

Article

Structural Characterization of the Body Frame and Spicules of a Glass Sponge

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Abstract: The nanostructure (atomic-scale structure) and water species in the body frame and spicules of the marine glass sponge, *Euplectella aspergillum*, collected from the sea floor around Cebu Island was characterized in detail by thermogravimetric differential thermal analysis, nuclear magnetic resonance spectroscopy, Raman and infrared spectroscopies, and X-ray diffraction method. The structural features of the nanostructure in the body frame and spicules were essentially similar to each other, although these were different from those of inorganic amorphous silica materials, such as silica gel and silica glass. In addition, the averaged short and medium range structures of the sponge may be similar to those of tridymite. The water content and water species included in the body frame and spicules were almost the same. More than half of the contained water was physisorbed water molecules, and the rest was attributed to Q₃ and Q₂ silanol groups. Most of the water species may be present at the surface and involved in hydrogen bonding.

Keywords: sponge; biogenic silica; nanostructure; silanol; ring structure; amorphous silica; Raman spectroscopy

1. Introduction

The classes of marine sponges, *Demospongiae* and *Hexactinellida*, have the biomineralized siliceous component in their body [1]. The sponges of the class *Hexactinellida* are commonly called glass sponges. These glass sponges can live in any ocean [2] and their skeletons are composed of hydrated amorphous silica [3]. Glass sponges have interesting silicic fibrous root-like structures, which appear to grow from the bottom of their cylindrical body frame. This root-like part is called a spicule and anchors the sponge to the soft sediment of the sea floor [2,3]. The cross-sectional scanning electron microscope (SEM) images of the glass sponge (*Euplectella aspergillum*) revealed that the spicules were formed from consolidated spherical silica particles with a diameter of 50–200 nm [4]. Using the X-ray small angle scattering method, Woesz et al. [5] demonstrated that the small particles were composed of even smaller particles, which were less than 3 nm in diameter. These small particles were formed around a proteinaceous axial filament in the center of the spicule [4,6]. In addition, SEM observations showed that the body frame is basically formed from a bundling spicule with an extremely intricate construction [4,6]. However, few studies have examined the silica nanostructure (atomic-scale structure) of the body frame and spicule. Gendron-Badou et al. [7] examined the sponge spicules using infrared, ²⁹Si {¹H} cross-polarization magic angle spinning (CP-MAS) as well as ¹H and ²⁹Si MAS nuclear magnetic resonance (NMR) spectroscopies, which revealed that the spicules have a Si–O–Si network structure containing single and germinal silanol (Si–OH) groups.

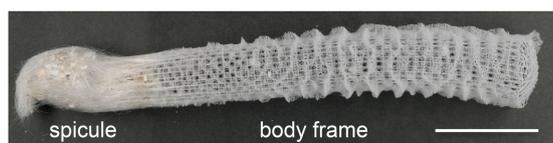


Figure 1. Photograph of the skeleton of *Euplectella aspergillum*. Scale bar is 5 cm.

2.2. TG-DTA

TG-DTA measurements for body frame, spicule, and silica gel were performed using a Rigaku Thermo Plus 2 TG 8120 instrument (Tokyo, Japan). Powdered samples (10 mg) were placed in a platinum pan and heated to 1400 °C at a heating rate of 10 °C/min under a nitrogen atmosphere.

2.3. NMR Spectroscopy

NMR measurements were performed using a JNM-ECX 500II (JEOL, Tokyo, Japan) spectrometer operating at resonance frequencies of 99.37 MHz and 500.16 MHz for ^{29}Si and ^1H , respectively. The powdered body frame of the sponge and silica gel were taken in a 3.2 mm zirconia rotor to perform ^{29}Si CP-MAS and ^1H static single pulse measurements. Both ^{29}Si and ^1H chemical shifts were referenced to the signal of tetramethylsilane (TMS). The ^{29}Si CP-MAS NMR spectra were collected with a $\pi/2$ pulse (2.73 μs) and high-power decoupling (HPD) using two-pulse phase-modulated decoupling [19] with a phase modulation angle of 15°. The ^1H decoupling frequency and spin-locking frequency were 73.3 kHz and 91.6 kHz, respectively. These spectra were collected using a contact time of 8 ms and a recycle delay of 1 s. A MAS speed of 10 kHz was employed for the samples. The ^1H static NMR spectra were collected with a recycle delay of 10 s. To estimate the water content of the samples from ^1H peak area (all ^1H peaks in the samples were assumed to be due to H_2O), adamantane was mixed with the sample as an internal reference. Unfortunately, the amount of spicule sample was not adequate to perform accurate NMR measurement in this study.

2.4. Raman and Infrared Spectroscopy

Raman spectra for all samples were recorded using a LabRAM HR800 spectrometer (Horiba Jobin Yvon, Kyoto, Japan) with 514.5 nm Ar laser light (Melles Griot, 43 Series Ion Laser, 543-GS-A02, Carlsbad, USA). A grating with 600 lines/nm provided a wavenumber resolution of 1.4–1.8 cm^{-1} and a spectral resolution of approximately $\pm 1.6 \text{ cm}^{-1}$ in the spectral range. Spectra were accumulated for 60 s in the range of $\nu = 50\text{--}1500 \text{ cm}^{-1}$ and $2500\text{--}3900 \text{ cm}^{-1}$. The body frame and spicule were set on clay perpendicularly with a longitudinal direction for analysis of cross-sectional portions of samples. The body frame samples for Raman analysis were prepared from the top, middle, and bottom of the sample. The bottom portion is near the anchoring spicule.

The Attenuated total reflection (ATR) infrared (IR) measurements for all samples were performed using a Nicolet iN10 spectrometer (Thermo Fisher Scientific, Tokyo, Japan) equipped with a diamond crystal. The ATR-IR spectra of the body frame and spicule samples were recorded in the range of $\nu = 600\text{--}4000 \text{ cm}^{-1}$. The band pass for all spectra was 4 cm^{-1} .

2.5. XRD Analysis

Powder XRD measurements for all samples were recorded using a Rigaku RINT 2200 diffractometer (Tokyo, Japan) with $\text{CuK}\alpha$ radiation under an applied acceleration voltage of 40 kV and current of 30 mA. A $\theta\text{--}2\theta$ scanning technique was used with a scan step of 0.05° with $2\theta = 2\text{--}120^\circ$ for sponge samples, silica gel, and silica glass. The total accumulation number was five for all measurements. For the samples after TG-DTA measurement, a $\theta\text{--}2\theta$ scanning technique was adopted with a scan speed of $1.00^\circ \text{ min}^{-1}$ with $2\theta = 2\text{--}60^\circ$.

