b). This technique was deemed most appropriate because many of the experimental constraints are only half-reversals (i.e., reaction 1). Moreover, experimental data inconsistent with the thermodynamic model of choice can be identified. However, this approach is not without drawbacks. Formal uncertainties cannot be assigned to the optimized parameters and, in many cases, the solution obtained is an element on the boundary of the feasible solution set (Olbricht et al. 1994). Nonetheless, in this context each phase-equilibrium experiment represents a change in the Gibbs free energy for a given reaction r that can be expressed in terms of phase components by the inequality:

$$\Delta G_r = \sum_i^n v_i' \cdot \mu_i' \leq 0 \quad (7)$$

(e.g., Gordon 1973; Day and Kumin 1980; Berman et al. 1986). In the above relation v_i' and μ_i' , respectively, denote the stoichiometric coefficient and the chemical potential of the i th component in the j th phase defined by the equation:

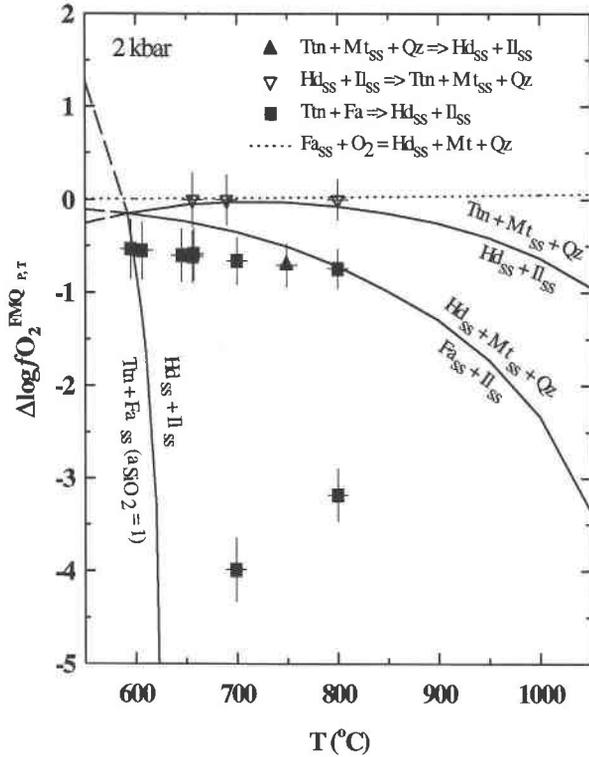


FIGURE 5. $\Delta \log f_{O_2}^{FMQ, r, T}$ plot of experimental results at 2 kbar and calculated phase relations among Ttn, Hd_{ss} , Fa_{ss} , Il_{ss} , Mt_{ss} , and Qz with the preferred thermodynamic data for titanite. Dashed lines represent the metastable extensions of the calculated reactions.

$$\mu_i^j = \Delta H_{i,T_r,P_r}^{0,j} - T \cdot S_{i,T_r,P_r}^{0,j} + \int_{T_r}^T C_p dT - T \cdot \int_{T_r}^T \frac{C_p}{T} dT + \int_{P_r}^P V_{i,T,P}^{0,j} dP + R \cdot T \cdot \ln a_i^j \quad (8)$$

where ΔH^0 and S^0 are the standard state molar enthalpy and entropy of formation of component i in phase j from the elements at the reference temperature (T_r) and pressure (P_r); C_p and $V_{i,T,P}$ are the molar heat capacity function and molar volume at T and P , respectively; and a_i^j is the activity of component i in phase j .

To evaluate Equation 8, standard state thermodynamic data and activity expressions for the solid-solution phases are needed. Thus it is necessary to select a thermodynamic database as a starting point. Almost any internally consistent database should be suitable. For example, we could have used the database of Berman (1988, 1991) as augmented by the oxide and pyroxene-olivine models of Ghiorso (1990) and Ghiorso and Sack (1991), and Sack and Ghiorso (1994a, 1994b, 1989), respectively, or the database of Holland and Powell (1990) if solution models internally consistent with it existed. We have chosen instead to adopt the QUIIF database, which comprises the silicate model of Davidson and Lindsley (1989), the Fe-

