# The crystal structure of julgoldite 

RUDOLF ALLMANN<br>Fachbereich Geowissenschaften, Philipps-Universität, 355 Marburg, Germany<br>and GABRIELLE DONNAY<br>Department of Geological Sciences, McGill University, Montreal IoI, Quebec, Canada

sUmmary. The structure of julgoldite, $\mathrm{Ca}_{2}\left(\mathrm{Fe}_{1-x}^{2+} \mathrm{Fe}_{x}^{3+}\right) \mathrm{Fe}_{2}^{3+} \mathrm{Si}_{3} \mathrm{O}_{10+x}(\mathrm{OH})_{4-x}(x \approx 0 \cdot 5)$, has been determined and refined to an $R$ value of $4.6 \%$ based on 2417 symmetry-independent reflections, of which 765 are unobserved (weighted $R=4.3 \%$ ). The cell dimensions are $a 8.922(4), b 6.081(3)$, c $19.432(9) \AA, \beta 97.60(6)^{\circ}$, the space group is $A 2 / m$ with $Z=4$ and $D_{\text {cale }} 3.56 \mathrm{~g} \mathrm{~cm}^{-3}$. Julgoldite is isostructural with pumpellyite; the $\mathrm{Si}-\mathrm{O}$ and $\mathrm{Ca}-\mathrm{O}$ distances of julgoldite are identical, within experimental accuracy, with corresponding distances in pumpellyite; the $\mathrm{Fe}-\mathrm{O}$ distances of julgoldite are about $0.09 \AA$ longer than the corresponding ( $\mathrm{Al}, \mathrm{Mg}$ ) - O distances in pumpetlyite. The four hydroxyl groups and the extent of O -for-OH substitution are identified by valence-sum calculations.

Pumpellyite, for which the formula $\mathrm{Ca}_{2} \mathrm{Al}_{2}(\mathrm{OH})_{2}\left(\mathrm{Al}_{0.5} \mathrm{Mg}_{0.5}\right)(\mathrm{OH}, \mathrm{O}) \mathrm{SiO}_{4} \cdot \mathrm{Si}_{2} \mathrm{O}_{6}$ $(\mathrm{OH}, \mathrm{O})$ has recently been proposed (Allmann and Donnay, 1971), has become a mineral of special interest because of its role in low-grade metamorphism (see, for example, Coombs, 196I). In the absence of good-quality single crystals, Galli and Alberti (1969) were unable to obtain the high-quality intensity data that are a prerequisite for structure refinement. The 793 visually estimated intensities, 205 of which were unobserved, led to $R=15.9 \%$, and the 588 observed reflections to $R=\mathrm{I} 2.3 \%$. Refining the pumpellyite structure type became possible when Moore (1971) described an iron analogue of pumpellyite, the new mineral julgoldite, $\mathrm{Ca}_{2}\left(\mathrm{Fe}_{1}^{2+}{ }_{-x} \mathrm{Fe}_{x}^{3+}\right) \mathrm{Fe}_{2}^{3+}$ $\mathrm{Si}_{3} \mathrm{O}_{10+x}(\mathrm{OH})_{4-x}$, from Långban, Sweden. Professor Moore kindly gave us crystals on which to work. Julgoldite crystals are of good quality and thus afford an accurate structure determination.

Crystal description and crystal data. A small portion $(\approx 0.3 \mathrm{~g})$ of a specimen from Långban, Sweden, contained several shiny black plates of julgoldite embedded in clear apophyllite. In addition we noted a small aggregate of tiny dull-black crystals; they were at first mistaken for julgoldite, but, on the basis of a microprobe analysis (see below), they are probably ilvaite, $\mathrm{CaFe}_{2}^{2+} \mathrm{Fe}^{3+}(\mathrm{OH}) \mathrm{OSi}_{2} \mathrm{O}_{7}$.

A crystal was first mounted along $b$ and used for the measurement of lattice constants and intensities $h \mathrm{ol}-h 9 l$ : it is a ( I 00 ) plate, in the shape of an isosceles trapezium, which has two edges parallel to $b$, with lengths 0.44 and 0.23 mm ; the height along $c$ is 0.29 mm and the thickness along $a^{*}$ is 0.06 mm . The same plate was then mounted parallel to the $a$ axis to measure the okl reflections. The cell dimensions, measured with Mo-K $\alpha_{1}$ radiation ( $\lambda 0.70926 \AA$ ) have the following values at $20^{\circ} \mathrm{C}: a 8.922(4)$, $b 6.08 \mathrm{I}(3), c \mathrm{I} 9.432(9) \AA, \beta 97 \cdot 60(6)^{\circ}, \mathrm{V}=1044 \AA^{3}$, with space $\operatorname{group} A 2 / m, Z=4$. (c) Copyright the Mineralogical Society.
$D_{\text {calc }} 3.56 \mathrm{~g} \mathrm{~cm}^{-3}$. We had too little material to remeasure the density, for which Moore (1971) reported the value $D_{\text {obs }} 3.602 \mathrm{~g} \mathrm{~cm}^{-3}$. His sample may have contained admixed ilvaite, whose reported density ranges from 3.8 to $4 \cdot I \mathrm{~g} . \mathrm{cm}^{-3}$ : this would account for the high measured value.

Crystal chemistry. A few hand-picked crystals of julgoldite were submitted to Dr. K. Abraham (Ruhr-Universität, Bochum) for a microprobe analysis on his Microscan V instrument run at $20 \mathrm{KeV}, 30 \mathrm{nA}$. His results (table I, columns $c$ and $d$ ) show reasonably good agreement with the theoretical composition given above for julgoldite (table I, a), except that about 5 atomic $\%$ of $\mathrm{Fe}^{3+}$ is replaced by Al

Table I. Electron-microprobe analyses of julgoldite and associated mineral from Långban, Sweden

|  | $a$ | $b$ | $c$ | $d$ | $e$ | $f$ | $g$ | $h$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{SiO}_{2}$ | $32 \cdot 2$ | 32.4 | $32 \cdot 4$ | 31.0-32.0 | $34^{\circ}$ | $34^{\circ}$ | 26.8 | $29 \cdot 3$ |
| $\mathrm{TiO}_{2}$ | - | - | - | - | $0 \cdot 1$ | - | - | - |
| $\mathrm{Al}_{2} \mathrm{O}_{3}$ | - | I•I | 1.0 | I. $0-1.9$ | I'3 | 1.0 | - | - |
| $\mathrm{Fe}_{2} \mathrm{O}_{3}$ | $35 \cdot 7$ | $34 \cdot 1$ | - | - | 29.6 | 31.4 | - | 19.6 |
| FeO | 6.4 | $6 \cdot 5$ | - | - | $8 \cdot 7$ | $6 \cdot 0$ | - | $35 \cdot 2$ |
| $\Sigma \mathrm{Fe}$ as $\mathrm{Fe}_{2} \mathrm{O}_{3}$ | (42.8) | (41.3) | $40 \cdot 6$ | 39.9-41-4 | (39.3) | (38. I) | $59 \cdot 7$ | (58.7) |
| CaO | $20 \cdot 1$ | $20 \cdot 2$ | $19 \cdot 3$ | 19.3-20.0 | 22.0 | $20 \cdot 6$ | 13.6 | 13.7 |
| MnO | - | - | , | - | 0.2 | - | 0.2 | - |
| MgO | - | - | - | - | 0.2 | - | - | - |
| BaO | - | - | - | - | 0.01 | - | - | - |
| $\mathrm{H}_{2} \mathrm{O}$ | 5.6 | $5 \cdot 7$ | not detd. | not detd. | 4.69 | $6 \cdot 5$ | not detd. | $2 \cdot 2$ |
| Total | $100 \cdot 0$ | $100 \cdot 0$ | 93.3* | 9I $\cdot 8-93 \cdot{ }^{*}$ | $100 \cdot 7$ | 100.0才 | 100.3* | $100 \cdot 0$ |