3. Results

3.1. TG-DTA Curves

The TG curves for the sponge body frame and spicules showed that weight loss was similar for two distinct reduction steps (Figure 2) and are listed in Table 2. The total weight losses for the sponge samples were very similar. In the result of silica gel, the weight loss observed below 200 °C was about two times of those of sponge samples and the total weight loss was about 8% greater than those for the sponge samples.

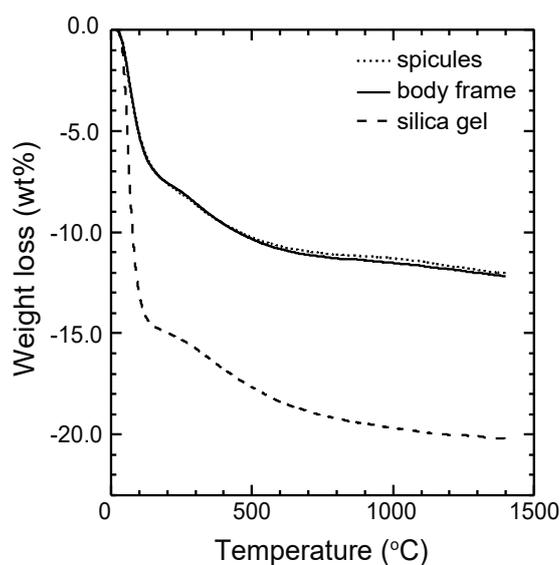


Figure 2. TG curves for body frame, spicules, and silica gel.

Table 2. Water content of samples estimated from TG results and ^1H NMR spectra.

Samples	Weight Loss Estimated from TG Results (wt %)				^1H NMR (wt %)
	RT-200	200–600	600–1400	Total	
body frame	7.25	3.89	0.88	12.02	10.5
spicules	7.37	3.73	0.94	12.04	-
silica gel	14.84	4.34	1.03	20.21	17.2

The DTA curves for the body frame and spicules have a broad endothermic band near 64–69 °C, which is similar to the intense band of silica gel observed at 67 °C (Figure 3). Strong exothermic bands were observed at 925 °C and 946 °C for the spicules and body frame, respectively. No obvious exothermic bands were observed for silica gel. In our preliminary experiment, we carried out the TG-DTA measurements for silica glass, although those curves for silica glass did not show the weight loss, the endothermic and exothermic bands.

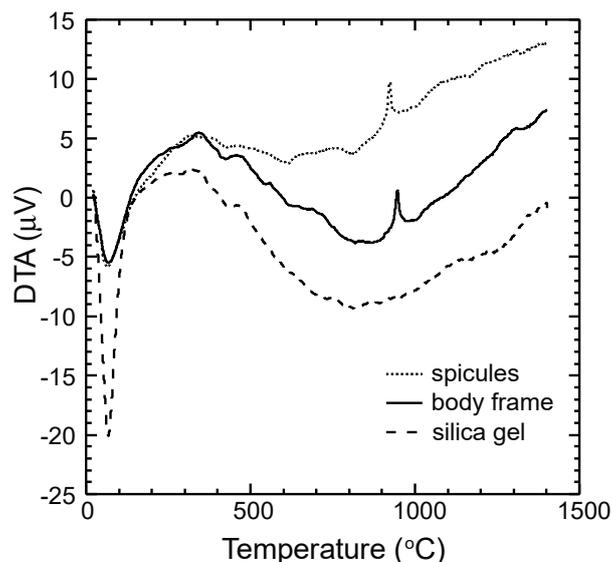


Figure 3. DTA curves for body frame, spicules, and silica gel.

3.2. ^1H Static and ^{29}Si $\{^1\text{H}\}$ CP-MAS NMR Spectra

The ^1H NMR spectrum for the sponge body frame is shown in Figure 4, along with the spectrum of silica gel. The ^1H NMR spectrum for the body frame had a signal at ca. 4.8 ppm, which was similar to that of silica gel. The physisorbed water molecules and/or silanol groups present on the surface of the sample were attributed to this peak [20–22]. The water content estimated from the ^1H peak was 10.5% for the body frame and 17.2% for silica gel (Table 2).

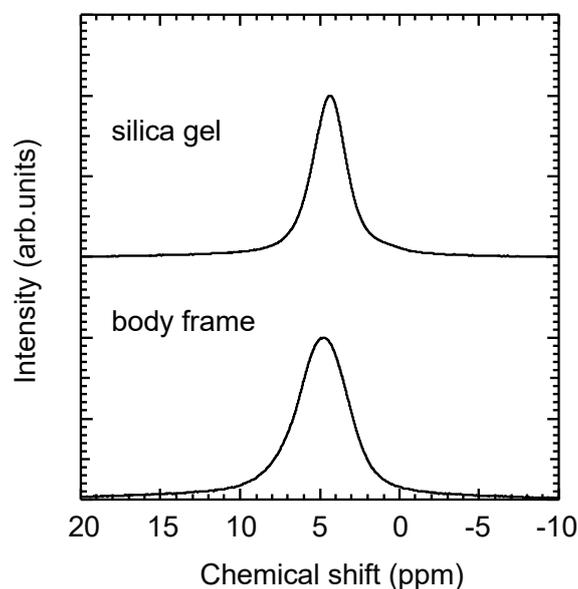


Figure 4. ^1H NMR spectra for sponge body frame and silica gel.

The ^{29}Si $\{^1\text{H}\}$ CP-MAS NMR spectrum for the body frame had three signals at ca. -92.4 , -101.4 , and -111.2 ppm (Figure 5). These signals usually represent Q_n species, where n is the number of bridging oxygens. These were assigned to Q_2 , Q_3 , and Q_4 , respectively. The positions of these signals were similar to those of the spicules reported in a previous study [7] and the silica gel measured in this study.