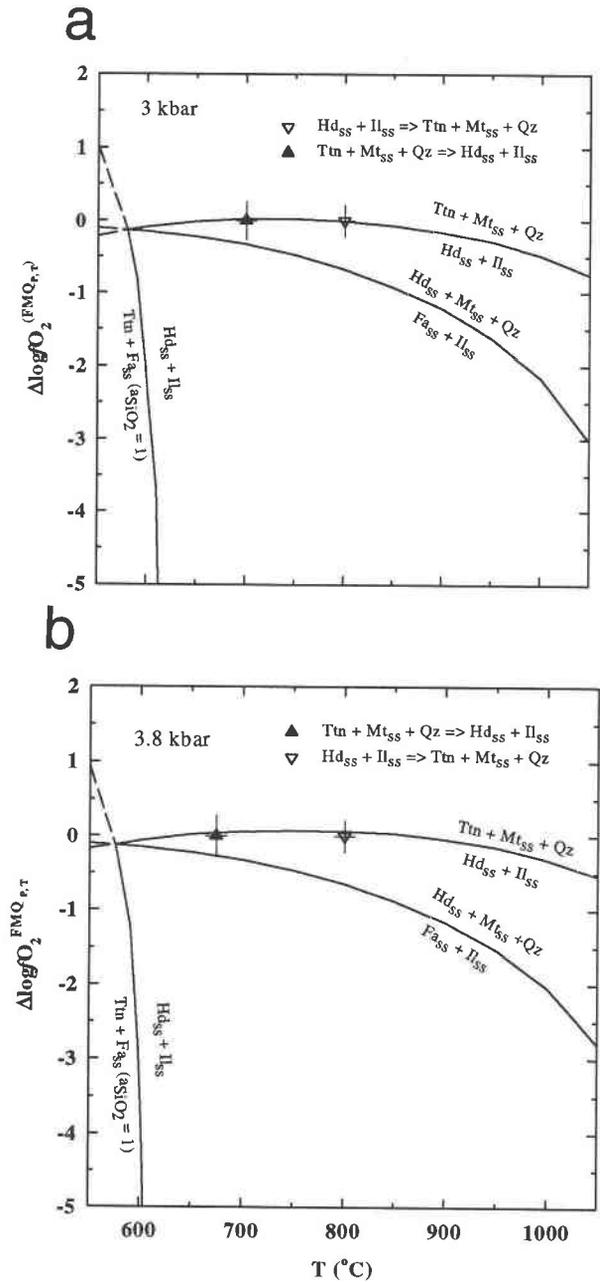


FIGURE 6. $\Delta \log f_{O_2}^{FMQ, r, T}$ plot of experimental results at (a) 3 kbar and (b) 3.8 kbar and calculated phase relations among Ttn, Hd_{ss} , Fa_{ss} , Il_{ss} , Mt_{ss} , and Qz with the preferred thermodynamic data for titanite. Dashed lines represent the metastable extensions of the calculated reactions.

Ti oxides model of Andersen and Lindsley (1988), and the combination of these into the "QUIIF" model (Frost et al. 1988; Lindsley and Frost 1992; Frost and Lindsley 1992; Andersen et al. 1993). Relative to the Ghiorso and Sack (1990; 1991) oxide model, the Andersen and Lindsley model (1988) more successfully reproduces the experimental results for Fe-Ti oxides at 600–700 °C, a key

range for the present study. Most importantly, use of the QUIF database permits relatively straightforward incorporation of the new titanite data into the widely used "QUILF" program (Andersen et al. 1993). The only modification was the addition of thermal expansivity and compressibility terms for ilmenite (FeTiO_3) from a least-squares fit to the data of Wechsler and Prewitt (1984).

The adoption of the QUIF database entails certain consequences. The oxide model and the combined oxide-silicate model were based on the unpublished data of Haas (personal communication to D.H.L.; see also discussion in Lindsley and Frost 1992, p. 988) for oxygen buffers, which were revised and reissued by Haas and Hemingway (1992). Thus a key decision was whether to incorporate the newer buffer data, which would involve remodeling both the oxide and the silicate portion of the "QUILF" model, a major undertaking. The actual differences between the 1986 and 1992 buffer data are small, between 500–1000 J/mol. Trial calculations indicated that remodeling the older data would result in negligibly small changes (well within the experimental uncertainty) in the oxide and the silicate models. Accordingly we decided to retain the older database in our calculations. To facilitate the inclusion of the derived thermodynamic data for titanite to the QUILF program the empirical heat capacity polynomial of Haas (personal communication to D.H.L.; see also discussion in Lindsley and Frost 1992, p. 988) and Haas and Hemingway (1992)

$$C_p = \frac{a_1}{T^3} + \frac{a_2}{T^2} + \frac{a_3}{T} + \frac{a_4}{\sqrt{T}} + a_5 + a_6 \cdot T + a_7 \cdot T^2, \quad (9)$$

and the volume equation of Robinson et al. (1982)

$$V_{T,P}^0 = b_1 + b_2 \cdot T + b_3 \cdot e^{(-T/300)} + b_4 \cdot P + b_5 \cdot e^{(-P/35000)} \quad (10)$$

were used in the calculations where T is in kelvins and P in bars. Despite its apparent complexity, Equation 9 is quite flexible and can be transformed to the heat capacity polynomials proposed by Haas and Fisher (1976) and Berman and Brown (1985). The standard state molar volume of titanite was taken from Xirouchakis et al. (1997a). The thermal expansivity and compressibility parameters were derived from nonlinear regression of unit-cell volume measurements at ambient pressure and up to 1000 °C (Taylor and Brown 1976), and at room temperature from 1 atm up to 70 kbar (Kunz et al. 1996; Angel et al., unpublished manuscript). Thus for a molar volume of 5.560 ± 0.002 J/bar the thermal expansivity (1/K) and compressibility parameters (1/bar) are: $b_1 = 5.497705$, $b_2 = 1.48606 \times 10^{-4}$, $b_3 = 4.8950 \times 10^{-2}$, $b_4 = -4.01012 \times 10^{-6}$, and $b_5 = 4.730 \times 10^{-2}$. The data of Taylor and Brown (1976) were preferred, despite the relatively high unit-cell volume of their sample, because their measurements agree better than the data of Ghose et al. (1991) with the data of Hughes et al. (1997) and Xirouchakis et al. (1997a) for end-member titanite at room pressure and temperature. Like the Ghose et al. (1991) titanite sample, the crystal used by Zhang et al. (1995) has a suspiciously large unit-cell volume and it apparently contains wollas-

tonite and quartz impurities (Tanaka et al. 1988). Thus, the high-temperature heat capacity data of Zhang et al. (1995) were also avoided.