Microprobe analyses by Dr. K. Abraham; emission spectrographic analysis by B. Rajandi; iron oxidation grade and water determination by P. Moore (Moore, 1971).
a. Theoretical composition for $\mathrm{Ca}_{2} \mathrm{Fe}_{2.5}^{3+} \mathrm{Fe}_{0.5}^{2+} \mathrm{Si}_{3} \mathrm{O}_{10.5}(\mathrm{OH})_{3.5}$.
b. Theoretical composition for $\mathrm{Ca}_{2}\left(\mathrm{Fe}_{0.5}^{2+} \mathrm{Fe}_{0 \cdot 5}^{3+}\right)\left(\mathrm{Fe}_{1 \cdot 875}^{3+} \mathrm{Al}_{0 \cdot 125}\right) \mathrm{Si}_{3} \mathrm{O}_{10 \cdot 5}(\mathrm{OH})_{3 \cdot 5}$.
c. Microprobe analysis of a nearly homogeneous area in one julgoldite specimen.
$d$. Microprobe analyses of numerous points on four different crystals.
e. Emission spectrographic analysis from Moore, 1971.
$f$. Theoretical composition as given under $b$ with the addition of $8 \mathrm{wt} \%$ apophyllite.
$g$. Microprobe analysis of one crystal of associated mineral. Sum too high, because all Fe is taken to be $\mathrm{Fe}^{3+}$.
$h$. Theoretical composition of ilvaite, $\mathrm{CaFe}_{2}^{2+} \mathrm{Fe}^{3+}(\mathrm{OH}) \mathrm{OSi}_{2} \mathrm{O}_{7}$.

* $\mathrm{H}_{2} \mathrm{O}$ not determined. $\quad+$ Including $0.5 \% \mathrm{KF}$.
(Table I, b). All crystals were observed to be somewhat inhomogeneous, especially in Al content, which ranged from o to $\mathrm{I} \cdot 5 \mathrm{wt} \%$; Ca was nearly constant; Fe and Si showed small variations with negative correlation; $\mathrm{Mn}, \mathrm{Mg}, \mathrm{Ti}, \mathrm{K}, \mathrm{Na}, \mathrm{Cr}, \mathrm{Ni}$, and Zn were looked for, but were not detected.

The observed range of the Al content is in line with a composition intermediate between julgoldite and pumpellyite, which is cited by Moore (1971); the three Fe atoms per formula unit of stoichiometric julgoldite are here replaced by
$\left(\mathrm{Fe}_{1 \cdot 17}^{3+} \mathrm{Al}_{1 \cdot 03} \mathrm{Fe}_{0.66}^{2+} \square_{0 \cdot 03}\right)$. The possibility of a complete solid solution, with $\mathrm{Fe}_{2 \cdot 5}^{3+5} \mathrm{Fe}_{0.5}^{2+}$ and $\mathrm{Al}_{2.5} \mathrm{Mg}_{0.5}$ representing the end-member contents in the two variable cation positions was already mentioned by Moore.

The emission spectrographic analysis of julgoldite performed by Benita Rajandi (Moore, 1971) is high in $\mathrm{SiO}_{2}$ and CaO (table $\mathrm{I}, e$ ). This fact could easily be due to a slight contamination of the sample with apophyllite, in which the julgoldite crystals are embedded: $8 \mathrm{wt} \%$ of admixed apophyllite would account for the reported results (table $\mathbf{I}, f$ ).

The oxidation state of the iron as determined by P. Moore (table I, e) is high in $\mathrm{Fe}^{2+}$ when compared with the rather low $\mathrm{H}_{2} \mathrm{O}$ content. We fixed the occupancy of $\mathrm{Fe}(2)$ at $\left(\mathrm{Fe}_{0.5}^{2+}+\mathrm{Fe}_{0.5}^{3+}\right)$ by crystallochemical considerations (mainly on the basis of the differences in distances $\mathrm{Fe}(\mathrm{I})-\mathrm{O}$ and $\mathrm{Fe}(2)-\mathrm{O}$, see table V ). But the $\mathrm{Fe}^{2+}$ content may well vary somewhat depending on the $\mathrm{O}_{2}$ partial pressure at the time of formation. The excess $\mathrm{Fe}^{2+}$ from Moore's analysis could be accounted for by associated ilvaite, whose structure is very similar to that of julgoldite. Crystals from the dull-black aggregate that is associated with julgoldite were also studied by Dr. Abraham (table I, $g$ ). They fit the description of one of the two unidentified phases described by Moore (1971). They contain no Al , but $0.2 \mathrm{wt} \% \mathrm{MnO}$. The agreement with the theoretical composition of ilvaite (table I, $h$ ) is quite acceptable.

Structure determination. Intensities were collected on an automated two-circle goniometer with Zr -filtered Mo-K $\alpha$ radiation out to a maximum $(\sin \theta) / \lambda$ value of $0 \cdot 8 \AA^{-1}$. The linear absorption coefficient is $55.8 \mathrm{~cm}^{-1}$, so that absorption corrections could be considered negligible for the above crystal. As can be seen in table III, the final $B_{11}$ values are on the average about $0.26 \AA^{2}$ larger and the $B_{33}$ values about $0.27 \AA^{2}$ smaller than the average $B$ values, mainly due to the missing absorption correction of the platy crystal set along $b$. The oko reflections were taken from the zero layer of the crystal, set along $a$. Of the 2417 symmetry-independent reflections that were measured, ${ }^{\text {I }} 765$ were too weak to be observed and were given an $F_{\text {obs }}$ value of zero. Intensities were corrected for Lorentz and polarization factors.

The layers with $k$ odd ( $\approx 40 \%$ unobserved reflections) are much weaker than the layers with $k$ even ( $\approx 24 \%$ unobserved reflections). As the Fe atoms are contributing little to the weak layers, the intensities of these are very similar to those of pumpellyite (Galli and Alberti, 1969), since all atoms are the same in julgoldite as in pumpellyite, except for Fe , which takes the place of Al and ( $\mathrm{Al}, \mathrm{Mg}$ ).

Starting with the atomic parameters of pumpellyite ( $R=8.6 \%$ ), the refinement quickly dropped to a conventional residual of $R=6.2 \%$ with isotropic temperature factors. The atomic form factors used for $\mathrm{Fe}^{3+}, \mathrm{Ca}^{2+}$, and $\mathrm{O}^{-}$were those of Hanson and Pohler (1966), and the one for SiO comes from Hanson et al. (1964). Then Cromer's (1965) correction terms for anomalous dispersion were applied to the form factors of the cations. The temperature factors did change somewhat (see Table II), but the $R$ value did not. After introduction of anisotropic temperature factors and anomalous

[^0]dispersion correction as above, $R$ further decreased to $4.6 \%$. The corresponding weighted $R$ value is $4.3 \%$. In the last cycle all but five of the 765 unobserved reflections had $F_{\text {cale }}$ values smaller than the assumed $F_{\min }$ value of 22 . For all unobserved reflections with $F_{\text {calc }}<F_{\min }$, the difference $F_{\text {calc }}-F_{\text {obs }}$ was set equal to zero. Thus the unobserved reflections, except the five mentioned above, do not contribute to the $R$ values, but their derivatives are included in the calculation of the parameter shifts. $\sigma\left(F_{\text {obs }}\right)$ was taken as $\left(0.67 F_{\min }^{2}+0.03 F_{\text {obs }}^{2}\right) / F_{\text {obs }}$ for the observed reflections and as $F_{\min }$ for the unobserved ones. ${ }^{1}$