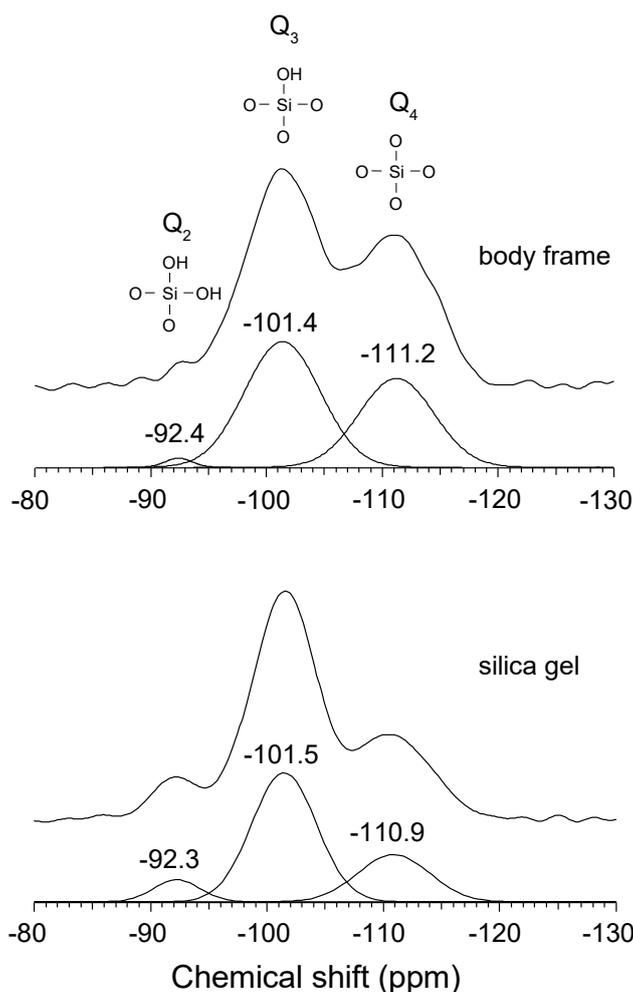


Figure 5. CP-MAS NMR spectra for sponge body frame and silica gel.

The Gaussian peak fitting of three signals observed in CP-MAS NMR spectra for sponge body frame and silica glass using IGOR Pro 6.3 software (Hulinks, Tokyo, Japan) showed the relative intensities of Q_n species (Table 3). Although the CP-MAS NMR technique is semi-quantitative, the relative intensity of $Q_4/(Q_2 + Q_3)$ for the sponge sample was determined to be two times greater than that of silica gel.

Table 3. Relative intensities of Q_n species.

Samples	Relative Intensities (%)			
	Q ₄	Q ₃	Q ₂	Q ₄ / (Q ₂ + Q ₃)
body frame	40.9	57.1	2.0	0.7
silica gel	26.6	65.3	8.1	0.4

3.3. Raman Spectra

The Raman spectra for the body frame obtained from each observation position (from top, middle, and bottom of body frame) were precisely consistent. Therefore, the Raman spectra for the middle part of body frame, spicule, silica gel, and silica glass are shown in Figure 6. The Raman spectra of samples had a broad band centered at $\nu = 450 \text{ cm}^{-1}$, which was attributed to the symmetrical Si–O–Si stretching mode and a D₁ band at $\nu = 480\text{--}490 \text{ cm}^{-1}$ due to the oxygen-breathing mode of the four-membered ring of SiO₄ tetrahedra [23–26]. The spectra of sponge samples and silica gel have a relatively sharp

band at $\nu = 960 \text{ cm}^{-1}$, which may be assigned to the Q_3 silanol as observed in hydrous amorphous silica materials [27,28]. The weak broad bands at ca. $\nu = 1060$ and 1200 cm^{-1} were observed in the spectra of sponge samples, which were corresponded to asymmetric Si–O stretching vibrations within the fully polymerized SiO_4 network [29]. A small band corresponding to the three-membered ring at ca. $\nu = 600 \text{ cm}^{-1}$ [17] observed for silica glass was not observed in spectra of the sponge and silica gel. Since the D_1 band is due to the main structure of the sponge and silica gel samples, it is assumed that the relative intensity of this band did not change significantly at the measurement point. Therefore, the intensities of spectra for all samples in Figures 6 and 7 were normalized to that of the D_1 band.

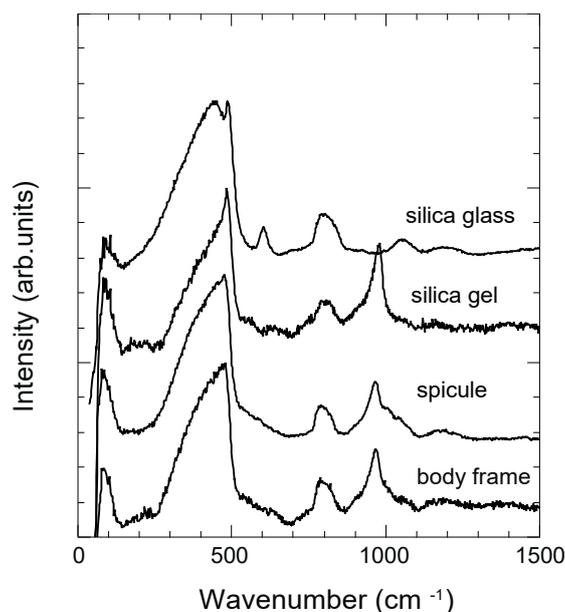


Figure 6. Raman spectra ($\nu = 50\text{--}1500 \text{ cm}^{-1}$) for body frame, spicule, silica gel, and silica glass.

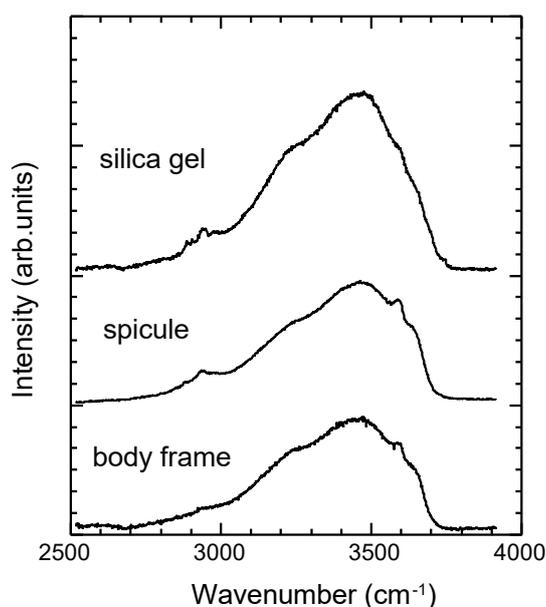


Figure 7. Raman spectra ($\nu = 2500\text{--}3900 \text{ cm}^{-1}$) for body frame, spicule and silica gel. The spectrum of silica glass had no peaks in this wavenumber region.