Once standard-state thermodynamic data and solution models were adopted, then the thermochemical properties of titanite could be constrained by the inequalities

$$\mu_{\text{CaFeSi}_2\text{O}_6}^{\text{hdss}} + \mu_{\text{FeTiO}_3}^{\text{ilss}} - \mu_{\text{CaTiSiO}_5}^{\text{Tn}} - \mu_{\text{Fe}_2\text{SiO}_4}^{\text{fa}} \leq 0 \quad (11)$$

$$3\mu_{\text{CaFeSi}_2\text{O}_6}^{\text{hdss}} + 3\mu_{\text{FeTiO}_3}^{\text{ilss}} + \mu_{\text{O}_2} - 3\mu_{\text{CaTiSiO}_5}^{\text{Tn}} - 2\mu_{\text{Fe}_3\text{O}_4}^{\text{mss}} - 3\mu_{\text{SiO}_2}^{\text{qtz}} \leq, \geq 0. \quad (12)$$

Experiments at $T > 900$ °C and < 1 atm were not included because of polymorphic changes, i.e., quartz to tridymite, and most importantly clinopyroxene to pyroxenoid. The uncertainties in pressure, temperature, oxygen fugacity, and the composition of the minerals were incorporated in the calculations by considering all possible combinations of their numerical values. Thus, the uncertainty ellipsoid of each datum (Clifford 1973) is approximated empirically as a range in ΔG_r values that were considered simultaneously in the linear programming problem for all the constraints. The experimental data considered in the thermodynamic modeling, a total of sixteen experiments, are marked by the letter C in Tables 5 and 6. Further constraints on the standard-state enthalpy of formation of titanite come from the most recent calorimetric study (Xirouchakis et al. 1997b). Under the conditions of the calorimetry and phase-equilibrium experiments, titanite has the high-temperature structure (A2/a). Except for the molar volume, the rest of the available data and constraints are also for the high-temperature phase. Thus, the $P2_1/a$ to A2/a transition in titanite (Taylor and Brown 1976; Kunz et al. 1996) was not considered, and it was assumed that the properties of titanite can be described effectively with the same equation of state above and below the transition.

Reaction and calorimetry constraints were considered simultaneously in conjunction with the following objective function:

$$F = \sum_1^n \frac{(Y_{\text{obs}} - Y_{\text{calc}})^2}{|Y_{\text{calc}}|} \quad (13)$$

which minimizes the difference between calculated and measured quantities (Taylor 1982; Grace 1990). Agreement between an enthalpy of formation of -2610.13 ± 2.90 kJ/mol (Xirouchakis et al. 1997b) and the phase-equilibrium constraints of this study required the simultaneous optimization of heat capacity equation terms. The final optimized parameters for titanite were chosen based on the agreement between calculated and experimentally determined phase relations and the behavior of the $C_p(T)$ and $(H_T - H_{298})$ functions at high temperatures.

The heat capacity polynomial coefficients were optimized by minimizing the discrepancy between the calculated and measured heat content ($H_T - H_{298}$) at 702 °C (122.985 kJ/mol). The latter value was obtained from the difference of the enthalpy of drop solution and prelimi-