Table II: Julgoldite. Final fractional coordinates with standard deviations (at $R=4.6 \%$ ) and two sets of isotropic temperature factors (at $R=6.2 \%$ ). $\bar{\sigma}$ is the absolute standard deviation averaged over the two or three varied coordinates

| Atom | $x$ | $y$ | $z$ | $\bar{\sigma}$ | B* | $\mathbf{B} \dagger$ | Point position |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ca}(\mathrm{I})$ | 0.2554(2) | $\frac{1}{2}$ | $0 \cdot 3397$ (r) | 2. $10^{-3} \mathrm{~A}$ | $0.43 \AA^{2}$ | $0.47 \AA^{2}$ | 4(i) |
| $\mathrm{Ca}(2)$ | $0 \cdot 1963$ (3) | $\frac{1}{2}$ | -.1563(I) | 2 | 0.74 | $0 \cdot 78$ | 4(i) |
| $\mathrm{Fe}(\mathrm{I})$ | $\frac{1}{2}$ | $\frac{1}{4}$ | $\frac{1}{4}$ | - | 0.67 | 0.75 | $4(\mathrm{f})$ |
| $\mathrm{Fe}(2)$ | 0.2543 (I) | $0.2465(2)$ | 0.4955(5) | I | 0.48 | 0.55 | 8(j) |
| $\mathrm{Si}(\mathrm{I})$ | $0.0506(3)$ | 0 | 0.0936 (1) | 3 | $0 \cdot 33$ | 0.34 | 4(i) |
| Si(2) | $0.1623(3)$ | $\bigcirc$ | 0.2478 (1) | 3 | $0 \cdot 4 \mathrm{I}$ | 0.42 | 4(i) |
| Si(3) | $0.4650(3)$ | $\bigcirc$ | 0.4013 (1) | 3 | $0 \cdot 30$ | $0 \cdot 30$ | 4(i) |
| $\mathrm{O}(\mathrm{I})$ | $0 \cdot 1353(6)$ | $0.2214(9)$ | $0.0755(2)$ | 5 | $0 \cdot 76$ | $0 \cdot 72$ | 8(j) |
| $\mathrm{O}(2)$ | $0.2582(5)$ | $0 \cdot 2292$ (9) | 0.2473 (2) | 5 | $0 \cdot 72$ | 0.69 | 8 (j) |
| O(3) | $0.3674(6)$ | 0.2192(9) | $0.4135(2)$ | 5 | 0.86 | 0.8 I | 8(j) |
| $\mathrm{O}(4)$ | - $\cdot 1283$ (8) | $\frac{1}{2}$ | $0.4428(4)$ | 7 | 0.52 | 0.47 | 4(i) |
| $\mathrm{O}(5) \mathrm{H}$ | -.1234(9) | 0 | $0.4556(4)$ | 8 | 1.06 | 1.01 | 4(i) |
| $\mathrm{O}(6)$ | $0.3717(8)$ | $\frac{1}{2}$ | $0.0453(4)$ | 7 | 0.55 | 0.52 | 4(i) |
| $\mathrm{O}(7) \mathrm{H}$ | $0.3757(9)$ | 0 | $0.0332(4)$ | 8 | 1-07 | 1.02 | 4(i) |
| O(8) | $0.0315(9)$ | 0 | -0.1780(4) | 7 | $0 \cdot 79$ | 0.75 | 4(i) |
| O(9) | $0.4728(8)$ | $\frac{1}{2}$ | $0.1746(4)$ | 7 | 0.56 | 0.52 | 4(i) |
| $\mathrm{O}(\mathrm{IO}) \mathrm{H}$ | $0.0615(9)$ | $\bigcirc$ | $0.3128(4)$ | 8 | I. 04 | 0.98 | 4(i) |
| $\mathrm{O}(\mathrm{II}) \mathrm{H}$ | $0.4974(9)$ | - | $0 \cdot 1835(4)$ | 8 | 0.93 | 0.88 | 4(i) |

* Without anomalous dispersion correction.
$\dagger$ With anomalous dispersion correction.

The final atomic coordinates and isotropic temperature factors are listed in table II; the anisotropic temperature factors are shown in table III. The final coordinates and our cell dimensions were used by Mrs. E. Evans, of the National Bureau of Standards, Washington, D.C., to calculate spacings and intensities of the theoretical julgoldite powder pattern for comparison with the experimental pattern of Moore (table IV).

Discussion of the structure. The julgoldite structure is best viewed down its $b$ axis, projected onto the $a c$ plane (fig. I). The slight differences between it and the pumpellyite structure are due to the different sizes of the octahedral cations. Bond lengths

[^1]and bond valences, to be discussed below, are summarized in table V; table VI gives the lengths of the edges of all coordination polyhedra and the angles they subtend at their central cation, as well as the cation-cation distances less than $3.3 \AA$.

The octahedra form chains along $b$ by sharing opposite edges. They are of two kinds: $\mathrm{Fe}(\mathrm{I})$, which makes up one-third of the total iron content, is characterized by an average $\mathrm{Fe}-\mathrm{O}$ distance $\breve{L}=2.084 \AA$. If we assume a linear relation between

Table III. Julgoldite. Anisotropic temperature factors (from the expression $\left.\exp \left(\mathrm{h}^{2} \mathrm{~B}_{11} \mathrm{a}^{* 2} / 4+\ldots+2 \mathrm{hkB}_{12} \mathrm{a}^{*} \mathrm{~b}^{*} / 4+\ldots\right)\right) . \bar{\sigma}(\mathrm{B})$ is the standard deviation averaged over all six or four varied $B_{\mathrm{ik}}$

| Atom | $\mathrm{B}_{11}$ | $\mathrm{B}_{22}$ | $\mathrm{B}_{33}$ | $\mathrm{B}_{12}$ | $\mathrm{B}_{13}$ | $\mathrm{B}_{23}$ | $\bar{\sigma}(\mathrm{B})$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ca}(\mathrm{I})$ | 0.64 | 0.64 | $0 \cdot 30$ | 0 | 0.08 | 0 | $0.05 \AA^{2}$ |
| $\mathrm{Ca}(2)$ | $1 \cdot 31$ | 0.66 | 0.44 | $\bigcirc$ | 0.00 | 0 | 0.06 |
| Fe (I) | 1.07 | $0 \cdot 74$ | 0.52 | 0.03 | $0 \cdot 12$ | 0.05 | 0.04 |
| $\mathrm{Fe}(2)$ | 0.80 | 0.63 | $0 \cdot 35$ | 0.04 | $0 \cdot 10$ | $0 \cdot 04$ | 0.03 |
| $\mathrm{Si}(1)$ | 0.46 | 0.49 | $0 \cdot 16$ | - | -0.06 | $\bigcirc$ | 0.07 |
| Si(2) | $0 \cdot 66$ | 0.39 | - 19 | 0 | -0.06 | 0 | 0.07 |
| Si(3) | 0.49 | $0 \cdot 49$ | $0 \cdot 14$ | 0 | 0.07 | $\bigcirc$ | 0.07 |
| $\mathrm{O}(\mathrm{I})$ | $1 \cdot 14$ | 0.66 | 0.46 | -0.19 | 0.08 | -0.16 | 0.15 |
| $\mathrm{O}(2)$ | 0.99 | 0.58 | $0 \cdot 40$ | 0.06 | -0.05 | -0.11 | 0.14 |
| O(3) | I. 88 | 0.84 | 0.51 | $0 \cdot 16$ | $0 \cdot 25$ | $0 \cdot 14$ | $0 \cdot 15$ |
| O(4) | 0.64 | 0.53 | 0.3 I | 0 | 0.08 | - | - 19 |
| $\mathrm{O}(5) \mathrm{H}$ | I 24 | I. 04 | 0.64 | 0 | 0.03 | 0 | 0.24 |
| O(6) | 0.94 | $0 \cdot 38$ | 0.27 | $\bigcirc$ | 0.05 | 0 | 0.20 |
| $\mathrm{O}(7) \mathrm{H}$ | 1.26 | $\mathrm{o} \cdot 81$ | 1.04 | $\bigcirc$ | $0 \cdot 24$ | $\bigcirc$ | 0.24 |
| $\mathrm{O}(8)$ | 0.96 | 1.07 | $0 \cdot 29$ | $\bigcirc$ | -0.06 | 0 | $0 \cdot 22$ |
| $\mathrm{O}(9)$ | $0 \cdot 80$ | 0.73 | 0.29 | 0 | $0 \cdot 14$ | - | $0 \cdot 20$ |
| $\mathrm{O}(\mathrm{I}) \mathrm{H}$ | 1.32 | I. 20 | 0.71 | 0 | 0.20 | 0 | 0.24 |
| $\mathrm{O}(\mathrm{It}) \mathrm{H}$ | I. 24 | 0.87 | $0 \cdot 56$ | $\bigcirc$ | 0.23 | 0 | 0.23 |