At higher wavenumbers, sponge samples and silica gel had a broad band in the range of $\nu = 3000\text{--}3800\text{ cm}^{-1}$ (Figure 7) due to overlap of the bands from molecular water and silanol groups [30,31]. The spectra for sponge samples were similar to each other. Compared with the spectrum of silica gel, the bands near $\nu = 3600$ and 3650 cm^{-1} attributed to silanol species [32] were more prominent in the sponge samples. A very small band at $\nu = 3750\text{ cm}^{-1}$, attributed to the vibration of isolated silanol at the surface [30,32], was observed only for silica gel.

3.4. ATR-IR Spectra

The ATR-IR spectra for sponge samples have three distinct peaks at $\nu = 1050, 950,$ and 795 cm^{-1} (Figure 8). The bands at $\nu = 1050$ and 795 cm^{-1} were attributed to the Si–O antisymmetric stretching band and Si–O–Si bending vibration, respectively [33]. The band that appeared in the range of $\nu = 1000\text{--}1300\text{ cm}^{-1}$ for silica glass was broader than other samples. The band at $\nu = 950\text{ cm}^{-1}$ was assigned to the stretching vibration of silanol groups [34]. A weak peak near $\nu = 1635\text{ cm}^{-1}$ was attributed to the H–O–H bending vibration of molecular water [35]. The IR spectra for sponge samples were basically similar to that of silica gel.

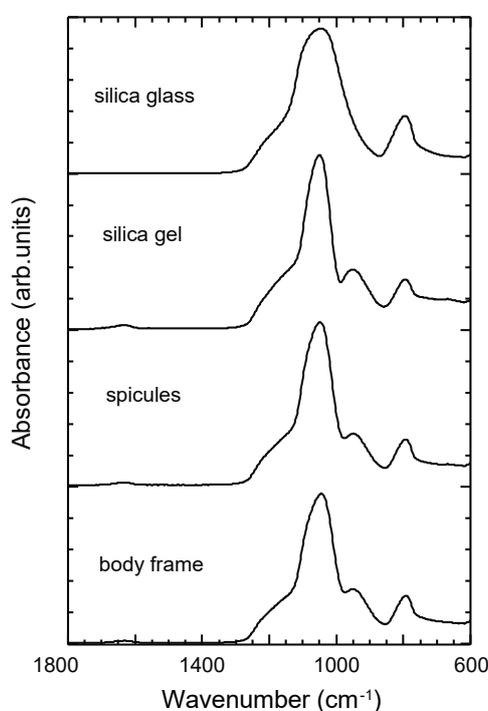


Figure 8. ATR-IR spectra ($\nu = 600\text{--}1800\text{ cm}^{-1}$) for body frame, spicules, silica gel, and silica glass.

The ATR-IR spectra in the range of $\nu = 2500\text{--}4000\text{ cm}^{-1}$ are shown in Figure 9. ATR-IR spectra for body frame, spicules, and silica gel have bands for water molecules, which appeared at $\nu = 3200$ and 3450 cm^{-1} [30,36,37], while the silanol group appeared at $\nu = 3600$ and 3650 cm^{-1} [30,32]. The intensities of spectra for all samples in the Figures 8 and 9 were normalized to the band intensity at ca. $\nu = 1050\text{ cm}^{-1}$.

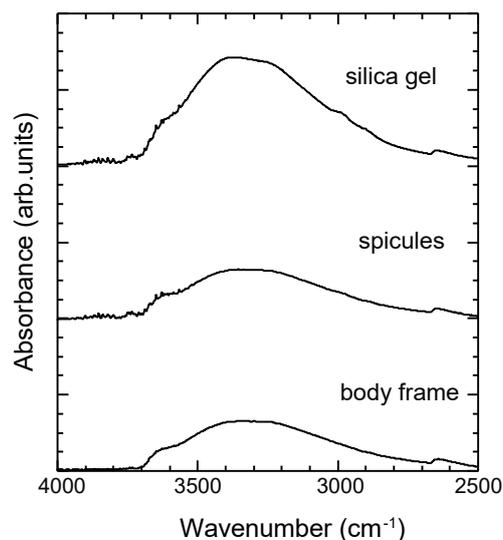


Figure 9. ATR-IR spectra ($\nu = 2500\text{--}4000\text{ cm}^{-1}$) for body frame, spicules, and silica gel. The spectrum of silica glass had no peaks in this wavenumber region.

3.5. XRD Analysis

The XRD patterns for the sponge body frame and spicules showed broad maxima centered at $2\theta = 22.7^\circ$ and 22.6° , respectively (Figure 10). This indicated that the sponge samples had no crystalline peak and thus, were similar to silica gel and silica glass.

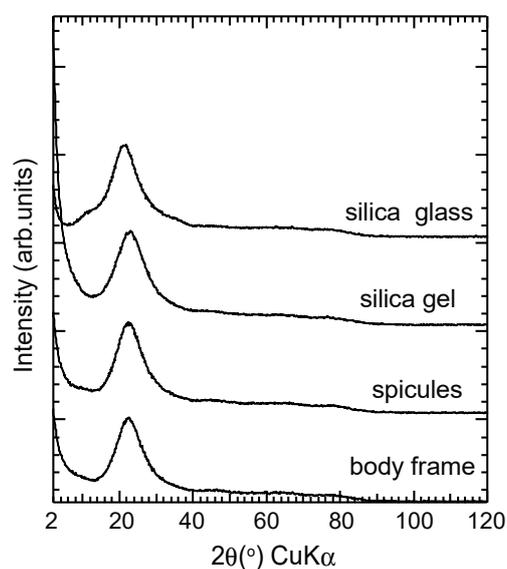


Figure 10. XRD patterns for body frame, spicules, silica gel, and silica glass.

4. Discussion

4.1. Water Content and Water Species in Sponge Samples

^1H NMR spectra and TG curves revealed that the water content for both spicules and body frames were 10–12 wt %, although the water content estimated by ^1H NMR was slightly less than that estimated by TG. This difference in estimated value was due to the TG curve being simply derived from sample weight loss corresponding to the release of water molecule and the dehydration of silanol,

while the water content was calculated from the integrated ^1H NMR peak area according only to H_2O . Moreover, the weight loss estimated from the TG measurement includes $\pm 0.2\%$ measurement error.