TABLE 6. Reaction 2: experiments with the wüstite-magnetite (WM) buffer

<i>P</i> (bar)	<i>T</i> (°C)	Duration (h)	Reactants	Products	Experiment no.
<1	1058(3)	66	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, glass, Usp _[8577] , Pxd, Il	79
<1	1058(3)	66	Ttn-Usp ₆₀ -Qz	Ttn, Trid, Usp _[8577] , <i>glass, Pxd?</i>	80
<1	1046(3)	157	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Pxd, Il, <i>glass?</i>	87
<1	1046(3)	157	Ttn-Usp ₆₀ -Qz	Pxd, Il, Ttn, Usp _[60] , <i>glass?</i>	88
<1	1037(4)	184	Wo ₅₀ Fs ₉₀ -2Il ₁₀₀	Pxd, Il	85
<1	1037(4)	184	Ttn-Usp ₆₀ -Qz	Pxd, Il	86
<1	1027(3)	63	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Pxd, Il	83
<1	1027(3)	63	Ttn-Usp ₆₀ -Qz	Pxd, Il, Trid? , Usp _[707]	84
<1	1000(3)	43.5	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Pxd, Il	68
<1	1000(3)	43.5	Ttn-½Mt-Qz	Pxd, Il, Ttn, Mt _{ss?} , Trid	69
Experiments with the Co-quartz-Co₂SiO₄ (CoQzOl) buffer					
<1	1007(2)	90.5	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Trid, Usp _[70±5] , Pxd, Il	54
<1	1007(2)	90.5	Ttn-Mt-Qz	Ttn, Trid, Usp _[70±5] , Pxd	55
<1	1002(2)	118	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Trid, Usp _[8077] , Hd/Pxd	72
<1	1002(2)	118	Ttn-Mt-Qz	Ttn, Usp _[8077] , Trid/Qz, Pxd	73
<1	972(2)	159.5	Wo ₅₀ Fs ₉₀ -2Il ₁₀₀	Pxd, Il, Trid??	97
<1	972(2)	159.5	Ttn-Usp ₆₀ -Qz	Pxd, Il _[99±1] , Ttn, Trid	98
<1	955(14)	237	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Pxd, Il, Trid??	89
<1	955(14)	237	Ttn-Usp ₆₀ -Qz	Pxd, Il _[99±1] , Ttn, Trid, Usp _[60] ?	90
Experiments with the Co-CoO (CCO) buffer					
<1	1064(1)	144	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Trid, Usp _[70±5] , <i>glass, Pxd</i>	39
<1	1064(1)	144	Ttn-Mt-Qz	Ttn, Trid, Usp _[70±5] , Pxd, <i>glass</i>	40
<1	1051(2)	186	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Trid, Usp _[70±5] , <i>glass, Pxd</i>	52
<1	1051(2)	186	Ttn-Mt-Qz	Ttn, Usp _[70±5] , Trid, Pxd, <i>glass</i>	53
<1	1005(2)	186	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Usp _[70±5] , Trid, Pxd/Hd	50
<1	1005(2)	186	Ttn-Mt-Qz	Ttn, Usp _[70±5] , Trid, Hd/Pxd	51
<1	938(3)	183	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Usp _[62±5] , Trid?, Hd	91
<1	938(3)	183	Ttn-Usp ₆₀ -Qz	Ttn, Usp _[60] , Trid, Hd?	92
<1	982(4)	428	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Usp _[62±5] , Trid, Hd/Pxd?	76
<1	982(4)	428	Ttn-Usp ₆₀ -Qz	Ttn, Usp _[60] , Trid/Qz	77
<1	950(3)	424.5	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Usp _[67±5] , Trid, Il, Pxd/Hd	74
<1	950(3)	424.5	Ttn-Usp ₆₀ -Qz	Ttn, Usp _[60] , Trid/Qz	75
1985(10)	749(10)	95.5	Wo ₄₅ Fs ₅₅ -Il ₉₀	Wo _{46±2} , Il _{90±2} , Usp _{37±3}	60
1985(10)	749(10)	95.5	Ttn-Usp ₅₀ -Qz	Wo _{44±2} , Il _{95±1}	61, C
1145(10)	675(10)	428.5	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	no change	27
1145(10)	675(10)	428.5	Ttn-½Mt-Qz	Wo _[43±3] , Il _[94±2] , Ttn, Usp _[157] , Qz	28
1145(10)	650(10)	432	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	no change	25
1145(10)	650(10)	432	Ttn-½Mt-Qz	Wo _[44±2] , Il _[94±2] , Ttn, Mt _{ss?} , Qz	26
1145(10)	599(10)	616.5	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	no change	29
1145(10)	599(10)	616.5	Ttn-½Mt-Qz	Wo _[43±3] , Il _[97±2] , Ttn, Mt _{ss?}	30
1001(10)	701(10)	409	Wo ₄₅ Fs ₅₅ -Il ₉₀	no change	62
1001(10)	701(10)	409	Ttn-Usp ₅₀ -Qz	Wo _[45±3] , Il _[95±2,93±2] , Ttn, Qz	63, C
1020(10)	699(10)	720	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Hd, Il, Qz??	9
1020(10)	699(10)	720	Ttn-½Mt-Qz	Wo _{45±3,45±5]} , Il _{94±1,96±3]} , Ttn, Usp _[50±5] , Qz	10
1002(8)	803(10)	95.5	Wo ₄₅ Fs ₅₅ -Il ₉₀	no change; Wo _{46±2} , Il _{99±2}	58
1002(8)	803(10)	95.5	Ttn-Usp ₆₀ -Qz	Wo _[44±3] , Il _[95±1,95±2] , Ttn, Mt _{ss?}	59, C
1000(10)	647(10)	621	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	no change	48
1000(10)	647(10)	621	Ttn-½Mt-Qz	Wo _[44±2] , Il _[93±3] , Ttn, Usp _[20] ?, Qz	49
Experiments with the fayalite-magnetite-quartz (FMQ) buffer					
<1	1121(4)	4	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Mt _{ss} , Trid, <i>glass, Il, Pxd?</i>	37
<1	1121(4)	4	Ttn-Mt-Qz	Ttn, Trid, Mt _{ss} , <i>glass, Pxd?</i>	38
<1	1058(4)	124	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Usp _[65±5] , Trid, Hd/Pxd	31
<1	1058(4)	124	Ttn-Mt-Qz	Ttn, Usp _[65±5] , Trid, Pxd	32
<1	947(20)	478	Wo ₄₅ Fs ₅₅ -Il ₉₅	Ttn, Usp _[58±5] , Trid, Hd/(Pxd?), Il	141
<1	947(20)	478	Ttn-½Mt-Qz	Ttn, Usp _[55±5] , Trid, Pxd	142
<1	800(10)	1052	Wo ₅₀ Fs ₉₀ -Il ₁₀₀ †	Hd, Il, growth of Ttn, Usp _{46±1} , Qz?	145, C
<1	800(10)	1052	Ttn-½Mt-Qz	Ttn, Usp _[40±5] , Qz, Hd	146
3800(10)	674(10)	72	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	no change	107
3800(10)	674(10)	72	Ttn-½Mt-Qz	Wo _{44±2,41±3]} , Il _[92±5] , Ttn, Qz	108
3700(30)	800(10)	9.5	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Usp _[35±5] , Qz, Hd	113, C
3700(30)	800(10)	9.5	Ttn-½Mt-Qz	Ttn, Usp _[40±5] , Qz, Hd	114
3410(10)	750(10)	89	Wo ₄₅ Fs ₅₅ -Il ₉₅	no change	115B
3410(10)	750(10)	89	Ttn-½Mt-Qz	Wo _{44±1,44±3]} , Il _[92±1,93±1] , Ttn, Qz, Usp _[37±5]	116B
3410(10)	700(10)	355	Wo ₄₅ Fs ₅₅ -Il ₉₅	no change, Qz?	117
3410(10)	700(10)	355	Ttn-½Mt-Qz	Wo _[44±3] , Il _[96±3] , Ttn, Usp _[25±3] , Qz	118
3010(10)	800(10)	12	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Hd, Il, appearance and growth of Ttn, Mt _{ss}	129
3010(10)	800(10)	12	Ttn-½Mt-Qz	Ttn, Usp _[38±3] , Qz, Hd	130
3000(10)	700(10)	381	Wo ₄₅ Fs ₅₅ -3Il ₉₅	Hd, Il, Mt _{ss} , Qz	147
3000(10)	700(10)	381	Ttn-Usp ₂₆ -Qz	Wo _[44±1] , Il _[92±3] , Ttn, Usp _[~25] , Qz	148, C
2000(10)	800(10)	281	Wo ₅₀ Fs ₉₀ -Il ₁₀₀	Ttn, Usp _{42±47} , Qz, Hd, Il	135
2000(10)	800(10)	281	Ttn-½Mt-Qz	Ttn, Usp _{20 to 38,35±5]} , Qz, Hd	136
2000(10)	690(10)	720	Wo ₄₅ Fs ₅₅ -3Il ₉₅	Hd, Il, Qz, Ttn, Usp _{18±17}	153, C