the $\mathrm{Fe}^{2+} /\left(\mathrm{Fe}^{2+}+\mathrm{Fe}^{3+}\right)$ ratio and the mean octahedral bond length, $L$ ( $\mathrm{Fe}-\mathrm{O}$ ) (as we do, for example, when deriving the Al occupancy in ( $\mathrm{Si}, \mathrm{Al}$ ) tetrahedra) we may predict the Fe ( I ) occupancy thus: from the effective ionic radii of Shannon and Prewitt (1969) $\left[r\left({ }^{\left({ }^{\mathrm{i}}\right.} \mathrm{Fe}^{2+}\right)=0.77 \AA, r\left({ }^{\mathrm{Vi}} \mathrm{Fe}^{3+}\right)=0.645 \AA\right.$ and $\left.r\left({ }^{\left({ }^{\mathrm{T}}\right.} \mathrm{O}^{2-}\right)=\mathrm{I} .38 \AA\right]$, we calculate the expected values for pure $\mathrm{Fe}^{2+}$ and pure $\mathrm{Fe}^{3+}$ occupancy as $\bar{L}\left(\mathrm{Fe}^{2+}-\mathrm{O}\right)=2 \cdot 15 \AA$ and $\tilde{L}\left(\mathrm{Fe}^{3+-}-\mathrm{O}\right)=2.025 \AA$; for the measured value $\bar{L}=2.084 \AA$, we obtain, by linear interpolation, an $\mathrm{Fe}(\mathrm{I})$ occupancy of $\mathrm{Fe}_{0.52}^{3+} \mathrm{Fe}_{0 \cdot 48}^{2+}$.

For $\mathrm{Fe}(2)$, which occupies two-thirds of the octahedra, the experimental $\bar{L}$ value is $2.011 \AA$. Even if the three-fold coordination of $\mathrm{O}(1)$ and $\mathrm{O}(3)$, which are bonded to $\mathrm{Fe}(2)$, is considered, the expected $\bar{L}$ value for $\mathrm{Fe}(2)$ is $2.018 \AA\left[r\left({ }^{\text {(ii }} \mathrm{O}^{2-}\right)=\mathrm{I} \cdot 36 \AA\right]$. Therefore $\mathrm{Fe}(2)$ should be the site of Al substitution $\left[r\left({ }^{\mathrm{Ti}} \mathrm{Al}\right)=0.53 \AA\right]$. A replacement of $6 \%$ of $\mathrm{Fe}^{3+}$ by Al, indicated by the microprobe analysis (Table $\mathrm{I}, c$ and $d$ ), reduces the calculated $\bar{L}$ exactly to 2 -OII $\AA$.

The octahedral chains are joined in the ac plane by $\mathrm{SiO}_{4}$ and $\mathrm{Si}_{2} \mathrm{O}_{6}(\mathrm{OH})$ groups, forming 5 -membered rings of two octahedra and three tetrahedra. Half of the rings are open and have a $\mathrm{Ca}(\mathrm{I})$ ion at their centre; the other rings are closed and