The TG curves for the body frame and spicules indicated that they underwent the same dehydration step. According to Graetsch et al. [38], the large weight loss up to $200\text{ }^\circ\text{C}$ can be attributed to the release of physisorbed water molecules, while the subsequent weight loss up to $600\text{ }^\circ\text{C}$ is due to the dehydration of silanol. The endothermic band that appeared at $64\text{--}69\text{ }^\circ\text{C}$ in the DTA curves corresponded to the release of water molecules on the sample surface. The small weight loss observed at temperatures above $600\text{ }^\circ\text{C}$ may be due to the dehydration of hydroxyl group located at the structural site [38]. The water content values estimated from each dehydration step for the body frame and spicules were similar, which indicates that the water species and their amounts were similar for both samples. The TG curve for silica gel showed that the weight loss at temperatures below $200\text{ }^\circ\text{C}$ was two times greater than those for sponge samples. However, the second weight loss of silica gel at $200\text{--}600\text{ }^\circ\text{C}$ was similar to that of sponges, which indicates that the difference in water content of silica gel and sponge samples was mainly due to the number of water molecules located at the surface. Micropores in the structure, which affects the number of water molecules that can be absorbed, may be smaller in sponge samples than in silica gel.

Bronnimann et al. [20] reported that the ^1H NMR spectrum for silica gel contained relatively sharp bands at 3.5, 3.0, and 1.7 ppm, respectively, which were attributed to physisorbed water molecules, H-bonded silanol, and isolated silanol, respectively. In the present study, the ^1H NMR spectrum for sponge sample had a band at ca. 4 ppm, which was narrower and less intense at $<2\text{ ppm}$ than that of silica gel. The Raman and ATR-IR spectra were consistent with the ^1H NMR of the sponge samples, which showed no evidence of isolated surface silanol (appearing ca. $\nu = 3750\text{ cm}^{-1}$). Thus, the Q_2 and Q_3 silanol species in the sponge samples have hydrogen bonds.

4.2. Nano-Structure of Spicules and Body Frame of Glass Sponges

The Raman bands of Si–O stretching vibration at $\nu = 1060$ and 1200 cm^{-1} were clearly observed in sponge samples, such as silica glass. However, the spectra of silica gel showed no evidence of these bands. This difference is also supported by the relatively greater intensity of $(\text{Q}_4/\text{Q}_3 + \text{Q}_2)$ of NMR signals for sponge samples compared to silica gel, which indicates the well-polymerized three-dimensional network of the sponge.

In general, the XRD patterns of amorphous silica materials show a broad scattering maximum centered at $2\theta = 22^\circ\text{--}23^\circ$, which is called the first sharp diffraction peak (FSDP). The position of the FSDP ($Q = 4\pi \sin \theta / \lambda$) can be estimated from XRD data, which assists in evaluating the size of the medium range structure [39]. A low Q value means the presence of large medium range structure in the sample. The values for the FSDP positions of sponge spicules and body frame are listed in Table 4. The FSDP position for silica gel used in this study is consistent with that of silica gel ($Q = 1.60\text{--}1.67\text{ \AA}^{-1}$) reported in Kamiya and Nasu [11]. As also described in the introduction, the medium range structure of silica gel is composed mainly of the four-membered ring of SiO_4 tetrahedra, although the structure of silica glass is made mainly of \geq six-membered ring. These results indicate that the medium range structure of the sponge spicules and body frame were similar, while the size may be smaller than that of silica glass but larger than that of silica gel.

Table 4. FSDP positions Q of X-ray scattering profiles for samples (Q is $4\pi \sin \theta / \lambda$; λ is wavelength of X-ray).

Samples	Q (\AA^{-1})
silica glass	1.51
silica gel	1.63
spicule	1.60
body frame	1.61

In addition, the intensity of Raman bands at $\nu < 600 \text{ cm}^{-1}$ clearly indicate the different features of the ring structures of silica gel, silica glass, and sponge samples (Figure 11a). The band attributed to a four-membered ring at around $\nu = 490 \text{ cm}^{-1}$ was present in all samples, although the band attributed to a three-membered ring at around $\nu = 600 \text{ cm}^{-1}$ was observed only in spectra for silica glass. The band below $\nu = 470 \text{ cm}^{-1}$ broadened in the order of silica glass > sponge > silica gel. This broad band may be a superposition of several bands for different ring structures and the appearance of the band at lower wavenumber may be attributed to the large ring structure [40,41]. Therefore, the result may indicate that the proportion of large ring structures included in sponge samples is less than that of silica glass. The wide distribution of ring structure in silica glass may be associated with the broad IR bands, which are attributed to the Si–O stretching mode that appears in the range of $\nu = 1000\text{--}1300 \text{ cm}^{-1}$. This broad band is formed from the superposition of several bands of different Si–O stretching modes [42]. Therefore, the various ring structure included in the silica glass may affect the broadness for this band. In the present study, Raman bands for the sponge body frame, silica glass, and silica gel in the wavenumber region of $\nu = 100\text{--}700 \text{ cm}^{-1}$ underwent detailed analysis based on the bands of crystalline SiO_2 polymorphs, such as quartz, cristobalite, and tridymite [41] using Gaussian peak fitting with the IGOR Pro 6.3 software (Figure 11b–d). Since the Raman spectra for spicule and body frame were very similar, peak fitting was performed only on that of body frame. When compared to silica glass, the fitting results indicated that bands below $\nu = 300 \text{ cm}^{-1}$ were weaker in the sponge sample compared with the normalized band of four-membered ring ($\nu = 482 \text{ cm}^{-1}$). In addition, the band ca. $\nu = 425 \text{ cm}^{-1}$ was intense for sponge samples, although the band at ca. $\nu = 460 \text{ cm}^{-1}$ observed in spectra for silica glass and silica gel was more intense. Kingma and Hemley [41] assigned the bands $<480 \text{ cm}^{-1}$; low-tridymite and low-cristobalite, including a six-membered ring, produced the strongest band at wavenumbers lower than ca. $\nu = 420 \text{ cm}^{-1}$, although the low-quartz produced an intense band near $\nu = 460 \text{ cm}^{-1}$. Moreover, the bands that appeared ca. $\nu = 350$ and 460 cm^{-1} are not observed in the spectrum of low-cristobalite, although these bands are observed in the spectrum for low-tridymite [41]. These results indicate that the medium range structure of sponges was composed mainly of six-membered rings with a disordered low-tridymite like structure.