TABLE 6—Continued

<i>P</i> (bar)	<i>T</i> (°C)	Duration (h)	Reactants	Products	Experiment no.
2000(10)	690(10)	720	Ttn-Usp ₆₀ -Qz	Ttn, Usp _{27.5±2.1} Qz, //	154
2000(10)	656(10)	360	Wo ₄₅ Fs ₅₅ -Il ₁₀₀	Ttn, Qz, Mt ₅₅ ?, Hd, //	11
2000(10)	656(10)	360	Ttn- ² / ₃ Mt-Qz	Ttn, Usp _{16±4.1} Qz, Hd	12
1020(10)	800(10)	281	Wo ₄₅ Fs ₅₅ -3Il ₁₀₀	Ttn, Usp _{44±3.1(43±2)} Qz, Hd, //	3, C
1020(10)	800(10)	281	Ttn- ² / ₃ Mt-Qz	Ttn, Usp _{143±4.1} Qz, Hd	4
1010(10)	700(10)	930.5	Wo ₄₅ Fs ₅₅ -3Il ₁₀₀	Ttn, Usp _{135?1} Qz, Hd, // _{97±3}	17
1010(10)	700(10)	930.5	Ttn- ² / ₃ Mt-Qz	Ttn, Usp _{127±5.1} Qz, Hd	18

Note: Representation as in Table 5.

nary solution calorimetry experiments of titanite at 702 °C in 2PbO·B₂O₃ solvent (S. Fritsch, written communication). This value is 2 kJ/mol greater than the one calculated with the existing heat capacity equations (Robie et al. 1978; Robie and Hemingway 1995; Berman 1988, 1991), which are based on the data of King et al. (1954). However, it compares well with the original data if the latter are corrected for 1% phase impurities (King et al. 1954), either CaSiO₃ or CaTiSiO₅ glass, or both (Xirouchakis et al. 1997b). The heat capacity equation parameters reported by Berman (1988, 1991) were used as starting points in the optimization. The *a*₃, *a*₆, and *a*₇ terms were set equal to zero, and *a*₂ and *a*₄ were constrained to be negative (Berman and Brown 1985). However, it was found best to fix the *a*₁ and *a*₂ terms at their given values 5.91460608 × 10⁸ and -5.118324 × 10⁶, respectively (Berman 1988, 1991). The optimized heat capacity equation:

$$C_p = \frac{591460608}{T^3} - \frac{5118324}{T^2} - \frac{1038.4}{\sqrt{T}} + 247.47 \quad (14)$$

results in positive deviations of 2% relative to the currently available experimental data at *T* ≤ 500 K (King et al. 1954; Zhang et al. 1995) and 5% at 500 ≤ *T* ≤ 850 K (Zhang et al. 1995). Consequently, the calculated heat content (*H*_{*T*} - *H*₂₉₈) deviates positively by 5% relative to the original high-temperature heat content data (King et al. 1954). Equation 14 may overestimate the heat capacity and heat content of titanite at high temperature. However, the positive deviations are consistent with corrections that account for contributions mainly from phase impurities

(e.g., CaTiSiO₅ glass, CaSiO₃). Contributions from other types of defects (vacancies?) in the samples of King et al. (1954) and Zhang et al. (1995) are probable but cannot be accounted for as readily.