Table IV. Observed and calculated powder patterns of julgoldite

| $I_{\text {obs }}{ }^{*}$ | $I_{\text {calc }}$ | $d_{\text {obs }}$ | $d_{\text {cale }}$ | $h k l$ | $I_{\text {obs }}$ | $I_{\text {calc }}$ | $d_{\text {obs }}$ | $d_{\text {calc }}$ | $h k l$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 2 | 16 | $9.57(8) \AA$ | $9.631 \AA 002$ |  | 4 | (4) | I.732(2) | (1.741 | 415 |
| 2 | 18 | 8.82(7) | 8.844 | 100 |  | \{21 |  | I.738 | $42 \overline{4}$ |
| - | 7 | - | 6.991 | $10 \overline{2}$ |  | (6) |  | 1.737 | 135 |
| 1 | 13 | 6.14(3) | $6 \cdot 123$ | 102 | 1 | $\{4$, | 1.715(2) | $\{1.720$ | 3.0.1̄ō |
| - | 9 | - | 4.940 | 11 I |  | ( 6 \} |  | 11.718 | 406 |
| 7 | $\left\{\begin{array}{l}58 \\ 38\end{array}\right\}$ | 4.80(2) | $\left\{\begin{array}{l}4.815 \\ 4.763\end{array}\right.$ | 004 | I | $\left\{\begin{array}{l}4 \\ 6\end{array}\right\}$ | 1-694(2) | $\left\{\begin{array}{l}1.696 \\ 1.693\end{array}\right.$ | $513$ |
| - | I1 | - | 4.763 4.486 | $110 \overline{4}$ | 2 | 10 | 1-673(2) | 1.693 1.675 | 1.1.1 |
| I | 18 | $4 \cdot 43$ (2) | $4 \cdot 422$ | 200 |  | - 4 \} | 1-622(2) | \{ 1.627 | 0.2.10 |
| - | 6 |  | 4.237 | 202 | $5 \frac{1}{2}$ | 154 |  | 11.621 | 424 |
| - | 6 | - | 4.102 | $11 \overline{3}$ | 1 | 19 | I.605(2) | I. 605 | 0.0.12 |
| - | 4 | - | 4.012 | 104 | 3 | 27 | I•585(2) | I. 586 | 2.2.1̄o |
| 8 | 83 | 3.84(1) | 3.831 | 202 |  | 5 | - | 1-569 | 1.2.10 |
| - | 15 |  | 3.495 | $20 \overline{4}$ | - | 11 | - | 1.531 | 408 |
| - | 4 | - | 3.451 | 211 | 52 | \{ 34 \} | 1.517(2) | (1.520 | 040 |
| I | 16 | 3.059(8) | 3.061 | 204 |  | (29) |  | (1.515 | $42 \overline{8}$ |
| - | 4 | 3059(8) | 2.991 | 213 | - | 5 | - | r.508 | $52 \overline{4}$ |
|  | (83) |  | (2.949 | 115 | 2 | 17 | 1-472(2) | I-475 | 2.2.10 |
| 10 | $\left\{\begin{array}{l}17 \\ 20\end{array}\right\}$ | 2•950(7) | $\{2.948$ | 300 | 1 | $\left\{\begin{array}{l}6 \\ 5\end{array}\right\}$ | I.429(1) | $\left\{\begin{array}{l}1.429 \\ 1.428\end{array}\right.$ | $602$ |
| - | $\left(\begin{array}{c}10 \\ 16\end{array}\right.$ | - | 2.929 2.899 | 302 022 | I | $\begin{array}{r}5 \\ \hline\end{array}$ |  | 1.428 1.419 | 1.2.12 0.2 .12 |
| - | 16 | - | 2.875 | 120 |  | , 6 | 1.416(1) | 1.413 | 242 |
|  |  |  | 12.788 | $12 \overline{2}$ | 2 | $(4)$ |  | 11.412 | 606 |
| 6 | $\{68$ \} | 2.778(7) | $\left\{\begin{array}{l}2.788 \\ 2.778\end{array}\right.$ | $20 \overline{6}$ |  | (4) | I•390(1) | 1.389 r. r | 4.0.17 $\overline{\mathrm{I}}$ |
| 3 | $\left\{\begin{array}{r}13 \\ 6\end{array}\right\}$ | 2.717(6) | $\left\{\begin{array}{l}2.723 \\ 2.720\end{array}\right.$ | 122 | I | $\left(\begin{array}{l}4 \\ 2\end{array}\right\}$ | 1-366(I) | $\left(\begin{array}{l}1 \cdot 367 \\ \times \cdot 365\end{array}\right.$ | 428 |
|  |  |  | $12 \cdot 720$ | 302 |  | $\left\{\begin{array}{l}2 \\ 2\end{array}\right\}$ |  | 1.365 |  |
| 4 | 34 | 2.680 (6) | 2.671 | $31 \overline{1}$ |  | $\binom{2}{2}$ |  | 1.363 | $52 \overline{8}$ |
| 7 | 8 I | 2.568(6) | 2.571 | 024 |  |  |  | [ $1 \cdot 353$ | 435 |
|  | 9 <br> 19 | - | 2.517 2. 2.505 | $12 \overline{4}$ |  | \{ 3 \} | - | $\left\{\begin{array}{l}1.351 \\ 1.351\end{array}\right.$ | 435 |
| 4 | $\left\{\begin{array}{l}19 \\ 24\end{array}\right\}$ | $2 \cdot 491(5)$ | $\left\{\begin{array}{l}2.505 \\ 2.492\end{array}\right.$ | 220 |  | $\left(\begin{array}{l}3 \\ \hline\end{array}\right.$ |  | $1 \cdot 349$ | $34 \overline{2}$ |
| 4 | (24) |  | 12.492 | 117 | $2 \frac{1}{2}$ | $\binom{12}{14}$ | 1-322(I) | (1.336 | $62 \overline{2}$ |
| - | 21 | - | 2.470 | $22 \overline{2}$ |  | I4 |  | I 334 | 246 |
| 1 $\frac{1}{2}$ | $\left\{\begin{array}{l}6 \\ 3\end{array}\right\}$ | 2.433(5) | $\{2.448$ | 206 |  | [4 |  | 1.332 | $53 \overline{3}$ |
|  | $3)$ | 2433 (5) | 12.423 | 124 |  | (3) |  | 1.330 | 1.3.īI |
| - | \{ 5 $\}$ |  | $\{2 \cdot 408$ | 008 | - | 9 | - | 1.320 | $62 \overline{4}$ |
|  | ( 5) |  | ( $2 \cdot 405$ | $10 \overline{8}$ |  | $\binom{3}{1}$ |  | (1.281 | 626 |
| 5 | 52 | $2 \cdot 385(5)$ | $2 \cdot 382$ | 222 | I | I | I.281(1) | 1.280 | $43 \overline{9}$ |
| - | 12 | - | 2.354 | 313 |  | ${ }^{1}$ |  | $1 \cdot 280$ | 533 |
| - | 6 | -- | $2 \cdot 317$ | 315 |  | (3) |  | 1.277 | 606 |
| 3 | 26 | $2 \cdot 240(4)$ | $2 \cdot 243$ | $20 \overline{8}$ | I | $\{9\}$ | I 266 (I) | [1.267 | 2.0.14 |
|  |  |  | ${ }^{2} 220$ | $40 \overline{2}$ |  | (2) |  | 1. $1 \cdot 267$ | I.I. $\overline{1} \overline{5}$ |
| 1 | \{15 | 2.209(4) | $\{2.211$ | 400 | I $\frac{1}{2}$ | $\left(\begin{array}{c}5 \\ 3\end{array}\right.$ | I-252(I) | ${ }^{1} \cdot 258$ | $24 \overline{8}$ |
|  | $(6)$ |  | 2.208 | 026 |  | 3 |  | $1 \cdot 255$ | 519 |
| - | 5 | - | $2 \cdot 189$ | 126 |  | 5 5 |  | 1.253 | 0.2.14 |
| 2 | 27 | 2.158(4) | 2.157 | 224 |  | - |  | 1.253 | 6.0.1̄0 |
|  |  |  | (2.118 | $40 \overline{4}$ | - | (6) | - | I. 253 | 440 |
| 2 rard | $\left\{\begin{array}{r}16 \\ 5\end{array}\right\}$ | 2.121(4) | $\{2.097$ | 126 |  | 5 |  | 1.235 | $44 \overline{4}$ |
| 1 | 3 | $2 \cdot 034(3)$ | 2.035 | 317 | 二 | 7 | - | I-144 | 155 |
| I | 3 | 1.959(3) | 1.960 | 131 | - | 10 | - | $\left\{\begin{array}{l}1 \cdot 111 \\ 1 \cdot 110\end{array}\right.$ | 0.2.16 |
| I | 7 | 1.948(3) | 1.945 | 406 |  | $\left\{\begin{array}{l}4 \\ 6\end{array}\right\}$ |  |  | 157 |
| 2 | 17 | 19902(3) | 1-907 | 226 | 2 | (5) | I•IO3(1) |  | 804 800 |
|  | ${ }^{29}$ ) |  | (1.888 | 028 |  | $\left(\begin{array}{l}5 \\ 3\end{array}\right.$ |  | $\left\{\begin{array}{l}\mathrm{I} \cdot \mathrm{IO5} \\ \mathrm{I} \cdot \mathrm{I} 05 \\ \mathrm{I} \cdot \mathrm{IO4} \\ \mathrm{I} \cdot \mathrm{I} 04 \\ \mathrm{I} \cdot \mathrm{IO}\end{array}\right.$ | 800 3.1 .15 |
| 3 | \{5 | r.885(3) | 1.886 | $12 \overline{8}$ |  | [ 2 |  |  | ${ }_{72}{ }^{3} \overline{8}$ |
|  | $(4)$ |  | I.873 | 324 |  | 18 |  |  | 0.4.12 |
| - | 5 | - | 1.788 | 420 |  | $(4)$ |  |  | 724 |

surround a $\mathrm{Ca}(2)$ ion. These same columns of 5 -membered rings also form structural building blocks in epidote, zoisite, ardennite, ilvaite, etc.

Bond-valence sums and hydrogen bonds. In order to assign bond valences, $v$, to the experimentally determined interatomic distances, $L$ (table V ), we use the formula discussed above: $\mathrm{Ca}_{2}\left(\mathrm{Fe}_{0.52}^{3+} \mathrm{Fe}_{0.48}^{2+}\right)\left(\mathrm{Fe}_{0.94}^{3+} \mathrm{Al}_{0.06}\right)_{2} \mathrm{Si}_{3} \mathrm{O}_{10 \cdot 52}(\mathrm{OH})_{3 \cdot 48}$. This formula contains $5.7 \mathrm{wt} \% \mathrm{H}_{2} \mathrm{O}$, thus bringing the total for the analysis (table I, c) to $99.0 \%$.