On the other hand, the XRD patterns for the sponge samples after the TG-DTA experiments were completely consistent with that of low-cristobalite (the strongest peak located at $2\theta = 22.0^\circ$, the secondary peak at $2\theta = 36.1^\circ$; smaller peaks at $2\theta = 28.5^\circ$ and 31.5° and peaks above 40° ; Figure 12). Therefore, DTA peaks at nearly 925°C in spicules and 946°C in the body frame may be related to this structural change into cristobalite by heating. Synthesized hydrous amorphous opal ($\text{SiO}_2 \cdot n\text{H}_2\text{O}$) with the medium range structure consisting of four-membered rings showed an exothermic band at ca. 1260°C , corresponding to the crystallization into cristobalite [43]. In addition, Wahl et al. [44] reported that the complete crystallization of silica gel to cristobalite occurred at 1400°C . Sponge samples crystallize into cristobalite at lower temperatures compared to silica gel and synthesized opal. In the temperature region of $870\text{--}1470^\circ\text{C}$, high-tridymite is a stable and high-cristobalite is a metastable phase [45,46]. The metastable phase with high free energy crystallizes prior to the stable phase with lower free energy minimum. Therefore, the metastable high-cristobalite phase may be crystallized first according to Ostwald's step rule. Moreover, high-cristobalite was converted to low-cristobalite during quenching below 250°C . This is the reason behind the structural change of sponge sample into low-cristobalite after TG-DTA experiment. The rapid crystallization of sponge samples may be due to their structural similarity to that of low-tridymite composed mainly of six-membered rings.

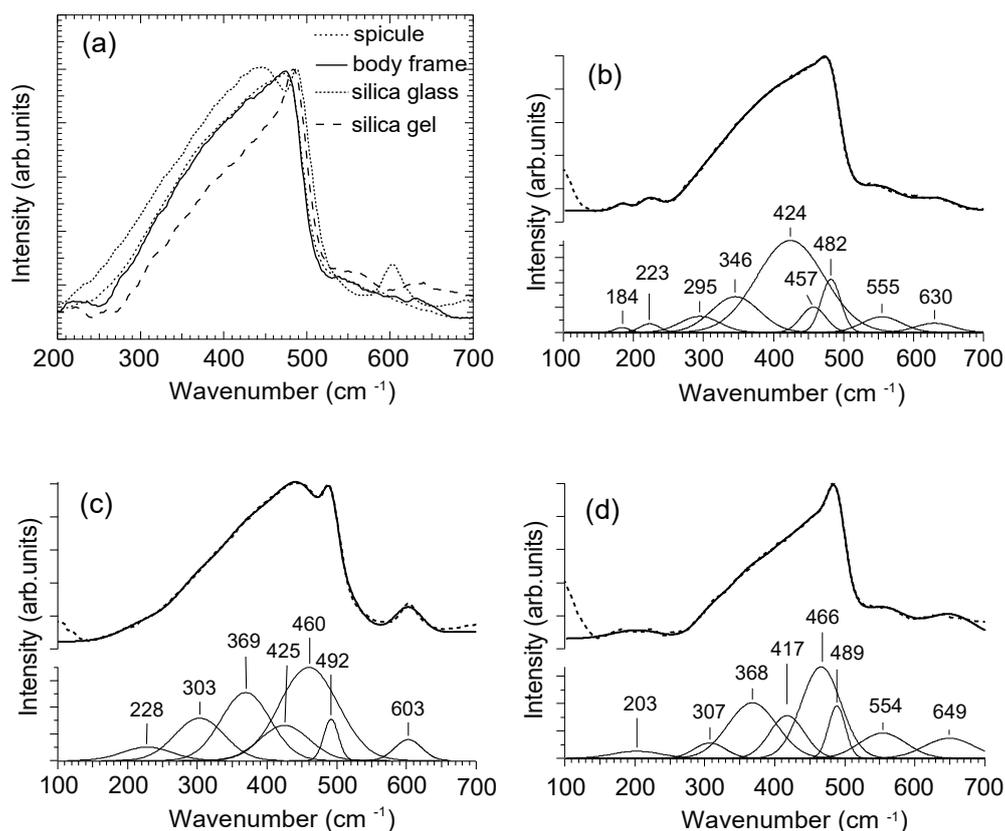


Figure 11. (a) Raman spectra ($\nu = 200\text{--}700\text{ cm}^{-1}$) for body frame, spicule, silica gel, and silica glass. These spectra were normalized to the D₁ band at ca. $\nu = 480\text{--}490\text{ cm}^{-1}$. The Gaussian peak fitting results for (b) body frame, (c) silica glass, and (d) silica gel. The dashed lines show the measured Raman spectra and the solid lines are Gaussian peak fitting results.

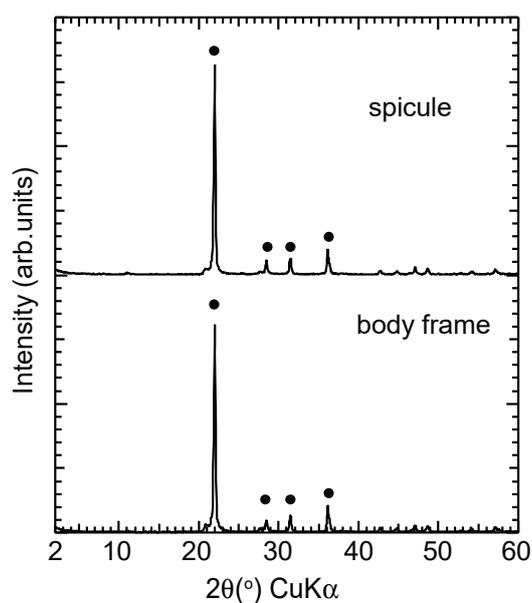


Figure 12. The XRD patterns for the sponge samples after TG-DTA measurements. *Solid circle* is the peak of cristobalite [47].

5. Concluding Remarks

The nanostructure and water species of the body frames and spicules of the marine sponge, *Euplectella aspergillum*, were determined and compared with other amorphous silica materials, such as silica gel and silica glass.