In a second set of calculations, the preferred set of parameters that describe the ilmenite and magnetite activities (Akimoto model) were allowed to vary within the reported range (Andersen and Lindsley 1988). Similarly, it was found best to optimize at least one of the heat capacity polynomial terms along with the rest of the parameters, i.e., Δ*H*⁰, *S*⁰, and *W*_{*ij*}. The differences in the derived Δ*H*⁰, *S*⁰, and heat capacity equation:

$$C_p = \frac{591460608}{T^3} - \frac{5118324}{T^2} - \frac{1040.1}{\sqrt{T}} + 244.34 \quad (15)$$

are marginal despite the somewhat smaller positive deviations (1–3%) relative to the currently available heat capacity and heat content when compared with Equation 14. The improvement in fitting the experimental data is also minimal. The enthalpy of formation and entropy for titanite consistent with the data in Andersen et al. (1993) are given in rows 1 (preferred) and 2 of Table 7. The graphical solution is shown in Figure 7. Subsequently, the positions of the Ttn-, Fa_{ss}-, and Mt_{ss} and Qz-absent reactions (Fig. 2) were determined (Fig. 3, 4, 5, and 6). The calculations were performed with the enthalpy and entropy values that are consistent with Equation 14 and the preferred *W*_{*ij*} values for ilmenite and magnetite (Andersen and Lindsley 1988), and by minimizing the absolute value (Grace 1990) of the Gibbs free energy,

TABLE 7. Thermodynamic data for titanite

Δ <i>H</i> _{1 bar, 298} ⁰ kJ/mol	<i>S</i> _{1 bar, 298} ⁰ J/mol·K	Δ <i>G</i> _{1 bar, 298} ⁰ kJ/mol	<i>V</i> _{1 bar, 298} ⁰ J/bar	Source
-2607.41	106.00	-2458.95	5.560 ± 0.002	1a, preferred
-2607.20	107.52	-2459.20	5.560 ± 0.002	1b
-2610.13 ± 2.90				2
-2596.60 ± 3.00	129.20 ± 0.80	-2454.60 ± 3.20	5.574 ± 0.007	3
-2601.40 ± 2.34	129.20 ± 0.84	-2459.86 ± 2.43	5.565	4
-2596.27	126.27	-2453.80	5.565	5
-2596.56	129.29	-2454.20	5.565	6
-2596.48 ± 1.77	129.20	-2459.60	5.565	7
-2601.40 ± 2.40	129.20	-2459.86	5.565	8

Note: (1a) This study, consistent with Equation 14; (1b) This study, consistent with Equation 15; (2) Xirouchakis et al. 1997b; (3) Robie and Hemingway 1995; (4) Robie et al. 1978; (5) Xirouchakis and Lindsley 1995; (6) Berman 1988, 1991; (7) Holland and Powell 1990; (8) Holland and Powell 1985.

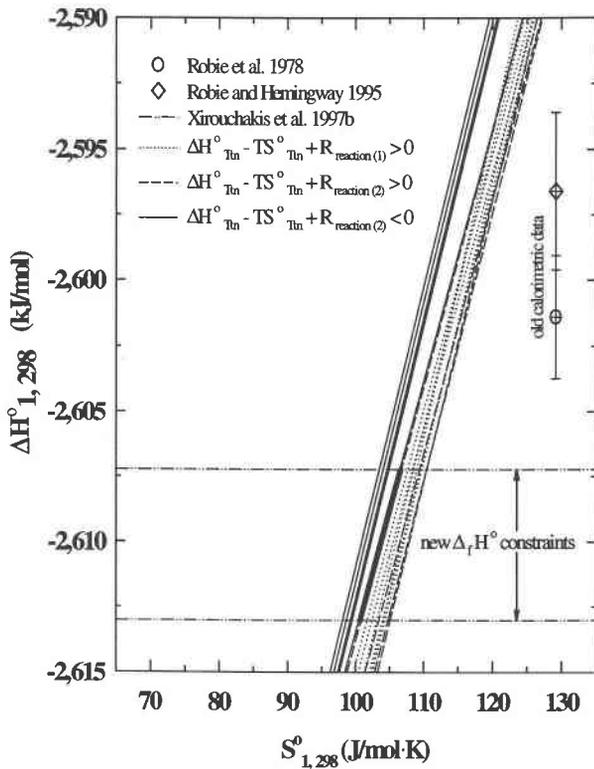


FIGURE 7. Feasible solution (filled area) for ΔH° and S° of titanite as defined by calorimetry and phase equilibrium constraints from reactions 1 and 2. The latter were constructed using the midpoints for each experiment and average mineral compositions and the derived heat capacity polynomial for titanite. R represents the contribution of all the input parameters but ΔH° and S° in Equations 7 and 8.