Fig. I. Julgoldite crystal structure projected parallel to [010] onto (oio). The origin is in the lower left-hand corner; a portion of the cell extending to about $\frac{1}{2} a$ and $\frac{1}{2} c$ is shown ( $\quad \cdots-$ hydrogen bonds). An asterisk, *, indicates the site for which co-ordinates are listed in Table II.

The following relations, given in detail in Donnay and Allmann (1970), are used to calculate bond valences: $v=v_{i} \cdot(\bar{L} / L)^{p}$ for $L \leqslant \bar{L}$ and $v=v_{\mathrm{i}}\left(L_{\max }-L\right) /\left(L_{\max }-\bar{L}\right)$ for $\bar{L} \geqslant L \geqslant L_{\mathrm{max}}$. The ideal valence $v_{\mathrm{i}}$, the ratio of formal cation charge to cation coordination number, is thus assigned to the mean bond length $L$ of every given cation polyhedron. In this way the relation of bond length to bond valence is tailored to fit each individual structure. For bonds longer than the mean, the relation used is linear, with the zero value of $v$ assigned to ' $L_{\max }$ ', the sum of the so-called maximum bonding

## NOTE TO TABLE IV

[^2]Table V. Bond lengths $(\AA)$ and bond valences $\left(\nu_{i}\right)$ of julgoldite, $\mathrm{Ca}_{2}\left(\mathrm{Fe}_{0.52}^{3+} \mathrm{Fe}_{0 \cdot 48}^{2+}\right)\left(\mathrm{Fe}_{0 \cdot 94}^{3+} \mathrm{Al}_{0 \cdot 06}\right)_{2} \mathrm{Si}_{3} \mathrm{O}_{10 \cdot 52}(\mathrm{OH})_{3 \cdot 48}$

| Anions | $\mathrm{Ca}(1)$ | $\mathrm{Ca}(2)$ | $\mathrm{Fe}(\mathrm{I})$ | $\mathrm{Fe}(2)$ | Si(1) | Si(2) | Si(3) | $\Sigma_{C} V$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| O(I) |  | $2 \cdot 324$ (6) |  | 2.005(5) | I.605(6) |  |  |  |
|  |  | - $339^{*}$ |  | 0.507 | I.061* |  |  | 1.908 |
| O(2) | $2.439(6)$ | $2 \cdot 427(6)$ | 2.155(5) |  |  | 1.636(6) |  |  |
|  | $0.276^{*}$ | - 296 * | $0.333^{*}$ |  |  | I.021* |  | I.927 |
| O(3) | $2 \cdot 365(6)$ |  |  | 2.002(5) |  |  | 1.626(6) |  |
|  | 0.302* |  |  | 0.511 |  |  | I.033* | I.847 |
| O(4) | $2 \cdot 428(8)$ |  |  | 2.095(7) | 1.657(8) |  |  |  |
|  | $0 \cdot 280$ |  |  | $0.389 \dagger$ | 0.954 |  |  | $2 \cdot \mathrm{OII}$ |
| $\mathrm{O}(5) \mathrm{H}$ |  |  |  | I'993(8) |  |  |  |  |
|  |  |  |  | $0.524 \dagger$ |  |  |  | 1.048 |
| O(6) |  | 2.826(8) |  | 2.003(7) |  |  | 1.673(8) |  |
|  |  | 0.153 |  | $0.510 \dagger$ |  |  | 0.936 | 2-109 |
| $\mathrm{O}(7) \mathrm{H}$ |  |  |  | I.969(8) |  |  |  |  |
|  |  |  |  | $0.559 \dagger$ |  |  |  | I-II7 |
| $\mathrm{O}(8)$ | 2.537(8) |  |  |  | 1.672(8) | 1.669(8) |  |  |
|  | 0.243 |  |  |  | 0.924 | 0.953 |  | 2.119 |
| O(9) |  | 2.446(8) | 2-103(7) |  |  |  | I.643(8) |  |
|  |  | 0.289 | 0.392* $\dagger$ |  |  |  | 0.998 | 2.072 |
| $\mathrm{O}(10) \mathrm{H}$ |  | $2.453(8)$ |  |  |  | I•644(8) |  | (1.22 estim.) |
|  |  | 0.287 |  |  |  | I.005 |  | I-29I |
| $\mathrm{O}(\mathrm{iI}) \mathrm{H}$ | 2.138(8) |  | I•993(8) |  |  |  |  | (r.30 estim.) |
|  | $0 \cdot 320$ |  | 0.534* $\dagger$ |  |  |  |  | $1 \cdot 389$ |
| $\bar{L}$ | 2.413 | 2.461 | 2.084 | 2.011 | 1.635 | 1.646 | 1.642 |  |
| $L$ | 3.250 | 3.250 | 2.448 | 2.390 | 2.130 | 2.130 | 2.130 |  |
| $p$ | 2.883 | 3.119 | $5 \cdot 719$ | 5.309 | $3 \cdot 301$ | $3 \cdot 403$ | $3 \cdot 365$ |  |
| $\nu_{i}$ | 2.00/7 | 2.00/7 | 2.52/6 | 3.00/6 | 4.00/4 | 4.00/4 | 4.00/4 |  |
| $N F$ | 0.998 | 0.993 | 0.987 | 0.998 | 0.999 | 1.000 | 1.000 |  |
| $\Sigma_{\text {A }} v$ | 2.000 | 2.000 | 2.520 | 3.000 | 4.000 | 4.000 | 4.000 |  |
| For comparison: $\bar{L}$ of pumpellyite |  |  |  |  |  |  |  |  |
|  | $2 \cdot 42$ | 2.45 | 1-99 | I 92 | 1. 63 | 1.64 | 1.645 |  |

* Two bonds per cation. $\dagger$ Two bonds per anion.
radii of cation and anion. These radii, $r_{\text {max }}$, are obtained by extrapolating $v_{i} v s r$ to zero $v_{1}$ ( $r$ refers to effective ionic radii given for different coordination numbers by Shannon and Prewitt, 1969). The exponential $p$, used to evaluate $v$ for bond lengths less than the mean, is chosen so as to avoid a singularity in slope where the two curve fragments meet; thus its value is $p=\bar{L} /\left(L_{\max }-\bar{L}\right)$. Because of the non-linear relation on the short bond length side of $\bar{L}$, the $\Sigma_{A} v$ values for the cations need a slight normalization to the ideal charge values of the cations, as indicated by the normalization factors in the penultimate row of table V .

From the magnitude of anionic valence sums ( $\Sigma_{\mathrm{C}} v$, table V ), we infer that $\mathrm{O}(5)$ and $\mathrm{O}(7)$, corners of the $\mathrm{Fe}(2)$ octahedron (fig. I), are hydroxyl groups. The other two anions with valence sums closer to I than 2 are $\mathrm{O}(10)$ and $\mathrm{O}(1 \mathrm{I})$, of which the former represents the free corner of the $\mathrm{Si}(2)$ tetrahedron and the latter, a corner of the $\mathrm{Fe}(\mathrm{I})$ octahedron. The values of $\mathrm{r} \cdot 29$ and r .39 valence units respectively leave no doubt that
the major occupants of these positions are hydroxyl groups, with some $\mathrm{O}^{2-}$ for- $\mathrm{OH}^{-}$ substitution. It is rare to find a silicon tetrahedron with a hydroxyl group at a corner, and we are therefore not surprised that partial substitution of O for OH takes place at the O (Io) position.