The structural features of the nano-silica network in the body frame and spicules were essentially similar, although these were different from those of silica gel and silica glass. The six-membered ring made of SiO₄ tetrahedra was the dominant component of the structure of sponge samples, which was similar to that of silica glass, although the ring size distribution was narrower than that of silica glass. In addition, the tridymite like six-membered ring structure was present in sponge samples without a long-range ordered structure.

The body frame and spicules contained similar water content and water species. The water greater than 60 wt% of the total water content in sponge samples was due to physisorbed water molecules at the surface, with the rest mainly attributed to silanol groups. In this study, silanols (Q₂ and Q₃) that are hydrogen-bonded to water molecules were detected at the surface. When compared to silica gel, the degree of polymerization of sponge samples appeared to be greater than that of silica gel.

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Conflicts of Interest: The authors declare no conflicts of interest.

References

1. Müller, W.E.G.; Wang, X.; Belikov, S.I.; Tremel, W.; Schloßmacher, U.; Natoli, A.; Brandt, D.; Boreiko, A.; Tahir, M.N.; Müller, I.M.; et al. Formation of Siliceous Spicules in Demosponges: Example *Suberites domuncula*. In *Handbook of Biomineralization Biological Aspects and Structure Formation*; Bäuerlein, E., Ed.; Wiley-VCH: Weinheim, Germany, 2007; pp. 59–82. ISBN 978-3-527-31804-9.
2. Ehrlich, H.; Worch, W. Collagen: A huge matrix in glass sponge flexible spicules of the meter-long *Hyalonema sieboldi*. In *Handbook of Biomineralization Biological Aspects and Structure Formation*; Bäuerlein, E., Ed.; Wiley-VCH: Weinheim, Germany, 2007; pp. 23–41. ISBN 978-3-527-31804-9.
3. Uriz, M.J.; Turon, X.; Becerro, G.; Ageli, G. Siliceous spicules and skeleton frameworks in sponges: Origin, diversity, ultrastructural patterns, and biological functions. *Microsc. Res. Tech.* **2003**, *32*, 186–193. [[CrossRef](#)] [[PubMed](#)]
4. Aizenberg, J.; Weaver, J.C.; Thanawala, M.S.; Sundar, V.C.; Morse, D.E.; Fratzl, P. Skeleton of *Euplectella* sp.: Structural hierarchy from the nanoscale to the macroscale. *Science* **2005**, *309*, 275–278. [[CrossRef](#)] [[PubMed](#)]
5. Woesz, A.; Weaver, J.C.; Kazanci, M.; Dauphin, Y.; Aizenberg, J.; Morse, D.E.; Fratzl, P. Micromechanical properties of biological silica in skeletons of deep-sea sponges. *J. Mater. Res.* **2006**, *21*, 2068–2078. [[CrossRef](#)]
6. Weaver, J.C.; Aizenberg, J.; Fantner, G.E.; Kisailus, D.; Woesz, A.; Allen, P.; Fields, K.; Porter, M.J.; Zok, F.W.; Hansma, P.K.; et al. Hierarchical assembly of the siliceous skeletal lattice of the hexactinellid sponge *Euplectella aspergillum*. *J. Struct. Biol.* **2007**, *158*, 93–106. [[CrossRef](#)] [[PubMed](#)]
7. Gendron-Badou, A.; Coradin, T.; Maquet, J.; Frohlich, F.; Livage, J. Spectroscopic characterization of biogenic silica. *J. Non-Cryst. Solids* **2003**, *316*, 331–337. [[CrossRef](#)]
8. Cha, J.N.; Shimizu, K.; Zhou, Y.; Christiansen, S.C.; Chmelka, B.F.; Stucky, G.D.; Morse, D.E. Silicatein filaments and subunits from a marine sponge direct the polymerization of silica and silicones in vitro. *Proc. Natl. Acad. Sci. USA* **1999**, *96*, 361–365. [[CrossRef](#)] [[PubMed](#)]
9. Shimizu, K.; Amano, T.; Bari, M.R.; Weaver, J.C.; Arima, J.; Mori, N. Glassin, a histidine-rich protein from the siliceous skeletal system of the marine sponge *Euplectella*, directs silica polycondensation. *Proc. Natl. Acad. Sci. USA* **2015**, *112*, 11449–11454. [[CrossRef](#)] [[PubMed](#)]
10. Fuchs, I.; Aluma, Y.; Ilan, M.; Mastai, Y. Induced crystallization of amorphous biosilica to cristobalite by silicatein. *J. Phys. Chem. B* **2014**, *118*, 2014–2111. [[CrossRef](#)] [[PubMed](#)]