$$|\Delta G_r| = \left| \sum_i v_i \cdot \mu_i \right| = 0, \quad (16)$$

of the appropriate oxide and oxide-silicate equilibria (Table 1). We did not calculate the Il_{ss} - and Hd_{ss} -absent reactions (Fig. 2) both because they are fairly well constrained by the rest of the equilibria and because of numerical problems at $T \leq 550$ °C. For reaction 1 we calculated the equilibrium conditions assuming $a_{\text{SiO}_2} = 1$ with respect to quartz and solved for f_{O_2} at various pressure and temperature conditions. The agreement between experimental data and calculated phase relations is good except for reaction 2 at 1 bar and at $T > 940$ °C (Fig. 3). The dashed-dotted line represents the approximate location of the metastable extension of the isobarically univariant assemblage $\text{Hd}_{ss} + \text{Il}_{ss} + \text{Ttn} + \text{Mt}_{ss} + \text{cristobalite}$, especially in the range 940–960 °C. The Haas database does not contain data for tridymite and no equation of state is presently available for pyroxenoid (ferrobustamite) solid solutions. End-member ($\text{CaFeSi}_2\text{O}_6$) hedenbergite transforms directly to pyroxenoid (Pxd) of the same composition at ≥ 940 °C. In the range 940–960 °C hedenbergite_{ss} transforms to a Ca-rich hedenbergite_{ss} and an Fe-rich pyrox-

enoid_{ss} (Pxd_{ss}), relative to the original composition of the low-temperature hedenbergite_{ss}. Finally, hedenbergite_{ss} transforms to pyroxenoid_{ss} of the same bulk composition at ≥ 960 °C (Lindsley and Munoz 1969). Calculations based on the phase relations established by Lindsley and Munoz (1969) in the range 940–960 °C and available thermochemical data (O'Neill and Navrotsky 1980; Haselton et al. 1987) for ($\text{CaFeSi}_2\text{O}_6$) ferrobustamite, suggest that the assemblage (Fe-rich) Pxd_{ss} + (Ca-rich) Hd_{ss} + Il_{ss} + Ttn + Mt_{ss} + cristobalite should lie at lower f_{O_2} conditions than the assemblage Hd_{ss} + Il_{ss} + Ttn + Mt_{ss} + cristobalite. Consequently, the univariant assemblage Pxd_{ss} + Il_{ss} + Ttn + Mt_{ss} + cristobalite has to originate at lower f_{O_2} conditions than the approximately calculated f_{O_2} for the assemblage Hd_{ss} + Il_{ss} + Ttn + Mt_{ss} + cristobalite at 960 °C. These estimates imply that the apparent disagreement is largely caused by the unaccounted effect of the hedenbergite_{ss} to pyroxenoid_{ss} transformation at ≥ 940 °C, and to a lesser extent by the quartz to tridymite transformation below 900 °C, and partial melting at $\geq 1050 \pm 10$ °C.

Our primary goal has been to incorporate titanite into the QUIIF database. The thermodynamic data for the rest of the phases in QUIIF (olivine, pyroxenes, Fe-Ti oxides, and quartz) have been optimized for a small, well-constrained experimental database (Davidson and Lindsley 1989; Andersen et al. 1991), and thus the inherent uncertainties are small. However, it is also useful to assess the impact of our new phase equilibrium and calorimetric data on other, more broadly based databases. In particular, it would be useful to know whether the need to revise the entropy and heat capacity for titanite depends only on our choice of the QUIIF thermodynamic database and solution models (Andersen et al. 1993).

Our new entropy values for titanite ($A2/a$ space group) are specific to the QUIIF database and must not be exported directly to other databases, but we can assess whether those databases require revision. As already noted, neither the database of Berman (1991) nor that of Holland and Powell (1990) predicts the location of reaction 1 accurately, giving 1 bar values of 1000 °C and -62 °C, respectively. Clearly, then, data for at least one of the phases hedenbergite, ilmenite, titanite, and fayalite in each of those databases requires revision. However, identifying the problem phase(s) is not straightforward. Revision of data for titanite alone is unlikely to resolve the discrepancy. Both databases reproduce the experimental data of Jacobs and Kerrick (1981), but only the Berman database also predicts the data of Manning and Bohlen (1991). This strongly suggests that the values for titanite in both these databases are consistent with the phases in the Jacobs and Kerrick (1981) experiments (i.e., rutile, quartz, calcite, CO_2 - H_2O -NaCl mixtures). Therefore, the inability of both databases to predict reaction 1, and of Holland and Powell (1990) to predict the Manning and Bohlen (1991) experiments, suggest inconsistencies among the values for titanite, fayalite, hedenbergite, ilmenite, anorthite, and kyanite in these databases.