Table VI. Lengths of edges of coordination polyhedra, angles subtended by edges at central cations, cation approaches, and possible hydrogen bond lengths.

$$
\left(\sigma(\mathrm{O}-\mathrm{O}) \leqslant 0.01 \AA, \sigma(\angle) \leqslant 0.1^{\circ}\right)
$$

| around Fe (I) | $d(\mathrm{O}-\mathrm{O}) \angle \mathrm{O}-\mathrm{Me}-\mathrm{O}$ |  | around $\mathrm{Si}(\mathrm{I})$ | $d(\mathrm{O}-\mathrm{O}) \angle \mathrm{O}-\mathrm{Me}-\mathrm{O}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{O}(2)-\mathrm{O}(9)$ | $3.02 \AA$ | $90.2{ }^{\circ}$ | $\mathrm{O}(\mathrm{I})-\mathrm{O}(\mathrm{I})$ | 2.69 A | $114.0^{\circ}$ |
| $\mathrm{O}(2)-\mathrm{O}\left(9^{\prime}\right)$ | 3.01 | $89 \cdot 8$ | $\mathrm{O}(\mathrm{I})-\mathrm{O}(4)$ | 2.69 | 111.2 |
| $\mathrm{O}(2)-\mathrm{O}\left(\mathrm{II}^{\prime}\right)$ | 2.92 | $89 \cdot 5$ | $\mathrm{O}(\mathrm{I})-\mathrm{O}(8)$ | $2 \cdot 67$ | 109.I |
| $\mathrm{O}(2)-\mathrm{O}(\mathrm{II})$ | $2 \cdot 95$ | 90.5 | $\mathrm{O}(4)-\mathrm{O}(8)$ | $2 \cdot 58$ | 101.6 |
| $\mathrm{O}(9)-\mathrm{O}\left(\mathrm{II}^{\prime}\right)$ | 2.73* | $83 \cdot 7$ |  |  |  |
| $\mathrm{O}(9)-\mathrm{O}(\mathrm{II})$ | 3.05 | $96 \cdot 3$ | around $\mathrm{Si}(2)$ |  |  |
|  |  |  | $\mathrm{O}(2)-\mathrm{O}(2)$ | 2.79 A | $116.8^{\circ}$ |
| around Fe (2) |  |  | $\mathrm{O}(2)-\mathrm{O}(8)$ | 2.67 | 1078 |
| $\mathrm{O}(\mathrm{I})-\mathrm{O}(4)$ | 2.90 A | $90.1{ }^{\circ}$ | $\mathrm{O}(2)-\mathrm{O}(10)$ | $2 \cdot 69$ | $110 \cdot 1$ |
| $\mathrm{O}(\mathrm{I})-\mathrm{O}(5)$ | 2.87 | 91.8 | $\mathrm{O}(8)-\mathrm{O}(\mathrm{IO})$ | $2 \cdot 60$ | 103.3 |
| $\mathrm{O}(\mathrm{I})-\mathrm{O}(6)$ | $2 \cdot 83$ | $89 \cdot 7$ |  |  |  |
| $\mathrm{O}(\mathrm{I})-\mathrm{O}(7)$ | $2 \cdot 75$ | 87.5 | around $\mathrm{Si}(3)$ |  |  |
| $\mathrm{O}(3)-\mathrm{O}(4)$ | $2 \cdot 85$ | $88 \cdot 1$ | $\mathrm{O}(3)-\mathrm{O}(3)$ | $2.67 \AA$ | $110.1{ }^{\circ}$ |
| $\mathrm{O}(3)-\mathrm{O}(5)$ | 2.77 | 87.7 | $\mathrm{O}(3)-\mathrm{O}(6)$ | $2 \cdot 71$ | $110 \cdot 4$ |
| $\mathrm{O}(3)-\mathrm{O}(6)$ | $2 \cdot 88$ | $92 \cdot$ I | $\mathrm{O}(3)-\mathrm{O}(9)$ | $2 \cdot 72$ | 112.5 |
| $\mathrm{O}(3)-\mathrm{O}(7)$ | $2 \cdot 88$ | $92 \cdot 9$ | $\mathrm{O}(6)-\mathrm{O}(9)$ | $2 \cdot 55$ | $100 \cdot 8$ |
| $\mathrm{O}(4)-\mathrm{O}(5)$ | 3.05 | $96 \cdot 5$ |  |  |  |
| $\mathrm{O}(4)-\mathrm{O}(7)$ | 2.63* | $80 \cdot 8$ | around $\mathrm{Ca}(\mathrm{I})(<$ | 3 A) |  |
| $\mathrm{O}(5)-\mathrm{O}(6)$ | 2.63* | 82.4 | $\mathrm{O}(2)-\mathrm{O}(2)$ | 3.29 A | $84.9{ }^{\circ}$ |
| $\mathrm{O}(6)-\mathrm{O}(7)$ | 3.05 | $100 \cdot 3$ | $\mathrm{O}(2)-\mathrm{O}(3)$ | $3 \cdot 25$ | $85 \cdot 1$ |
|  |  |  | $\mathrm{O}(3)-\mathrm{O}(3)$ | (3.42) | 92.4 |
|  |  |  | $\mathrm{O}(3)-\mathrm{O}(\mathrm{II})$ | $2 \cdot 93$ | 77.3 |
| $\mathrm{Fe}(\mathrm{I})-\mathrm{Fe}(\mathrm{I})$ | 3.041 A |  | around $\mathrm{Ca}(2)(<3.4 \AA)$ |  |  |
| $\mathrm{Fe}(2)-\mathrm{Fe}(2)$ | $\left\{\begin{array}{l} 3.083 \\ 2.998 \end{array}\right.$ |  |  |  |  |
| $\mathrm{Si}(1)-\mathrm{Si}(2)$ | 12.9983.031 |  | $\mathrm{O}(2)-\mathrm{O}(2)$ | 3.29 | 93.5 85.5 |
| $\mathrm{Fe}(2)-\mathrm{Si}(\mathrm{I})$ | 3.198 |  |  |  |  |
| $\mathrm{Fe}(2)-\mathrm{Si}(3)$ | 3.168 |  | $d(\mathrm{O}-\mathrm{O})$ of possible hydrogen bonds |  |  |
| $\mathrm{Ca}(\mathrm{I})-\mathrm{Si}(\mathrm{I})$ | 3.174 |  |  |  |  |
|  |  |  | $\mathrm{O}(\mathrm{I}) \ldots \mathrm{O}(5)$ | $2 \cdot 86$ A |  |
| $\angle(\mathrm{Si}(\mathrm{r})-\mathrm{O}(8)-\mathrm{Si}(2))$ | $130 \cdot{ }^{\circ}$ |  | $\mathrm{O}(3) \ldots \mathrm{O}(7)$ | $2 \cdot 93$ |  |
|  |  |  | $\left.\mathrm{O}(5) \ldots \mathrm{O} 5^{\prime}\right)$ | $2 \cdot 97$ |  |
|  |  |  | $\mathrm{O}(5) \ldots \mathrm{O}$ (10) | $2 \cdot 76$ |  |
|  |  |  | $O(7) \ldots O\left(7^{\prime}\right)$ | $2 \cdot 71$ |  |
|  |  |  | $\mathrm{O}(7) \ldots \mathrm{O}(\mathrm{II})$ | $2 \cdot 98$ |  |

* Shared edge.