11. Kamiya, K.; Nasu, H. Structure and thermal change of alkoxyderived silica gel fibers and films. *Ceram. Trans.* **1998**, *81*, 21–28.
12. Pasquarello, A.; Roberto, C. Identification of Raman defect Lines as signatures of ring structures in vitreous silica. *Phys. Rev. Lett.* **1998**, *80*, 5145–5147. [[CrossRef](#)]
13. Shimada, Y.; Okuno, M.; Syono, Y.; Kikuchi, M.; Fukuoka, K.; Ishizawa, N. An X-ray diffraction study of shock-wave-densified SiO₂ glasses. *Phys. Chem. Miner.* **2002**, *29*, 233–239. [[CrossRef](#)]
14. Huang, L.; Kieffer, J. Amorphous-amorphous transitions in silica glass. I. Reversible transitions and thermomechanical anomalies. *Phys. Rev. B* **2004**, *69*, 224203. [[CrossRef](#)]
15. Guerette, M.; Ackerson, M.R.; Thomas, J.; Yuan, F.; Watson, E.B.; Walker, D.; Huang, L. Structure and properties of silica glass densified in cold compression and hot compression. *Sci. Rep.* **2015**, *5*, 15343. [[CrossRef](#)] [[PubMed](#)]
16. Arasuna, A.; Okuno, M.; Chen, L.; Mashimo, T.; Okudera, H.; Mizukami, T.; Arai, S. Shock-wave compression of silica gel as a model material for comets. *Phys. Chem. Miner.* **2016**, *43*, 493–502. [[CrossRef](#)]
17. Okuno, M.; Reynard, B.; Shimada, Y.; Syono, Y.; Willaime, C. A Raman spectroscopic study of shock-wave densification of vitreous silica. *Phys. Chem. Miner.* **1999**, *26*, 304–311. [[CrossRef](#)]
18. Fukushima, Y. Structural Changes of Glass Sponge by Heat-Treatment and Compression. Master's Thesis, Kanazawa University, Kanazawa, Japan, 2018.
19. Bennett, A.E.; Rienstra, C.M.; Auger, M.; Lakshmi, K.V.; Griffin, R.G. Heteronuclear decoupling in rotating solids. *J. Chem. Phys.* **1995**, *103*, 6951. [[CrossRef](#)]
20. Bronnimann, C.E.; Zeigler, R.C.; Maciel, G.E. Proton NMR Study of Dehydration of the Silica Gel Surface. *J. Am. Chem. Soc.* **1988**, *110*, 2023–2026. [[CrossRef](#)]
21. Eckert, H.; Yesinowski, J.P.; Silver, L.A.; Stolper, E.M. Water in silicate glasses: Quantitation and structural studies by ¹H Solid echo and MAS-NMR Methods. *J. Phys. Chem.* **1988**, *92*, 2055–2064. [[CrossRef](#)]
22. Kinney, D.R.; Chuang, I.-S.; Marciel, G.E. Water and the silica surface as studied by variable-temperature High-Resolution ¹H NMR. *J. Am. Chem. Soc.* **1993**, *115*, 6786–6794. [[CrossRef](#)]
23. Galeener, F.L. Planar rings in vitreous silica. *J. Non-Cryst. Solids* **1982**, *49*, 53–62. [[CrossRef](#)]
24. Galeener, F.L. Planar rings in glasses. *Solid State Commun.* **1982**, *44*, 1037–1040. [[CrossRef](#)]
25. Galeener, F.L.; Geissberger, A.E. Vibrational dynamics in ³⁰Si substituted vitreous SiO₂. *Phys. Rev. B* **1983**, *27*, 6199–6204. [[CrossRef](#)]
26. Sharma, S.K.; Matson, D.W.; Philpotts, J.A.; Roush, T.L. Raman study of the structure of glasses along the join SiO₂-GeO₂. *J. Non-Cryst. Solids* **1984**, *68*, 99–114. [[CrossRef](#)]
27. Stolen, R.H.; Walrafen, G.E. Water and its relation to broken bond defects in fused silica. *J. Chem. Phys.* **1976**, *64*, 2623–2631. [[CrossRef](#)]
28. Murray, C.A.; Greytak, T.J. Intrinsic surface phonons in amorphous silica. *Phys. Rev. B* **1979**, *20*, 3368–3387. [[CrossRef](#)]
29. McMillan, P. Structural studies of silicate glasses and melts-applications and limitations of Raman spectroscopy. *Am. Mineral.* **1984**, *69*, 622–644.
30. Davis, K.M.; Tomozawa, M. An infrared spectroscopic study of water-related species in silica glasses. *J. Non-Cryst. Solids* **1996**, *201*, 177–198. [[CrossRef](#)]
31. Anedda, A.; Carbonaro, C.M.; Clemente, F.; Corda, L.; Corpino, R.; Ricci, P.C. Surface hydroxyls in porous silica: A Raman spectroscopy study. *Mater. Sci. Eng. C* **2003**, *23*, 1069–1072. [[CrossRef](#)]
32. Bergna, H.E. Colloid chemistry of silica: An overview. In *Colloidal Silica: Fundamentals and Applications*; Bergna, H.E., Roberts, W.O., Eds.; CRC Press: Boca Raton, FL, USA, 2006; pp. 9–35. ISBN 0-8247-0967-5.
33. Handke, M.; Mozgawa, W. Vibrational spectroscopy of the amorphous silicates. *Vib. Spectrosc.* **1993**, *5*, 75–84. [[CrossRef](#)]
34. Kamiya, K.; Oka, A.; Nasu, H.; Hashimoto, T. Comparative study of structure of silica gels from different sources. *J. Sol-Gel Sci. Technol.* **2000**, *19*, 495–499. [[CrossRef](#)]
35. Benesi, H.A.; Jones, A.C. An infrared study of the water-silica gel system. *J. Phys. Chem.* **1959**, *63*, 179–182. [[CrossRef](#)]
36. McDonald, R.S. Surface functionality of amorphous silica by infrared spectroscopy. *J. Phys. Chem.* **1958**, *62*, 1168–1178. [[CrossRef](#)]
37. Orcel, G.; Phalippou, J.; Hench, L.L. Structural changes of silica xerogels during low temperature dehydration. *J. Non-Cryst. Solids* **1986**, *88*, 114–130. [[CrossRef](#)]

38. Graetsch, H.; Flörke, O.W.; Miehe, G. The nature of water in chalcedony and opal-C from Brazilian agate geodes. *Phys. Chem. Miner.* **1985**, *12*, 300–306. [[CrossRef](#)]
39. Elliott, S.R. Medium-range structural order in covalent amorphous solids. *Nature* **1991**, *354*, 445–452. [[CrossRef](#)]
40. Sharma, S.K.; Mammone, J.F.; Nicol, M.F. Raman investigations of ring configurations in vitreous silica. *Nature* **1981**, *292*, 140–141. [[CrossRef](#)]
41. Kingma, K.J.; Hemley, R.J. Raman spectroscopic study of microcrystalline silica. *Am. Mineral.* **1994**, *79*, 269–273.
42. Innocenzi, P. Infrared spectroscopy of sol-gel derived silicabased films: A spectra-icrostructure orverview. *J. Non-Cryst. Solids* **2003**, *316*, 309–319. [[CrossRef](#)]
43. Arasuna, A.; Okuno, M.; Okudera, H.; Mizukami, T.; Arai, S.; Katayama, S.; Koyano, M.; Ito, N. Structural changes of synthetic opal by heat treatment. *Phys. Chem. Miner.* **2013**, *40*, 747–755. [[CrossRef](#)]
44. Wahl, F.M.; Grim, R.E.; Graf, R.B. Phase transformations in silica as examined by continuous X-ray diffraction. *Am. Mineral.* **1961**, *46*, 196–208.
45. Sosman, R.B. *The Properties of Silica*; The Chemical Catalog Company Inc.: New York, NY, USA, 1927; p. 856.
46. Sosman, R.B. New and old phase of silica. *Trans. Br. Ceram. Soc.* **1955**, *54*, 655–670.
47. Graetsch, H. Structural characteristics of opaline and microcrystalline silica minerals. *Rev. Mineral.* **1994**, *29*, 209–232.



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