In the following discussion we concentrate on reaction

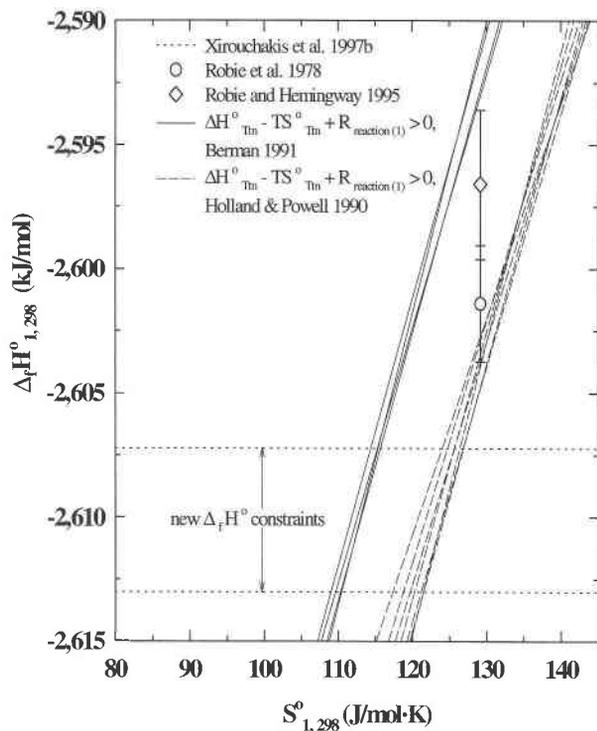


FIGURE 8. Feasible solution for $\Delta_f H^\circ$ and S° of titanite as defined by calorimetry and phase-equilibrium constraints based only on the reaction $Ttn + Fa \rightarrow Hd_{ss} + Il_{ss}$. The lines were constructed using the midpoints for each experiment, average mineral compositions, ideal site mixing for Hd_{ss} and Il_{ss} solid solutions, and thermodynamic data for all phases from the Berman (1991) (straight lines) and Holland and Powell (1990) (dashed lines). R represents the contribution of all the input parameters but $\Delta_f H^\circ$ and S° in Equation 7.

1 because the phases involved are more nearly pure, and thus there is less need to invoke activity models than would be required for reaction 2. We performed calculations using the experimental results for reaction 1 and the Berman (1991) and Holland and Powell (1990) thermodynamic databases. The activities of hedenbergite_{ss} and ilmenite_{ss} were approximated using ideal site mixing models. We used the heat capacities given by Berman (1991) and Holland and Powell (1990), and thermal expansivity and compressibility terms for titanite based on the data of Taylor and Brown (1976), Kunz et al. (1996), and Angel et al. (unpublished manuscript). The results of these calculations are shown graphically in Figure 8.

Regardless of the database used or the choice of heat capacity equation, the derived entropy values are smaller than the currently accepted entropy of 129.20 ± 0.84 J/(mol·K). Most importantly, the calculations suggest that the proposed change in entropy is largely independent of the thermodynamic database but it is required by the adopted enthalpy of formation (cf. Fig. 7 and 8). An enthalpy of formation for titanite of -2610.13 ± 2.90 kJ/mol (Xirouchakis et al. 1997b) is more consistent with entropy values either smaller than 110 J/(mol·K) (Berman

1991) or 120 J/(mol·K) (Holland and Powell 1990) (Fig. 8). In contrast, an entropy value at about 129.20 J/(mol·K) (King et al. 1954) appears to be consistent with the older enthalpy values either less negative than -2597 kJ/mol (Holland and Powell 1990; Berman 1991; Robie and Hemingway 1995; Xirouchakis and Lindsley 1995) or -2602 kJ/mol (Robie et al. 1978; Holland and Powell 1985) (Fig. 8). It would be presumptuous and highly inappropriate for us to recommend specific revisions to the databases of Berman (1991) and of Holland and Powell (1990). The values we have calculated for titanite and present graphically using those databases are for comparison only and must not be viewed as "corrections" to them. Clearly, new measurements on the heat capacity of titanite, using well-characterized material that is stoichiometric and pure, are urgently needed to test this conclusion.

This study clearly supports Wones (1989) argument that the assemblage hedenbergite_{ss}-ilmenite_{ss} indicates relatively reducing conditions compared to titanite-magnetite_{ss}-quartz. However, our results demonstrate quantitatively that the stable equilibrium positions of reaction 1 and 2 reflect changes in pressure, temperature, f_{O_2} , a_{SiO_2} , $\mu_{Fe^{2+}Ti^{4+}Fe^{3+}_2}$, and $\mu_{CaFe_{-1}}$. The differences in temperature and oxygen fugacity between the equilibrium position of the reactions as calculated for pure phases vs. solid solutions with our data can be significant in the range 650 to 1000 °C (i.e., 50–60 °C for reaction 1, and 0.1 to 1 log f_{O_2} units for reaction 2). Furthermore, the stable assemblage $Ttn-Mt_{ss}-Qz$ is not necessarily indicative of oxygen fugacity conditions higher than that of the FMQ buffer. The experimental and calculated phase relations indicate that reaction 2 is not constrained to lie either below or above the FMQ buffer curve. We also estimate that at 1 bar, the stable invariant point I should lie at 612 °C and -0.19 log unit below FMQ (Fig. 2); in contrast, the metastable invariant point should lie at 675 °C and on the FMQ buffer. The stable invariant point I is displaced to lower temperatures with increasing pressure but also it remains below the FMQ buffer reaction curve (Fig. 2). Programs to calculate titanite-bearing oxide-silicate equilibria will be made available upon request to the second author.

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