Next we investigate the oxygen-oxygen approaches below $3 \cdot 1 \AA$ (table VI) to see what hydrogen bonding is possible: 3.0 out of the 3.48 hydrogen atoms per formula unit can be placed between $\mathrm{O}(\mathrm{IO}) \ldots \mathrm{O}(5) \ldots \mathrm{O}\left(5^{\prime}\right) \ldots \mathrm{O}\left(\mathrm{IO}^{\prime}\right)$ and between $\mathrm{O}(\mathrm{Ir}) \ldots$ $\mathrm{O}(7) \ldots \mathrm{O}\left(7^{\prime}\right) \ldots \mathrm{O}\left(\mathrm{I} I^{\prime}\right)$ (for distances see table VI and fig. I; for angles, table VII), both sets of atoms possessing a twofold symmetry axis. Because the bonds are too
large for symmetrical hydrogen bonds, the proton arrangement must be disordered as shown for the first group: $\mathrm{O}(\mathrm{I} 0)-\mathrm{H} . . \mathrm{O}(5)-\mathrm{H} . . \mathrm{O}\left(5^{\prime}\right)-\mathrm{H} \ldots \mathrm{O}\left(10^{\prime}\right)$ or $\mathrm{O}(10) \ldots$ $\mathrm{H}-\mathrm{O}(5) \ldots \mathrm{H}-\mathrm{O}\left(5^{\prime}\right) \ldots \mathrm{H}-\mathrm{O}\left({ }^{\prime} \mathbf{o}^{\prime}\right)$. Because of this disordered arrangement, no net charge transfer occurs within these hydrogen bond chains.

Table VII. Interbond angles at hydrogen bonded oxygen atoms

| Angles at $\mathrm{O}(\mathrm{x})$ | Angles at O(3) |
| :---: | :---: |
| $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Ca}(2) \quad=110 \cdot 6^{\circ}$ | $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Ca}(\mathrm{I}) \quad=101.9{ }^{\circ}$ |
| $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Si}(\mathrm{r}) \quad=124.3$ | $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Si}(3) \quad=12 \mathrm{I} \cdot 3$ |
| $\mathrm{Fe}(2)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(5)=105.9$ | $\mathrm{Fe}(2)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(7)=97 \cdot \mathrm{I}$ |
| $\mathrm{Ca}(2)-\mathrm{O}-\mathrm{Si}(\mathrm{I}) \quad=122.6$ | $\mathrm{Ca}(1)-\mathrm{O}-\mathrm{Si}(3)=134.0$ |
| $\mathrm{Ca}(2)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(5)=79.7$ | $\mathrm{Ca}(\mathrm{I})-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(7)=92.4$ |
| $\mathrm{Si}(1)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(5)=98 \cdot 8$ | $\mathrm{Si}(3)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(7)=96.9$ |
| Angles at $\mathrm{O}(5) \mathrm{H}$ | Angles at $\mathrm{O}(7) \mathrm{H}$ |
| $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Fe}\left(2^{\prime}\right) \quad=97.5^{\circ}$ | $\mathrm{Fe}(2)-\mathrm{O}-\mathrm{Fe}\left(2^{\prime}\right) \quad=103 \cdot 1^{\circ}$ |
| ${ }^{*} \mathrm{Fe}(2)-\mathrm{O}(5)-\mathrm{H} \ldots \mathrm{O}\left(5^{\prime}\right)=102.4(2 \times)$ | $* \mathrm{Fe}(2)-\mathrm{O}-\mathrm{H} \ldots \mathrm{O}\left(7^{\prime}\right)=105.5(2 \times)$ |
| $\mathrm{Fe}(2)-\mathrm{O}(5) \ldots \mathrm{H}-\mathrm{O}(10)=114.9(2 \times)$ | $\mathrm{Fe}(2)-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(\mathrm{II})=118.4(2 \times)$ |
| $\mathrm{O}\left(5^{\prime}\right)-\mathrm{H}-\mathrm{O}(5) \ldots \mathrm{H}-\mathrm{O}(10)=12 \mathrm{I} \cdot 3$ | $\mathrm{O}\left(7^{\prime}\right) \ldots \mathrm{H}-\mathrm{O} \ldots \mathrm{H}-\mathrm{O}(\mathrm{II})=104.8$ |
| Angles at $\mathrm{O}(\mathrm{ro}) \mathrm{H}^{\dagger}$ | Angles at $\mathrm{O}(\mathrm{II}) \mathrm{H}$ |
| $\mathrm{Ca}(2)-\mathrm{O}-\mathrm{Si}(2) \quad=144.5{ }^{\circ}$ | $\mathrm{Ca}(\mathrm{I})-\mathrm{O}-\mathrm{Fe}(\mathrm{I}) \quad=101 \cdot 1^{\circ}(2 \times)$ |
| $\mathrm{Ca}(2)-\mathrm{O}-\mathrm{H} \ldots \mathrm{O}(5)=79.8$ | $\mathrm{Ca}(\mathrm{I})-\mathrm{O}-\mathrm{H} \ldots \mathrm{O}(7)=92.3$ |
| $\mathrm{Si}(2)-\mathrm{O}-\mathrm{H} \ldots \mathrm{O}(5) \quad=\mathrm{r} 35 \cdot 7$ | $\mathrm{Fe}(\mathrm{I})-\mathrm{O}-\mathrm{Fe}\left(\mathrm{I}^{\prime}\right) \quad=99.4$ |
|  | $\mathrm{Fe}(\mathrm{I})-\mathrm{O}-\mathrm{H} \ldots \mathrm{O}(7) \quad=127.5(2 \times)$ |

[^3]If both O (Io) and O (Io') (or O (II) and $\mathrm{O}\left(\mathrm{II}^{\prime}\right)$ ) are hydroxyl groups ( 56 and $40 \%$ respectively of all cases), the above chains are impossible. In this case the $3^{-}$ coordinated oxygen ions $O(\mathrm{r})$ and $\mathrm{O}(3)$ are possible acceptor ions, judging by their distance from $\mathrm{O}(5) \mathrm{H}$ and $\mathrm{O}(7) \mathrm{H}$ respectively, but among the interbond angles that would be subtended at the latter hydroxyl ions, assuming linear hydrogen bonds, are $\mathrm{Fe}(2)-\mathrm{O}(5)-\mathrm{H} \ldots \mathrm{O}(1)$ of $160^{\circ}$ and $\mathrm{Fe}(2)-\mathrm{O}(7)-\mathrm{H} \ldots \mathrm{O}(3)$ of $162^{\circ}$, which deviate too far from tetrahedral angles to be acceptable. The bifurcated hydrogen bonds

on the other hand, leave the hydrogen atoms on the mirror planes and reduce the above angles to $\mathbf{1 2 8 ^ { \circ }}$. Note that the high B values for the four hydroxyl groups (table II) may well be due to slight positional disorder of these atoms. A neutron-diffraction study is needed to test our predictions for the disordered hydrogen bonds.

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[Manuscript received 2 October 1972]


[^0]:    ${ }^{1}$ Most reflections were measured twice. The symmetry-related intensities varied less than $10 \%$, except for some weak reflections.

[^1]:    ${ }^{\text {I }}$ A table of $F_{\text {obs }} F_{\text {cale }}$ has been deposited in the library of the Dept. of Mineralogy, British Museum (Natural History), London SW 7; copies may be obtained on request.

[^2]:    * Observed data from the literature (Moore, 1971) taken with $\mathrm{Fe} / \mathrm{Mn}$ radiation ( $\sigma$ ( $\theta$ ) assumed as $0.05^{\circ}$ ). Calculated data (E. Evans, private communication) based on $\mathrm{Fe}-K_{\alpha}$ radiation. $I_{\text {calc }}$ is an integrated intensity based on $I(115)+I(300)=100$. All $I_{\text {calc }} \geqslant 4 \circ 0$ are listed out to $d_{\mathrm{min}}=\mathrm{I} \cdot 10 \AA$.

[^3]:    * The central atom is assumed to be the donor atom, which will be true in half the cases. The angles remain unchanged for the other case.
    $\dagger$ Planar surrounding indicates $\mathrm{sp}^{2}$ state for $\mathrm{O}(\mathrm{r})$ ).

